# Ln3[SiON3]O (Ln = La, Ce, Pr) - Three Oxonitridosilicate Oxides with Crystal Structures Derived from the Anti-Perovskite Structure Type 

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# $\mathrm{Ln}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}, \mathrm{Pr})$ - Three Oxonitridosilicate Oxides with Crystal Structures Derived from the Anti-Perovskite Structure Type 

Juliane A. Kechele, Christian Schmolke, Saskia Lupart, and Wolfgang Schnick*<br>Department Chemie und Biochemie der Ludwig-Maximilians-Universität München

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#### Abstract

The oxonitridosilicate oxides $\mathrm{Ln}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ with $\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}, \mathrm{Pr}$ were obtained by hightemperature syntheses ( $1450-1600{ }^{\circ} \mathrm{C}$ ) in a radio-frequency furnace. The crystal structures were determined from single-crystal X-ray diffraction data and confirmed by Rietveld refinements. The compounds $\mathrm{Ln}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ crystallize in space group I4/mcm $\left(\mathrm{La}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}\right.$ : $a=6.8224(10), \quad c=11.074(2) \AA, \quad Z=4 ; \quad R 1=0.0240 ; \quad \mathrm{Ce}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}: \quad a=6.7233(10)$, $c=11.069(2) \AA, Z=4, R 1=0.0234 ; \operatorname{Pr}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}: a=6.6979(9), c=11.005(2) \AA, Z=4$; $R 1=0.0156)$ and they were found to be isotypic with $\mathrm{Gd}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$, whose structure derives from the anti-perovskite structure type. In the crystal there are elongated $\mathrm{OLn}_{6}$-octahedra which are interconnected through common vertices. In the cavities of this network there are exclusively non-condensed $\mathrm{SiON}_{3}$ tetrahedra. ${ }^{29} \mathrm{Si}$ solid-state NMR investigations, lattice energy calculations (MAPLE) and EDX measurements confirmed the crystal structure and the chemical compositions. For $\mathrm{Ce}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$, magnetic susceptibility measurements were carried out and the paramagnetic behavior with an experimental magnetic moment of $2.2 \mu_{\mathrm{B}}$ per cerium atom indicates $\mathrm{Ce}^{3+}$.


## Keywords

Magnetic properties; Oxonitridosilicate oxide; Rietveld refinement; Solid-state NMR; Structure elucidation

## Introduction

In the literature there are multitudinous reports on varied possibilities for substitution and distortion in perovskite type materials. ${ }^{[1]}$ In this context, the oxonitridosilicate oxide $\mathrm{Gd}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}{ }^{[2]}$ has been investigated which is isotypic to $\mathrm{Ba}_{3}\left[\mathrm{SiO}_{4}\right] \mathrm{O}{ }^{[3,4]}$ and $\mathrm{Cs}_{3}\left[\mathrm{CoCl}_{4}\right] \mathrm{Cl} .{ }^{[5]}$ All three compounds can be derived from the anti-perovskite structure type in a hierarchical sense with tetrahedral building blocks $\left(\mathrm{SiON}_{3}, \mathrm{SiO}_{4}, \mathrm{CoCl}_{4}\right)$ substituting the large alkaline earth ions in perovskite.
Oxonitridosilicates are intermediates between classical oxosilicates and nitridosilicates and can be formally derived by partial substitution of O for N . While oxygen in classical oxosilicates usually occur in a terminal $\mathrm{O}^{[1]}$ or simply bridging $\mathrm{O}^{[2]}$ function, nitrogen atoms can additionally act as a triply $\left(\mathrm{N}^{[3]}\right)$ or even quadruply $\left(\mathrm{N}^{[4]}\right)$ bridging atom, leading to a manifold of additional structural possibilities. Consequently, nitridosilicates exhibit a more
variable degree of condensation in the range $1: 4 \leq \kappa \leq 3: 4$ (i.e. the molar ration $\mathrm{Si}:(\mathrm{N}, \mathrm{O})$ ) as compared to classical oxosilicates $(1: 4 \leq \kappa \leq 1: 2)$. Due to the enhanced connectivity of $\mathrm{N}^{[3]}$ and even $\mathrm{N}^{[4]}$ nitrogen atoms, new topological building blocks are possible. For example, in $\mathrm{BaYbSi}_{4} \mathrm{~N}_{7}$ a "star-shaped" unit $\left.\left[\mathrm{N}^{[4]}\left(\mathrm{SiN}_{3}\right)_{4}\right)\right]$ has been observed for the first time. ${ }^{[6]}$ Here, the quadruply bridging nitrogen atom $\left(\mathrm{N}^{[4]}\right)$ adopts ammonium character. A further structural motif, which is unknown for oxosilicates so far, is the highly-condensed dreier ring layer. ${ }^{[7]}$ This kind of layer can be found in the layer oxosilicates $\mathrm{MSi}_{2} \mathrm{O}_{2} \mathrm{~N}_{2}(\mathrm{M}=\mathrm{Ca}, \mathrm{Sr}, \mathrm{Ba}, \mathrm{Eu})^{[8-11]}$ or it is part of the silicate framework (e.g. in $\mathrm{M}_{2} \mathrm{Si}_{5} \mathrm{~N}_{8}{ }^{[12,13]}\left(\mathrm{M}=\mathrm{Ca}, \mathrm{Sr}, \mathrm{Ba}\right.$ ), $\mathrm{MSi}_{7} \mathrm{~N}_{10}$ $(\mathrm{M}=\mathrm{Sr}, \mathrm{Ba})) .{ }^{[14,15]}$ Most structures of (oxo)nitridosilicates analyzed so far, contained a highly condensed framework. ${ }^{[16,17]}$ However, the anionic part of the structure of $\mathrm{Gd}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ contains exclusively non-condensed $\mathrm{SiON}_{3}$ tetrahedra. Further examples for oxonitridosilicates with non-condensed silicate substructure are $\mathrm{Li}_{6}\left[\mathrm{SiN}_{2} \mathrm{O}_{2}\right],{ }^{[18]}$ $\mathrm{Ln}_{10}\left[\mathrm{SiO}_{3.67} \mathrm{~N}_{0.33}\right]_{6} \mathrm{O}_{2}(\mathrm{Ln}=\mathrm{La}, \mathrm{Nd}, \mathrm{Sm}, \mathrm{Gd}),{ }^{[19]} \mathrm{M}_{10}\left[\mathrm{SiO}_{2} \mathrm{~N}_{2}\right]_{6} \mathrm{O}_{2}(\mathrm{M}=\mathrm{Ti}, \mathrm{Ge}),{ }^{[19]}$ and $\mathrm{LnEu}\left[\mathrm{SiO}_{3} \mathrm{~N}\right](\mathrm{Ln}=\mathrm{La}, \mathrm{Nd}, \mathrm{Sm}) .{ }^{[20]}$
In this contribution, we report on the syntheses, crystal structures and magnetic properties of $\mathrm{Ln}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ with $\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}$, Pr. These compounds are isotypic to $\mathrm{Gd}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$, which has been described in the literature before.

## Results and discussion

Syntheses and sample characterization
As exemplified by the synthesis of $\mathrm{Pr}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$, the compounds $\mathrm{Ln}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}$, Pr ) can be obtained by various synthetical routes (cf. equation (1) - (5)) at temperatures ranging between 1450 and $1600^{\circ} \mathrm{C}$ with varying purity degrees. The metals $\mathrm{Ln}(\mathrm{La}, \mathrm{Ce}, \mathrm{Pr})$, different Ln oxides, silicon diimide, and $\mathrm{SiO}_{2}$ were used as starting materials for the syntheses. The oxidation state of the Ln in the starting material is apparently irrelevant. Starting from the respective metals leads to intermediate formation of the nitrides LnN $(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}, \mathrm{Pr})$.
$10 \mathrm{La}+4 \mathrm{La}_{2} \mathrm{O}_{3}+6$ " $\mathrm{Si}(\mathrm{NH})_{2}{ }^{"}+5 \mathrm{~N}_{2} \rightarrow 6 \mathrm{La}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}+4 \mathrm{NH}_{3}$
$12 \mathrm{Ce}+6 \mathrm{CeO}_{2}+6$ " $\mathrm{Si}(\mathrm{NH})_{2}{ }^{\prime}+5 \mathrm{~N}_{2} \rightarrow 6 \mathrm{Ce}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}+4 \mathrm{NH}_{3}$
$126 \mathrm{Pr}+12 \mathrm{Pr}_{6} \mathrm{O}_{11}+66$ " $\mathrm{Si}(\mathrm{NH})_{2} "+55 \mathrm{~N}_{2} \rightarrow 66 \mathrm{Pr}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}+44 \mathrm{NH}_{3}$
$6 \mathrm{Pr}+2 \mathrm{SiO}_{2}+3 \mathrm{~N}_{2} \rightarrow 2 \mathrm{Pr}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$
$18 \mathrm{Pr}+6$ " $\mathrm{Si}(\mathrm{NH})_{2} "+3 \mathrm{O}_{2} \rightarrow 6 \mathrm{Pr}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}+4 \mathrm{NH}_{3}$
The chemical composition and the phase purity of the samples were confirmed by EDX measurements and Rietveld refinements of the X-ray powder diffraction data (cf. Figure 1).

## Crystal structure description

The structure of $\mathrm{Ln}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}, \mathrm{Pr})$ is isotypic to $\mathrm{Gd}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$, which is derived from the anti-perovskite structure type. ${ }^{[2]}$ In Figure 2, the crystallographic relationship between the two structures is clarified using a group-subgroup scheme. Due to the hierarchical structural relation $\left(\mathrm{SiON}_{3}\right.$ units substitute for alkaline earth in the perovskite structure type) the complex $\mathrm{SiON}_{3}$ tetrahedra have been approximated by the coordinates of their tetrahedral center ( Si ).

The crystal structure of the compounds consists of a framework which is built up of exclusively vertex-sharing $\mathrm{OLn}_{6}$ octahedra $(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}, \mathrm{Pr})$. These octahedra are elongated along [001] and tilted as compared to the ideal cubic perovskite structure type (cf. Figure 3). The twisting is caused by electrostatic reasons or size effects, as the Ca atoms in the perovskite structure type are substituted by $\mathrm{SiON}_{3}$ tetrahedra in $\mathrm{Ln}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}$, $\mathrm{Pr})$. These $\mathrm{SiON}_{3}$ tetrahedra are exclusively non-condensed and fill the voids of the $\mathrm{OLn}_{6}$ octahedra framework. A differentiation of the O and N atoms was not possible by X-ray methods, as there is only one crystallographic position for these atoms belonging to the tetrahedra. Thus, the O and N atoms were distributed to the light atom positions on the basis of the crystal structure of the isotypic compound $\mathrm{Gd}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$. This distribution was corroborated by lattice energy calculations (MAPLE, Madelung part of lattice energy). ${ }^{[21,22]}$ The results of the calculations based on the $\mathrm{O} / \mathrm{N}$ distribution in $\mathrm{Gd}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ are summarized in Table 1.

The distances $\mathrm{Si}-(\mathrm{O}, \mathrm{N})$ (cf. Table 2) are in the typical range for terminal bonds $\mathrm{Si}-(\mathrm{O}, \mathrm{N})$ in oxonitridosilicates (cf. $\mathrm{La}_{16}\left[\mathrm{Si}_{8} \mathrm{~N}_{22}\right]\left[\mathrm{SiON}_{3}\right]_{2}: 1.689(5)-1.743(8) \AA \AA^{[25]} \mathrm{Ln}_{10}\left[\mathrm{Si}_{10} \mathrm{O}_{9} \mathrm{~N}_{17}\right] \mathrm{X}$ : $1.664(4)-1.709(7) \AA(\mathrm{Ln}=\mathrm{Ce}, \mathrm{X}=\mathrm{Br}), 1.666(3)-1.713(5) \AA(\mathrm{Ln}=\mathrm{Nd}, \mathrm{X}=\mathrm{Br}), 1.665(3)-$ 1.718 (5) $\AA \quad(\mathrm{Ln}=\mathrm{Nd}, \quad \mathrm{X}=\mathrm{Cl}) ;{ }^{[26]} \quad \mathrm{Pr}_{10}\left[\mathrm{Si}_{10-\mathrm{x}} \mathrm{Al}_{\mathrm{x}} \mathrm{O}_{9+\mathrm{x}} \mathrm{N}_{17-\mathrm{x}}\right] \mathrm{Cl} \quad(\mathrm{x} \approx 1): \quad 1.674(7)-$ $1.739(6) \AA$ ). ${ }^{[27]}$ The angles $\mathrm{N} / \mathrm{O}(1)^{[1]}-\mathrm{Si}(1)-\mathrm{N} / \mathrm{O}(1)^{[1]}$ (cf. Table 2) are close to the regular tetrahedral angle.

The structures of $\mathrm{Ln}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}, \mathrm{Pr})$ contain two crystallographically independent $\mathrm{Ln}^{3+}$ sites, which are surrounded by eight and ten ( $\mathrm{N}, \mathrm{O}$ ) atoms, respectively (cf. Figure 4). The distances $\mathrm{Ln}-(\mathrm{N}, \mathrm{O})(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}, \mathrm{Pr})$ are listed in Table 2. Some distances $\operatorname{Ln}(2)^{[8]}-(\mathrm{N}, \mathrm{O})$ are slightly shorter than the smallest sum of the ionic radii $\left(\mathrm{La}^{[8]}-\mathrm{O}: 2.56 \AA\right.$; $\left.\mathrm{La}^{[8]}-\mathrm{N}: 2.62 \AA ; \mathrm{Ce}^{[8]}-\mathrm{O}: 2.54 \AA ; \mathrm{Ce}^{[8]}-\mathrm{N}: 2.60 \AA ; \operatorname{Pr}^{[8]}-\mathrm{O}: 2.53 \AA ; \operatorname{Pr}^{[8]}-\mathrm{N}: 2.59 \AA\right)^{[28]}$ but the distances are in the same range than in other well characterized oxonitrido(alumo)silicates (e.g.: $\mathrm{La}_{4} \mathrm{Si}_{2} \mathrm{O}_{7} \mathrm{~N}_{2}: \mathrm{La}-(\mathrm{O}, \mathrm{N}) 2.298(6)-2.903(5) \AA{ }^{\circ} ;{ }^{[29]} \mathrm{Ce}_{10}\left[\mathrm{Si}_{10} \mathrm{O}_{9} \mathrm{~N}_{17}\right] \mathrm{Br}: \mathrm{Ce}-(\mathrm{O}, \mathrm{N}) 2.409(4)$
$-2.7465(7) \AA, \mathrm{Ce}-\mathrm{N} 2.447(7)-3.016(5) \AA \AA^{[26]} \operatorname{Pr}_{10}\left[\mathrm{Si}_{10-\mathrm{x}} \mathrm{Al}_{\mathrm{x}} \mathrm{O}_{9+\mathrm{x}} \mathrm{N}_{17-\mathrm{x}}\right] \mathrm{Cl}(\mathrm{x} \approx 1): \operatorname{Pr}-(\mathrm{O}, \mathrm{N})$ $2.390(4)-2.7465(7) \AA, \operatorname{Pr}-\mathrm{N} 2.439(7)-3.08(5) \AA$ ). ${ }^{[27]}$

The anisotropic displacement parameters of the $\operatorname{Ln}(1)$ site are larger by the factor of about 2-3 in comparison with the values of the $\operatorname{Ln}(2)$ site $(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}, \mathrm{Pr})$. Most probably, local deviations from the special site of $\operatorname{Ln}(1)$ causes the increase of the parameters. This effect was also observed for $\mathrm{Gd}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$. ${ }^{[2]}$
Furthermore, the displacement ellipsoid of $O(2)$ is elongated, which is presumably due to the deformation of the $\mathrm{OLn}_{6}$ octahedra $(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}, \mathrm{Pr})$, mentioned above. Therefore, a splitting of this site was assumed at lower temperatures analogously with the Gd containing compound. A crystal structure investigation for $\mathrm{Ce}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ at 130 K confirmed this assumption.

## ${ }^{29}$ Si solid-state NMR

The ${ }^{29} \mathrm{Si}$ solid-state NMR spectrum of $\mathrm{La}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ shows one broad signal peaking at 56.2 ppm with a full width at half maximum of 8 ppm (cf. Figure 5) This broadness of the signal is caused by the quadrupole momentum of the La atoms. The signal corresponds to $\mathrm{SiON}_{3}$ tetrahedra as evidenced by comparable shifts in La containing oxonitridoalumosilicate glasses ( -54.4 and -56.0 ppm ). For these compounds also the shifts for $\mathrm{SiN}_{4}(-43.3$ and $46.9 \mathrm{ppm})$ and $\mathrm{SiN}_{2} \mathrm{O}_{2}(-64.0,-62.5$ and $-60.9 \mathrm{ppm})$ were reported, which are not in the same range. ${ }^{[30]}$ Therefore, the NMR results corroborate that $\mathrm{La}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ exhibits exclusively $\mathrm{SiON}_{3}$ tetrahedra as evidenced by lattice energy calculations (MAPLE).

## Magnetical behavior

In the insert of Figure 6 the isothermal magnetization (at 2 and 300 K ) of $\mathrm{Ce}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ is depicted. Below the Curie-temperature $\mathrm{T}_{\mathrm{C}} \approx 7 \mathrm{~K}$, a ferromagnetic ordering was observed. The saturation magnetization converges to a value of $3 \mu_{\mathrm{B}}$ per formula unit, which is in accordance with three $\mathrm{Ce}^{3+}$ ions per formula unit.
The temperature dependence of the reciprocal magnetic susceptibility is shown in Figure 6. The bending of the reciprocal susceptibility curve is most probably caused by an amorphous by-product, which could not be determined by powder diffraction. Due to crystal field effects a certain deviation from Curie-Weiss behavior is observed. Therefore, the data between 200 and 300 K were fitted with an extended Curie-Weiss equation $\chi_{\mathrm{m}}=\chi_{0}+\mathrm{C} /(\mathrm{T}-\theta)$, which results in an effective magnetic moment of $2.2 \mu_{\mathrm{B}}$ per Ce . The temperature independent part of the susceptibility $\chi_{0}$ is comparatively large as the crystal field effects are neglected and has
no physical meaning. The magnetic moment is slightly below the theoretical value for free $\mathrm{Ce}^{3+}\left(2.54 \mu_{\mathrm{B}}\right)$. However, most of the experimentally determined values for $\mathrm{Ce}^{3+}$ containing compounds have been reported between 2.3 and $2.5 \mu_{\mathrm{B}}{ }^{[31]}$ Therefore, the measurements corroborate occurrence of solely $\mathrm{Ce}^{3+}$ in $\mathrm{Ce}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$.

## Comparison of $\mathrm{Ln}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ with other $\mathrm{A}_{3} \mathrm{BX}_{5}$ phases

The a/c ratio of the compounds $\mathrm{Ln}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}, \mathrm{Pr}, \mathrm{Gd})$ decreases between La and Gd according to the ionic radii. For the Ce and Pr containing compounds the values are identical within the margin of error, as the two ions have almost identical radii. However, this trend could not be observed for the lattice parameters of the compounds. Therefore, the correlation between the ratio a/c and the ionic radii is probably caused by an anisotropic framework folding mechanism, which modulates the cavities. The same trend was observed for the deviation of the $\mathrm{OLn}_{6}$ units from regular octahedral symmetry ( $\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}, \mathrm{Pr}, \mathrm{Gd}$ ) (cf. Table 3). A correlation between the $a / c$ ratio and the distortion of the octahedra could also be observed for isotypic compounds $\mathrm{Cs}_{3}\left[\mathrm{CoCl}_{4}\right] \mathrm{Cl}{ }^{[5]}$ and $\mathrm{Ba}_{3}\left[\mathrm{SiO}_{4}\right] \mathrm{O} .^{[3,4]}$ However, the tilting angle $\xi$ is independent from the $a / c$ ratio of the compounds. For $\mathrm{Ln}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}$, $\mathrm{Pr}, \mathrm{Gd})$ the values are in the same range, but there is no trend observable according to the ionic radii.

## Conclusions

In the last years, (oxo)nitridosilicates have received remarkable attention in materials science, as these compounds are both thermally and chemically very stable. With the syntheses of $\mathrm{Ln}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}, \mathrm{Pr})$, oxonitridosilicates with non-condensed tetrahedra could be obtained, although this group of compounds has a stronger tendency to form condensed networks than classical oxosilicates. Due to the non-condensed silicate substructure, the compounds are air- and water sensitive.
Although it was possible to investigate the distribution $\mathrm{O} / \mathrm{N}$ for the La containing compound by means of solid-state NMR measurements, it would be interesting to precisely determine the distribution using neutron diffraction. However, the available single crystals are yet too small for single-crystal neutron diffraction. Moreover powder data are most probably not conclusive enough, as one crystallographic light atom position has to be occupied statistically with both nitrogen and oxygen atoms and there was no indication for a decrease in symmetry.

## Experimental part

## Syntheses

Synthesis of silicon diimide " $\mathrm{Si}(\mathrm{NH})_{2}$ "
Instead of the relatively unreactive $\mathrm{Si}_{3} \mathrm{~N}_{4}$ silicon diimide was used as starting material. Previous syntheses of several (oxo)nitridosilicates and oxonitridoalumosilicates showed, that " $\mathrm{Si}(\mathrm{NH})_{2}$ " is well suited as starting material. ${ }^{[16,17,32]}$ According to equation 1 and 2, the X-ray amorphous and relatively undefined compound was obtained by ammonolysis of $\mathrm{SiCl}_{4}$ ( $>98 \%$, Merck, Darmstadt), followed by a thermal treatment at $600^{\circ} \mathrm{C}$ under an atmosphere of pure $\mathrm{NH}_{3}$.

$$
\begin{align*}
& \mathrm{SiCl}_{4}+6 \mathrm{NH}_{3} \xrightarrow[\mathrm{NH}_{3} \text { atmosphere }]{-78^{\circ} \mathrm{C}} \quad " \mathrm{Si}(\mathrm{NH})_{2} "+4 \mathrm{NH}_{4} \mathrm{Cl}  \tag{6}\\
& \xrightarrow[\mathrm{NH}_{3} \text { atmosphere }]{60{ }^{\circ} \mathrm{C} ; \mathrm{NH}_{4} \mathrm{Cl}} \quad " \mathrm{Si}(\mathrm{NH})_{2} " \tag{7}
\end{align*}
$$

## Synthesis of $\mathrm{La}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$

For the synthesis of $\mathrm{La}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ La ( $231.5 \mathrm{mg}, 1.67 \mathrm{mmol}, 99.9 \%$, Smart Elements) as small pieces, silicon diimide ( $53.8 \mathrm{mg}, 0.25 \mathrm{mmol}$ ) and $\mathrm{La}_{2} \mathrm{O}_{3}(217.2 \mathrm{mg}, 0.67 \mathrm{mmol}$, 99.999 \%, Chempur) were mixed in an agate motar and filled in a tungsten crucible under argon atmosphere in a glove box (Unilab, MBraun, $\mathrm{O}_{2} \leq 1 \mathrm{ppm}, \mathrm{H}_{2} \mathrm{O} \leq 1 \mathrm{ppm}$ ). Under purified $\mathrm{N}_{2}$, the crucible was heated to $1600{ }^{\circ} \mathrm{C}$ within 1 h in the reactor of a radio-frequency furnace ${ }^{[32]}$ and this maximum temperature was kept for 10 h . Subsequently, the crucible was cooled down to $900{ }^{\circ} \mathrm{C}$ with a rate of about $16^{\circ} \mathrm{C} \mathrm{h}^{-1}$ before quenching to room temperature by switching off the furnace. The sample contains exclusively $\mathrm{La}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ as green-yellow crystals.

## Synthesis of $\mathrm{Ce}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$

$\mathrm{Ce}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ was synthesized starting from $\mathrm{Ce}(280.2 \mathrm{mg}, 2.0 \mathrm{mmol}, 99.9 \%$, Smart Elements) as small pieces, $\mathrm{CeO}_{2}(172.1 \mathrm{mg}, 1.0 \mathrm{mmol}, 99.99 \%$, Chempur) and silicon diimide $(53.8 \mathrm{mg}, 0.25 \mathrm{mmol})$. After thoroughly mixing the starting materials in an agate mortar, the mixture was filled in a tungsten crucible under argon atmosphere in a glove box (Unilab, MBraun, $\mathrm{O}_{2} \leq 1 \mathrm{ppm}, \mathrm{H}_{2} \mathrm{O} \leq 1 \mathrm{ppm}$ ). The crucible was heated in a radio-frequency furnace analogously with the synthesis of the La containing compound. $\mathrm{Ce}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ was obtained without any by-products as dark-red crystals.

## Synthesis of $\mathrm{Pr}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$

$\mathrm{Pr}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ was obtained with varying degrees of purity by diverse synthetical routes using different starting materials. For example, $\mathrm{Pr}_{4} \mathrm{Si}_{2} \mathrm{O}_{7} \mathrm{~N}_{2}{ }^{[33]}$ or $\operatorname{PrN}{ }^{[34]}$ were observed as byproducts. For all syntheses, the starting materials were thoroughly mixed in an agate mortar in a glove box (Unilab, MBraun, $\mathrm{O}_{2} \leq 1 \mathrm{ppm}, \mathrm{H}_{2} \mathrm{O} \leq 1 \mathrm{ppm}$ ) and subsequently heated in a tungsten crucible in a radio-frequency furnace under purified $\mathrm{N}_{2}$.
One possibility to synthesize the compound starts from metallic $\operatorname{Pr}(178.8 \mathrm{mg}, 1.27 \mathrm{mmol}$, $99.9 \%$, Chempur, Karlsruhe) as small pieces, silicon diimide ( $37.0 \mathrm{mg}, 0.17 \mathrm{mmol}$ ) and $\operatorname{Pr}_{6} \mathrm{O}_{11}$ ( $110.1 \mathrm{mg}, 1.1 \mathrm{mmol},>99.9 \%$, Auer Remy, Hamburg). The mixture was heated to $950{ }^{\circ} \mathrm{C}$ within 5 min and kept for 25 min . Afterwards the temperature was increased to $1450{ }^{\circ} \mathrm{C}$ with a rate of about $42^{\circ} \mathrm{C} \mathrm{h}^{-1}$ and kept at that temperature for 8 h . Subsequently, the crucible was cooled down to $750{ }^{\circ} \mathrm{C}$ with a rate of about $140^{\circ} \mathrm{C} \mathrm{h}^{-1}$ before switching off the furnace.

For an alternative synthetical route $\operatorname{Pr}(211.3 \mathrm{mg}, 1.5 \mathrm{mmol}, 99.9 \%$, Chempur, Karlsruhe) as small pieces and $\mathrm{SiO}_{2}$ ( $30.4 \mathrm{mg}, 0.5 \mathrm{mmol}, \geq 99.8 \%$, Degussa, Frankfurt/Main) were used as starting materials. The heating program is similar to the program for the synthesis of $\mathrm{La}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$, but as maximum temperature $1550{ }^{\circ} \mathrm{C}$ was chosen.
$\mathrm{Pr}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ was also observed as product of the reaction between $\operatorname{Pr}(169.1 \mathrm{mg}, 1.2 \mathrm{mmol}$, $99.9 \%$, Chempur, Karlsruhe) and silicon diimide ( 53.8 mg ; 0.25 mmol ). Probably, the starting materials were contamined with oxygen in the latter case. The crucible was heated with the same temperature program as used for the syntheses of $\mathrm{Ln}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce})$.

## Chemical Analyses

Elemental analyses on selected crystals were performed for all compounds by energy dispersive X-ray spectroscopy (EDX) using a JSM-6500F scanning electron microscope (Jeol) with a Si/Li EDX detector (Oxford Instruments, model 7418). The measurements indicate a molar ratio $\mathrm{Ln} / \mathrm{Si}=3: 1(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}, \mathrm{Pr})$, which corroborates the formation of $\mathrm{Ln}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}, \mathrm{Pr})$. The determined compositions are in accordance with the structure model within the typical error ranges, as the result for $\operatorname{Pr}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ demonstrates (calc. (at\%): Pr 33, Si 11, N 33, O 22; found (at\%): Pr 30, Si 9, N 36 O 24).

## Single-crystal X-ray analysis

Single crystals of $\mathrm{Ln}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}, \mathrm{Pr})$ were isolated, enclosed in glass capillaries $(0.2 \mathrm{~mm})$, and sealed under argon atmosphere. Initially, the crystals were examined by Laue photographs recorded on a Bueger camera equipped with an image plate system. Singlecrystal X-ray data of all compounds were collected at room temperature on a STOE IPDS I diffractometer (Stoe and Cie GmbH; Darmstadt) with $\mathrm{Mo}-\mathrm{K}_{\alpha}$ radiation $(0.71073$ Å, graphite monochromator). The structures were solved by direct methods and refined after a numerical or semiempirical absorption correction of the diffraction data. ${ }^{[35,36]}$ The SHELX suite of programs was used for all calculations. ${ }^{[37,38]}$ Details concerning the data collection and refinements are summarized in Table 4. Further details of the crystal-structure investigations may be obtained from the Fachinformationszentrum Karlsruhe, D-6344 EggensteinLeopoldshafe, Germany, on quoting the depository numbers CSD-420626 ( $\mathrm{Ln}=\mathrm{La}$ ), 420627 $(\mathrm{Ln}=\mathrm{Ce})$, and $420628(\mathrm{Ln}=\mathrm{Pr})$. The atomic coordinates, the equivalent displacement parameters, and the site occupancy factors of the three compounds are listed in Table 5-7

## X-ray powder diffraction

X-ray powder diffraction data were collected on a STOE STADI P diffractometer with Mo$\mathrm{K}_{\alpha}$ radiation $(0.71073 \AA$ ) in Debye-Scherrer geometry. The samples were enclosed in silica tubes ( 0.2 mm ), and sealed under argon atmosphere. For pattern fitting (LeBail algorithm) and Rietveld refinement, the GSAS program package ${ }^{[39]}$ was used. The results of the singlecrystal X-ray analyses were used as starting models. The crystallographic data and details of the Rietveld refinements are given in Table 8. The isotropic displacement parameters were constrained to be equal for both the two Ln sites $(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}, \mathrm{Pr})$ and the N and O positions, respectively. Furthermore, the ( $\mathrm{N}, \mathrm{O}$ ) site occupancies were fixed. The analyses of the data unequivocally confirmed the purity of the compounds.

## Solid-state NMR

Solid-state NMR measurements have been performed on a Bruker Avance DSX 500 spectrometer with an external magnetic field of 11.75 T for $\mathrm{La}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$. The ${ }^{29} \mathrm{Si}$-1D-MAS spectra were recorded with direct excitation using a commercial triple resonance MAS probe, which was equipped with a rotor made of $\mathrm{ZrO}_{2}$ (diameter: 4 mm ). A repetition delay of 4096 s ( $>3 \mathrm{~T}_{1}$ ) and a rotation frequency of 9 kHZ were chosen. The $90^{\circ}$ impulse length was $2.5 \mu \mathrm{~s}$. The given chemical shift values refer to TMS as external chemical shift reference.

## Magnetic Measurements

Magnetic properties of $\mathrm{Ce}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ were measured utilizing a SQUID magnetometer (MPMS-XL Quantum Design Inc.). The sample was inserted in a capsule as fine ground powder which was fitted onto a straw of known diamagnetism. The magnetic susceptibility of the sample was collected between 2 and 300 K with magnetic flux densities up to 0.9 T . The obtained data were corrected for diamagnetic contributions of the capsule, the straw, and the sample using diamagnetic increments and analyzed with a modified Curie-Weiss law $\chi_{\mathrm{m}}=\chi_{0}+\mathrm{C} /(\mathrm{T}-\theta)$. From the Curie constant C , the effective magnetic moment per formula unit is calculated using the spin-only approximation.

Supporting Information: Anisotropic displacement parameters for the three compounds $\mathrm{Ln}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ with $\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}, \mathrm{Pr}$.

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Table 1 Results of the MAPLE calculations $[\mathrm{kJ} / \mathrm{mol}]$ for $\mathrm{Ln}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}, \mathrm{Pr})$ based on the $\mathrm{O} / \mathrm{N}$ distribution in $\mathrm{Gd}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O} . \Delta=$ difference.

|  | $\mathrm{Ln}^{3+}$ | $\mathrm{Si}^{4+}$ | $(\mathrm{N}, \mathrm{O})^{2.75-}$ | $\mathrm{O}^{2-}$ | Total <br> MAPLE | $\Delta$ |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  |  |  |  |
| La: | $3957 ; 4215$ | 10021 | 4029 | 1942 | 40466 | 0.9 |
| Ce: | $4010 ; 4275$ | 9924 | 4058 | 1943 | 40656 | 0.4 |
| Pr: | $4022 ; 4295$ | 10000 | 4078 | 1955 | 40878 | 0.5 |

Total MAPLE ( $3 \mathrm{LaN}+\mathrm{SiO}_{2}$ ): $40089 \mathrm{~kJ} / \mathrm{mol}$
Total MAPLE ( $3 \mathrm{CeN}+\mathrm{SiO}_{2}$ ): $40499 \mathrm{~kJ} / \mathrm{mol}$
Total MAPLE ( $3 \mathrm{PrN}+\mathrm{SiO}_{2}$ ): $40670 \mathrm{~kJ} / \mathrm{mol}$
Typical partial MAPLE values [kJ/mol]: $\mathrm{La}^{3+}: 3500-5000 ; \mathrm{Ce}^{3+}$ :
3800-4800; $\mathrm{Pr}^{3+}: 3800-4500 ; \mathrm{Si}^{4+}: 9000-10200 ; \mathrm{O}^{2-}: 2000-$
2800 ; (N,O) ${ }^{2.75-}: 4000-5200 .^{[23,24]}$

Table 2 Selected interatomic distances $(\AA)$ and angles $\left({ }^{\circ}\right)$ for $\operatorname{Ln}_{3}\left[\mathrm{SiN}_{3} \mathrm{O}\right] \mathrm{O}(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}, \mathrm{Pr})$. Standard deviations are given in parentheses.

|  |  | La | Ce | Pr |
| :--- | :--- | :--- | :--- | :--- |
| $\mathrm{Ln}(1)-\mathrm{O}(2)^{[0]}$ | $(2 \mathrm{x})$ | $2.7684(5)$ | $2.7671(5)$ | $2.7513(5)$ |
| $\operatorname{Ln}(1)-\mathrm{N} / \mathrm{O}(1)^{[1]}$ | $(8 \mathrm{x})$ | $2.858(4)$ | $2.818(3)$ | $2.8078(19)$ |
| $\mathrm{Ln}(2)-\mathrm{N} / \mathrm{O}(1)^{[1]}$ | $(2 \mathrm{x})$ | $2.450(6)$ | $2.411(5)$ | $2.402(3)$ |
| $\mathrm{Ln}(2)-\mathrm{O}(2)^{[0]}$ | $(2 \mathrm{x})$ | $2.5157(4)$ | $2.4804(4)$ | $2.4701(3)$ |
| $\mathrm{Ln}(2)-\mathrm{N} / \mathrm{O}(1)^{[1]}$ | $(4 \mathrm{x})$ | $2.734(5)$ | $2.712(4)$ | $2.699(3)$ |
| $\mathrm{Si}(1)-\mathrm{N} / \mathrm{O}(1)^{[1]}$ | $(4 \mathrm{x})$ | $1.728(6)$ | $1.736(5)$ | $1.724(3)$ |
| $\mathrm{N} / \mathrm{O}(1)^{[1]}-\mathrm{Si}(1)-\mathrm{N} / \mathrm{O}(1)^{[1]}$ | $101.5(4)$, | $101.1(3)$, | $101.1(2)$, |  |
|  |  | $113.6(2)$ | $113.79(17)$ | $113.83(13)$ |

Table 3 Comparison of the relevant data of isotypic compounds $\operatorname{Ln}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}, \mathrm{Ba}_{3}\left[\mathrm{SiO}_{4}\right] \mathrm{O}$, $\mathrm{Cs}_{3}\left[\mathrm{CoCl}_{4}\right] \mathrm{Cl}$ with perovskite.

| compound | $a / c$ ratio | angle $\xi /^{\circ}$ | deviation from the average X-M distance |
| :--- | :--- | :--- | :--- |
| $\mathrm{Gd}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ | $0.603(1)$ | $16.47(1)$ | $5.3 \%(\mathrm{X}=\mathrm{O}, \mathrm{M}=\mathrm{Gd})$ |
| $\mathrm{Pr}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ | $0.609(2)$ | $16.53(1)$ | $4.9 \%(\mathrm{X}=\mathrm{O}, \mathrm{M}=\mathrm{Pr})$ |
| $\mathrm{Ce}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ | $0.607(2)$ | $16.60(1)$ | $4.9 \%(\mathrm{X}=\mathrm{O}, \mathrm{M}=\mathrm{Ce})$ |
| $\mathrm{La}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ | $0.616(2)$ | $16.50(1)$ | $4.3 \%(\mathrm{X}=\mathrm{O}, \mathrm{M}=\mathrm{La})$ |
| $\mathrm{Cs}_{3}\left[\mathrm{CoCl}_{4}\right] \mathrm{Cl}$ | 0.635 | 19.34 | $2.4 \%(\mathrm{X}=\mathrm{Cl}, \mathrm{M}=\mathrm{Cs})$ |
| $\mathrm{Ba}_{3}\left[\mathrm{SiO}_{4}\right] \mathrm{O}$ | 0.651 | 14.96 | $2.2 \%(\mathrm{X}=\mathrm{O}, \mathrm{M}=\mathrm{Ba})$ |
| $\mathrm{Ca}\left[\mathrm{TiO}_{3}\right]$ | $2^{-1 / 2}$ | 0 | none $(\mathrm{X}=\mathrm{Ti}, \mathrm{M}=\mathrm{O})$ |

Table 4 Crystallographic data and details of the refinement procedures for $\mathrm{Ln}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}$, $\mathrm{Pr})$. Standard deviations are given in parentheses.

| Formula | $\mathrm{La}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ | $\mathrm{Ce}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ | $\mathrm{Pr}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ |
| :---: | :---: | :---: | :---: |
| Molar mass / g mol ${ }^{-1}$ | 518.85 | 522.48 | 524.85 |
| Crystal system |  | tetragonal |  |
| Space group |  | I4/mcm (no. 140) |  |
| Cell parameters / Å | $a=6.8224(10)$ | $a=6.7233$ (10) | $a=6.6979(9)$ |
|  | $c=11.074(2)$ | $c=11.069(2)$ | $c=11.005(2)$ |
| Cell volume / $\AA^{3}$ | 515.43(15) | 500.33(14) | 493.72(14) |
| Formula units / cell |  | $Z=4$ |  |
| X-ray density / $\mathrm{g} \mathrm{cm}^{-3}$ | 6.686 | 6.936 | 7.061 |
| Abs. coefficient / mm ${ }^{-1}$ | 24.550 | 26.969 | 29.275 |
| F(000) | 888 | 900 | 912 |
| Crystal size / mm ${ }^{3}$ | $0.07 \times 0.06 \times 0.04$ | $0.08 \times 0.05 \times 0.04$ | $0.06 \times 0.05 \times 0.04$ |
| Diffractometer |  | STOE IPDS |  |
| Radiation, monochromator |  | $\mathrm{K}_{\alpha}(\lambda=0.71073 \AA), \mathrm{g}$ |  |
| Temperature / K | 293(2) | 293(2) | 293(2) |
| $2 \theta$ range $/{ }^{\circ}$ | $4.22 \leq \theta \leq 28.47$ | $3.68 \leq \theta \leq 29.98$ | $3.70 \leq \theta \leq 30.22$ |
| Total no. of reflections | 2016 | 2409 | 2435 |
| Independent reflections | 195 | 218 | 220 |
| Observed reflections | 164 | 200 | 202 |
| Absorption correction | numerical | semiempirical (from equivalents) | numerical |
| Refined parameters | 18 | 18 | 18 |
| GOF | 1.050 | 1.152 | 1.077 |
| $R$-values [ $I>2 \sigma(I)]$ | $R 1=0.0182$ | $\mathrm{R} 1=0.0208$ | $\mathrm{R} 1=0.0131$ |
|  | $w \mathrm{R} 2=0.0363$ | $\mathrm{wR} 2=0.0430$ | $\mathrm{wR} 2=0.0276$ |
| all data | $R 1=0.0240$ | $\mathrm{R} 1=0.0234$ | $\mathrm{R} 1=0.0156$ |
|  | $w R 2=0.0370$ | $\mathrm{wR} 2=0.0435$ | $w R 2=0.0281$ |
| Max. / min. residual electron density / e $\AA^{-3}$ | -0.892; 0.962 | -1.311; 1.920 | -0.763; 0.771 |

Table 5 Atomic coordinates, equivalent displacement parameters, and site occupancy factors for $\mathrm{La}_{3}\left[\mathrm{SiN}_{3} \mathrm{O}\right] \mathrm{O} .(\mathrm{T}=293 \mathrm{~K})$. Standard deviations are given in parentheses.

| Atom | Wykoff <br> symbol | s.o.f. | $x$ | $y$ | $z$ | $U_{\text {eq }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  |  |  |  |
| $\mathrm{La}(1)$ | $4 a$ | 1 | 0 | 0 | $1 / 4$ | $0.0215(3)$ |
| $\mathrm{La}(2)$ | $8 h$ | 1 | $0.67593(5)$ | $0.17593(5)$ | 0 | $0.0119(2)$ |
| $\mathrm{Si}(1)$ | $4 b$ | 1 | 0 | $1 / 2$ | $1 / 4$ | $0.0092(7)$ |
| $\mathrm{N} / \mathrm{O}(1)$ | $16 l$ | $0.75 ; 0.25$ | $0.1387(6)$ | $0.6387(6)$ | $0.1512(6)$ | $0.0179(11)$ |
| $\mathrm{O}(2)$ | $4 c$ | 1 | 0 | 0 | 0 | $0.024(3)$ |

Table 6 Atomic coordinates, equivalent displacement parameters, and site occupancy factors for $\mathrm{Ce}_{3}\left[\mathrm{SiN}_{3} \mathrm{O}\right] \mathrm{O} .(\mathrm{T}=293 \mathrm{~K})$. Standard deviations are given in parentheses.

| Atom | Wykoff <br> symbol | s.o.f. | $x$ | $y$ | $z$ | $U_{\text {eq }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  |  |  |  |
| $\mathrm{Ce}(1)$ | $4 a$ | 1 | 0 | 0 | $1 / 4$ | $0.0143(3)$ |
| $\mathrm{Ce}(2)$ | $8 h$ | 1 | $0.67548(4)$ | $0.17548(4)$ | 0 | $0.0059(3)$ |
| $\mathrm{Si}(1)$ | $4 b$ | 1 | 0 | $1 / 2$ | $1 / 4$ | $0.0057(8)$ |
| $\mathrm{N} / \mathrm{O}(1)$ | $16 l$ | $0.75 ; 0.25$ | $0.1410(6)$ | $0.6410(6)$ | $0.1504(4)$ | $0.0105(11)$ |
| $\mathrm{O}(2)$ | $4 c$ | 1 | 0 | 0 | 0 | $0.021(3)$ |

Table 7 Atomic coordinates, equivalent displacement parameters, and site occupancy factors for $\mathrm{Pr}_{3}\left[\mathrm{SiN}_{3} \mathrm{O}\right] \mathrm{O}$. $(\mathrm{T}=293 \mathrm{~K})$. Standard deviations are given in parentheses.

| Atom | Wykoff <br> symbol | s.o.f. | $x$ | $y$ | $z$ | $U_{\text {eq }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $\operatorname{Pr}(1)$ | $4 a$ | 1 | 0 | 0 | $1 / 4$ | $0.01389(15)$ |
| $\operatorname{Pr}(2)$ | $8 h$ | 1 | $0.67581(3)$ | $0.17581(3)$ | 0 | $0.00488(13)$ |
| $\mathrm{Si}(1)$ | $4 b$ | 1 | 0 | $1 / 2$ | $1 / 4$ | $0.0039(4)$ |
| $\mathrm{N} / \mathrm{O}(1)$ | $16 l$ | $0.75 ; 0.25$ | $0.1405(4)$ | $0.6405(4)$ | $0.1504(3)$ | $0.0125(6)$ |
| $\mathrm{O}(2)$ | $4 c$ | 1 | 0 | 0 | 0 | $0.0233(16)$ |

Table 8 Crystallographic data of $\operatorname{Ln}_{3}\left[\mathrm{SiN}_{3} \mathrm{O}\right] \mathrm{O}$ derived from Rietveld refinement of X-ray data. Standard deviations are given in parentheses.

| Formula | $\mathrm{La}_{3}\left[\mathrm{SiN}_{3} \mathrm{O}\right] \mathrm{O}$ | $\mathrm{Ce}_{3}\left[\mathrm{SiN}_{3} \mathrm{O}\right] \mathrm{O}$ | $\mathrm{Pr}_{3}\left[\mathrm{SiN}_{3} \mathrm{O}\right] \mathrm{O}$ |
| :---: | :---: | :---: | :---: |
| Crystal system |  | tetragonal |  |
| Space group |  | I4/mcm (no. 140) |  |
| Lattice parameters / A | $a=6.82172(17)$ | $a=6.72040$ (10) | $a=6.68498$ (8) |
|  | $c=11.0981(3)$ | $c=11.0544(2)$ | $c=10.99356(17)$ |
| Cell volume / $\AA^{3}$ | 516.46(4) | 499.26(2) | 491.291(17) |
| Formula units / cell |  | $Z=4$ |  |
| Radiation |  | Mo-K $\mathrm{K}_{\alpha}(\lambda=0.70933 \AA)$ |  |
| Profile range $/{ }^{\circ}$ | $2 \leq 2 \theta \leq 50$ | $2 \leq 2 \theta \leq 60$ | $2 \leq 2 \theta \leq 60$ |
| No. of data points | 4800 | 5790 | 5500 |
| Observed reflections | 142 | 218 | 215 |
| No. of refined parameters | 61 | 60 | 61 |
| $R_{\mathrm{p}} / w R_{\mathrm{p}}$ | 0.0990 / 0.1325 | $0.0538 / 0.0716$ | $0.0466 / 0.0617$ |
| $R_{\text {\|F\| }} 2$ | 0.0753 | 0.0406 | 0.0458 |
| GOF | 0.97 | 1.36 | 1.16 |
| Reduced $\chi^{2}$ | 0.936 | 1.842 | 1.348 |



Fig. 1 Observed (crosses) and calculated (line) X-ray powder diffraction patterns as well as difference profile for the Rietveld refinement of $\operatorname{Ln}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ (left: $\mathrm{La}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$; middle: $\mathrm{Ce}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$; right: $\left.\mathrm{Pr}_{3} \mathrm{SiON}_{3}\right] \mathrm{O}$ ).
$P m \overline{3} m$

| Si: $\mathbf{1 b} \boldsymbol{b}$ <br> $m \overline{3} m$ | $\mathbf{O}: \mathbf{1} \boldsymbol{a}$ <br> $m \overline{3} m$ | Pr: $\mathbf{3 a}$ <br> $4 / m m \cdot m$ |
| :---: | :---: | :---: |
| $1 / 2$ | 0 | $1 / 2$ |
| $1 / 2$ | 0 | 0 |
| $1 / 2$ | 0 | 0 |
| $\qquad$ | $t 3$ |  |
|  |  |  |
|  |  |  |

$P 4 / \mathrm{mmm}$

| Si: $\mathbf{1 d}$ <br> $4 / \mathrm{mmm}$ | O: $\mathbf{a} \boldsymbol{a}$ <br> $4 / \mathrm{mmm}$ | Pr: $\mathbf{1 b}$ <br> 4/mmm | Pr: $\mathbf{2 f}$ <br> mmm. |
| :---: | :---: | :---: | :---: |
| $1 / 2$ | 0 | 0 | 0 |
| $1 / 2$ | 0 | 0 | $1 / 2$ |
| $1 / 2$ | 0 | $1 / 2$ | 0 |

k2
a-b, $\mathbf{a}+\mathrm{b}, 2 \mathrm{c}$
I4/mcm

| Si: $\mathbf{4 b}$ <br> $\overline{4} 2 m$ | O: $\mathbf{4 c}$ <br> $\mathbf{4} / m$ | Pr: $\mathbf{4 d}$ <br> $m \cdot m m$ | Pr: $\mathbf{8} \boldsymbol{h}$ <br> $m .2 m$. |
| :---: | :---: | :---: | :---: |
| 0 | 0 | 0 | 0.67581 |
| $1 / 2$ | 0 | 0 | 0.17581 |
| $1 / 4$ | 0 | $1 / 4$ | 0 |

Fig. 2 Crystallographic group-subgroup relation (Bärnighausen tree) ${ }^{[40]}$ between $\mathrm{Pr}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ and the perovskite structure type.


Fig. 3a: Structure of $\mathrm{Ln}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}, \mathrm{Pr})$ viewed along [010]. b: Structure viewed along [001]; the twisting angle $\xi$ is indicated. The $\mathrm{OLn}_{6}$ octahedra are depicted light gray, the $\mathrm{SiON}_{3}$ tetrahedra dark gray. The black spheres symbolize $\mathrm{Ln}^{3+}$ atoms.


Fig. 4 Coordination spheres of the $\mathrm{Ln}^{3+}$ ions (left: $\operatorname{Ln}(1)$; right: $\left.\operatorname{Ln}(2)\right)(\mathrm{Ln}=\mathrm{La}, \mathrm{Ce}, \mathrm{Pr}) .(\mathrm{Ln}$ : black; O: light gray; ( $\mathrm{N}, \mathrm{O}$ ): dark gray).


Fig. 5: ${ }^{29} \mathrm{Si}$ solid-state MAS NMR of $\mathrm{La}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$.


Fig. 6: Temperature dependence of the reciprocal susceptibility (black squares) and the susceptibility (gray squares) of $\mathrm{Ce}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$. Insert: Magnetization of $\mathrm{Ce}_{3}\left[\mathrm{SiON}_{3}\right] \mathrm{O}$ as a function of varying field at different temperatures.

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[^0]:    * Prof. Dr. W. Schnick

    Department Chemie und Biochemie
    Lehrstuhl für Anorganische Festkörperchemie
    Ludwig-Maximilians-Universität München
    Butenandtstraße 5-13(D)
    D-81377 München, Germany
    Fax: +49-(0)89-2180-77440
    E-mail: wolfgang.schnick@uni-muenchen.de

