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Lithium insertion/deinsertion of boron doped graphitic carbons synthesized by different procedure

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Abstract

The lithium insertion/deinsertion behaviour for a series of pure and boron doped graphites prepared from anthracene oil-derived pitch has been studied. Materials treated with boron were pitch-coke and graphitic carbon produced by heat-treatment at 1000°C and 2900°C, respectively. The powdered samples were mixed with 2 wt% of nanocrystalline boron and annealed at 2300°C for 0.5 h under argon. Additionally, posttreatment of samples in hydrogen at 700°C for 1 h was applied. XRD analysis was performed. Lithium intercalation/deintercalation was monitored using voltammetry and galvanostatic methods. The results show that boron is very effective catalyst of pitchcoke graphitization during heat-treatment at 2300°C. When added to graphitic carbon, boron induces slight improvement of structural ordering which has been established by preceding heat-treatment at 2900°C, but the final effect is inferior to that of heattreatment at 3100°C without boron. Boron seems to have no positive effect on lithium insertion, whatever the procedure is used. For all the graphitized materials lithium insertion/deinsertion occurs within a narrow potential range (0-0.5 V) with reversible capacity X_{rev} between 0.75 and 0.83 (where X stands for Li insertion degree in the Li_xC₆ compound, X = 1 corresponds to 372 mAh/g). Post-treatment of graphitic materials in hydrogen at 700°C has a profitable effect on lithium insertion/deinsertion giving the formation of stable passivating layer at first cycle and a decrease of irreversible capacity. Graphite produced by heat-treatment at 2900°C followed by treatment in hydrogen gives most promising electrochemical characteristics ($X_{rev} = 0.83$ and $X_{irr} = 0.10$).

Keywords: A. inorganic compounds, C. X-ray diffraction D. electrochemical properties

Introduction

Theoretical considerations suggest that the presence of boron substituted for carbon in the lattice should be beneficial for lithium insertion/deinsertion behaviour of carbon material. It is expected that boron possessing one electron less than carbon can locally modify an electronic structure increasing its electron acceptor character. However, due to various side effects associated with boron doping this assumption has not found a clear experimental confirmation, despite several attempts [1-3].

In the present work we study the lithium insertion/deinsertion characteristics for a series of pure and boron doped graphitic carbons obtained by different procedure.

Experimental

The materials for the synthesis of boron doped carbons were produced by heat treatment of anthracene oil derived pitch at 1000°C and 2900°C. The powdered samples were mixed with 2 wt% of nanocrystalline boron and annealed at 2300°C for 0.5 h under argon. Post-treatment in hydrogen at 700°C for 1 h was applied to remove some chemisorbed oxygen and to saturate possible dangling bonds. Respective boron-free graphites were used as reference materials.

Powder X-ray diffractograms were recorded with a Philips PW 3710 diffractometer using Ni radiation with graphite monochromator. The interlayer distance d_{002} was measured from the maximum of the (002) line. The apparent crystallite height L_c was determined from the width of the (002) band and the crystallite diameter L_a from the (10) or (110) bands. The L_{112} parameter was calculated from the width of the (112) band. The electrochemical Li intercalation/deintercalation was performed on pellets prepared by mixing and pressing 85% of carbon, 5% of acetylene black and 10% of PVDF and

mounted in a Li/carbon cell (Swagelok® system). Voltammetry (0.05 mV/s) and galvanostatic (20 mA/g) experiments were performed using a multichannel potentiostat/galvanostat Mac Pile II (Biologic, France).

Results and discussion

Optical texture of raw material, i.e. anthracene oil-derived pitch coke P-1 is shown in Fig. 1. Flow type anisotropy is well observed in this image. From this material two series of samples have been prepared where boron (2 wt%) was introduced thermally at 2300°C either into coke prepared at 1000°C or into previously graphitized carbon at 2900°C.

The materials graphitized using different procedures were assessed in terms of structural ordering using XRD analysis. Structural parameters of the graphitic carbons and boronated samples are shown in Table 1. It can be seen that boron independently of the form used (amorphous boron B_a , nanocrystalline boron B_c or as a compound B_4C) is at the origin of slight improvement of structural ordering, however, the final effect is inferior to that of heat-treatment at $3100^{\circ}C$ without boron.

Electrochemical characteristics (galvanostatic and voltammetry) allowed to estimate a degree of reversible lithium insertion X_{rev} as well as irreversibly stored part of lithium X_{irr} . Irreversible capacity X _{irr} (where X=1 for the charge of 372 mAh/g) is a difference between the Li amount inserted during the first reduction process and lithium removed during the first oxidation cycle. Reversible capacity X_{rev} corresponds to the lithium amount inserted/deinserted during subsequent cycles. For all the graphitized materials lithium storage occurs within a narrow potential range (0-0.5 V) with reversible capacity X_{rev} between 0.75 and 0.83. Post-treatment of graphitic materials in hydrogen

at 700°C has a profitable effect on lithium insertion/deinsertion. This is reflected mostly by a decrease of irreversible capacity and the formation of stable passivating layer (solid electrolyte interphase SEI) at first and subsequent cycles. For observation of SEI formation and the detailed qualitative analysis of insertion/deinsertion process, voltammetry characteristics at a slow scan rate of potential (0.05 mV/s) were especially adapted. Even if an essential insertion/deinsertion process proceeds in the narrow range of voltage (0-0.5 V vs Li) two significant peaks can be observed on voltammetry characteristics during de-intercalation suggesting existence of two domains where different energy is necessary for lithium extraction.

Galvanostatic intercalation/deintercalation into graphitic carbon obtained by heattreatment at 2900°C followed by treatment in hydrogen at 700°C (P-1-2900 H) gives most promising electrochemical characteristics, i.e. high reversible capacity $X_{rev} = 0.83$ and the lowest irreversible capacity $X_{irr} = 0.10$. It is noteworthy that stable SEI is already formed during the first cycle. Such characteristic has been presented in Fig. 2. On the other hand, it is well seen that the sample (P-1-2900/B-2300 H) with 2% of boron (Fig. 3) gives an increase of irreversible capacity with cycling which is unwanted phenomenon ($X_{irr}=0.2$) whereas reversible capacity remains the same $X_{rev}=0.83$. For all the boron doped samples their characteristics of lithium insertion/deinsertion are not in comparison to non-doped samples.

Conclusion

The effect of electron acceptor properties of boron substituted for carbon in graphitic carbons is negligible. On the other hand, presence of boron accelerates a graphitization process. Lithium insertion is rather determined by structural organization of carbon than

its electron properties modified with boron. Treatment by hydrogen slightly reduces irreversible capacity being at the origin of more stable passivating layer formed already during the first cycle.

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Figure captions

Figure 1. Flow type anisotropy of antracene oil derived-pitch coke observed by optical microscopy.

Figure 2. Galvanostatic charge/discharge characteristics of graphitic carbon P-1-2900 H. Current load of 20 mA/g.

Figure 3. Galvanostatic charge/discharge characteristics of boronated graphitic carbon P-1-2900/B-2300 H. Current load of 20 mA/g.

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Tables

 Table 1. Structural parameters of graphitic carbons and boronated graphitic

 carbons

Sample	d_{002} nm	L_c nm	L_a nm	L ₁₁₂ nm
P-1-2300	0.3385	33	52	4.2
P-1-2300B _a	0.3371	43	61	10.5
P-1-2900	0.3364	47	83	14.4
P-1-2900/B _c -2300	0.3357	52	60	12.2
P-1-2900/ B ₄ C-2300	0.3357	47	52	11.8
P-1-3100	0.3366	56	94	17.1

Figures







Fig. 2. Frackowiak, et al



Fig. 3. Frackowiak, et al