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The Crystal Structures of Er_3OFS_3 and $Er_3OF_3S_2$: Two Erbium Oxide Fluoride Sulfides with Condensed Tetrahedral [ZEr₄] Units (Z = O and F)

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Dedicated to Professor Martin Jansen on the Occasion of his 65th Birthday

Abstract. The two new compounds Er₃OFS₃ and Er₃OF₃S₂ vary mostly in their fluoride and sulfide content. The fluoride-poor erbium(III) oxide fluoride sulfide Er₃OFS₃ crystallizes in the tetragonal space group *I4/mmm* (*a* = 374.86(2), *c* = 2068.41(9) pm, *Z* = 2) with two crystallographically different Er³⁺ cations. (Er1)³⁺ is surrounded by six S²⁻ anions in the shape of a slightly distorted octahedron, whereas (Er2)³⁺ exhibits a coordination number of eight ($4 \times O^2/F^-$ and $4 \times S^2^-$ arranged as a square antiprism). In the crystal structure of Er₃OFS₃ edge-sharing tetrahedral [ZEr₄]^{10.5+} units (Zⁿ⁻ = 50% O²⁻ + 50% F⁻) form condensed $\frac{2}{\infty}$ [OFEr₂]³⁺ layers. These layers alternate with $\frac{2}{\infty}$ [ErS₃]³⁻ sheets, built up of [ErS₆]⁹⁻ octahedra, which are connected via four common edges, to complete the three-dimensional crystal structure of Er₃OFS₃. The fluoride-rich erbium(III) oxide fluoride sulfide Er₃OF₃S₂ crystallizes in the orthorhombic space group *Ccce* (*a* = 538.31(2), *b* = 1892.03(9), *c* = 538.27(2) pm, *Z* = 4). Two crystallographically distinct Er³⁺ cations are present in its crystal structure. (Er1)³⁺ is surrounded by eight

Introduction

Oxide fluoride sulfides of the rare-earth metals demonstrate an interesting research topic, since they are closely related to rare-earth metal oxide sulfides, which are currently used in technical applications, for example as phosphors in X-ray imaging detectors [1-3]. Although, the mere oxide sulfides appear yet in many different formulae and structure types (e. g. M2O2S, M2OS2 or $M_{10}OS_{14}$, M = rare-earth metal [4–7]) substitution of oxygen by fluorine (e. g. MFS, M = La - Nd, Gd - Lu [8]) or plain insertion of fluoride anions into rare-earth metal oxide sulfides (e. g. $Ce_6O_2F_8S_3$ [9], Dy₃OF₅S [10]) may result in many new materials with special characteristics. Remarkably, in the crystal structures of almost all rare-earth metal compounds comprising light anions ($Z^{n-} = F^{-}, O^{2-}$ or N³⁻), condensed [ZM₄]^{m+} tetrahedra provide an important building block, which is responsible for several physical properties. For example, layers of edge-sharing [OM₄]¹⁰⁺ tetrahedra can cause magnetic frustration [11] or work as energy preserving hosts for Z^{n-} anions (Z = 25% O^{2-} + 75% F^-) in the shape of a twisted square antiprism, while (Er2)³⁺ holds a coordination number of nine (4 × O^{2-}/F^- and 5 × S^{2-} arranged as a monocapped square antiprism), copying the coordination geometry of the Er³⁺ cations in PbFCI-type ErFS. The threedimensional structure of Er₃OF₃S₂ is built up of S²⁻ sheets and $\frac{2}{\omega}$ [OF₃Er₃]⁴⁺ double layers, which are formed by strongly contorted tetrahedral [ZEr₄]^{10.75+} units (Zⁿ⁻ = 25% O²⁻ + 75% F⁻). The crystals of this compound show a prominent superstructure in the tetragonal space group *I4/mmm* with *a* = 380.59(3), c = 1892.03(9) pm and Z = 2. In the crystal structures of both title compounds, the light anions O²⁻ and F⁻ occupy one crystallographically unique *Wyckoff* position in a statistic allocation. Subject to the O/F content in the formulae, the ratio of O²⁻ : F⁻ is 1 : 1 in Er₃OFS₃ and 1 : 3 in Er₃OF₃S₂.

Keywords: Erbium; Oxides; Fluorides; Sulfides; Crystal Structures

luminescence [12, 13]. Even the nowadays so prominent high- T_{C^-} superconductors with the formula type $M(O_{1-x}F_x)FeAs$ [14–16] exhibit this kind of layers, although there, these { $M(O_{1-x}F_x)$ } layers serve only as spacers. However, not just the light-anion part of the crystal structure, but also the interaction of light and heavy anions, as well as the coordination geometry around the rare-earth cations account for property variations in light-anion derivatives of rare-earth metal sulfides. For instance optical characteristics of lanthanum oxide sulfides compared to those of lanthanum oxide fluoride sulfides vary significantly [17], which is not least due to the different surroundings for the La³⁺ cations within these compounds. To obtain a good understanding of structure-property relationships, a precise description of the atomic arrangement is essential. Thus, it is potentially feasible to gain a tool to "design" desired materials.

In this paper, the crystal structures of two new erbium(III) oxide fluoride sulfides are presented, to show the diversities of two, at the first glance, quite similar compounds. On the one hand the fluoride-poor but sulfur-rich Er_3OFS_3 and on the other hand the fluoride-rich but sulfur-poor $Er_3OF_3S_2$.

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Synthesis

Both title compounds accrued as unintentionally formed by-products, due to some oxygen contamination of the reaction mixtures. In attempts to synthesize ErFS, quite a few transparent, pinkish, flake-shaped crystals of Er₃OF₃S₂ developed in a reaction of elemental erbium (99.9 % ChemPur, Karlsruhe, Germany) and sulfur (99.999 % ChemPur, Karlsruhe, Germany) with erbium trifluoride (99.9 % ChemPur, Karlsruhe, Germany) in silicacovered, arc-welded tantalum capsules after five days at 850 °C. NaCl (99.9 % E. Merck, Darmstadt, Germany) as fluxing agent helped to grow crystals, which were large enough for single crystal X-ray diffraction. A pure sample of Er₃OF₃S₂ was obtained by annealing a mixture of erbium metal (99.9 % ChemPur, Karlsruhe, Germany), ErF₃ (99.9 % ChemPur, Karlsruhe, Germany), Er₂O₃ (99.9 % ChemPur, Karlsruhe, Germany) and elemental sulfur (99.999 % ChemPur, Karlsruhe, Germany) in a molar ratio of 4:3:1:6 for seven days at 850 °C in silica-jacketed, gastightly sealed tantalum ampoules. This procedure is not limited to erbium, as in that way phase-pure samples of $M_3OF_3S_2$ -type oxide fluoride sulfides with M = Y, Ho – Yb can be synthesized as well [18]. The other compound Er₃OFS₃ arose as transparent, orange, slat-shaped crystals from a product lump after a reaction of Er₂S₃ (made from elemental erbium metal and sulfur, according to [19]), ErF3 (99.9 % ChemPur, Karlsruhe, Germany), As2S3 (99.9 % E. Merck, Darmstadt, Germany) and CsI (99.9 % E. Merck, Darmstadt, Germany) as fluxing agent in evacuated silica ampoules at 700 °C within 7 days. Attempts to synthesize the pure material Er₃OFS₃ without As₂S₃ as fluxing agent and additional Er2O3 (99.9 % ChemPur, Karlsruhe, Germany) as oxygen source were not successful so far.

X-Ray Crystallography

Intensity data sets for single crystals of Er_3OFS_2 and $Er_3OF_3S_3$ were collected on a Nonius Kappa-CCD diffractometer using graphite-monochromatized Mo-K α radiation (wavelength: $\lambda = 71.07$ pm). Numerical absorption corrections were performed with the help of the program HABITUS [20]. The structure solutions and refinements were carried out by using the program package SHELX-97 [21]. Details of the data collections and the structure refinements [22] are summarized in Table 1, atomic positions and coefficients of the isotropic thermal displacement parameters [23] are given in Table 2, interatomic distances and selected bond angles are listed in Table 3. Further details of the crystal structure investigations are available from the Fachinformationszentrum (FIZ) Karlsruhe, D-76344 Eggenstein-Leopoldshafen, Germany (Fax: +497247-808-666; E-mail: crysdata@fizkarlsruhe.de), on quoting the depository numbers CSD-420552 for Er₃OFS₃, and CSD-420551 for Er₃OF₃S₂.

Electron-Beam Microprobe Analysis

The chemical compositions of Er₃OFS₂ and Er₃OF₃S₃ were approved by electron-beam microprobe analysis. An energy dispersive X-ray analysis (EDX) was performed on a Cameca SX 100 device to get a general composition for the crystals already used in the X-ray diffraction measurements. To affirm the presence of oxygen and fluorine and to obtain experimental data about the correct O^{2-} : F^- ratios a wavelength dispersive X-ray analysis (WDX) was carried out as well.

Results and Discussion

The crystal structures of the erbium(III) oxide fluoride sulfides Er_3OFS_3 and $Er_3OF_3S_2$ share some general features. On the one hand the light anions O^{2-} and F^- in both compounds occupy a common crystallographic site in a statistic way. Neither a crystallographic symmetry reduction, nor bond valence calculations

[24, 25] give hints of any ordering in the $\mathrm{O}^{2\text{-}}$ and $\mathrm{F}^{\text{-}}$ positions. Furthermore, in theory the empirical formulae of the title compounds Er_3OFS_3 and $Er_3OF_3S_2$ can be derived from $3 \times ErFS =$ Er₃F₃S₃ (PbFCl-type structure [26]) by substitution of two fluoride anions with one oxygen anion: $Er_3F_3S_3 - 2 F^- + O^{2-} = Er_3OFS_3$ or one sulfur anion with one oxygen anion: $Er_3F_3S_3 - S^{2-} + O^{2-} =$ Er₃OF₃S₂, respectively. This formal consideration can even be extended, regarding the crystal parameters. A-ErFS crystallizes in the tetragonal space group P4/nmm (Z = 2) with a = 374.61(3), c =681.34(5) pm [8], whereas the lattice constants of likewise tetragonal Er_3OFS_3 are a = 374.86(2), c = 2068.41(9) pm (Z = 2) in space group I4/mmm. $Er_3OF_3S_2$ shows a superstructure with a =380.59(3), c = 1892.03(9) pm (Z = 2) in space group I4/mmm, whichimplies a tripling in the [001] direction for both erbium(III) oxide fluoride sulfides as compared to A-type ErFS. But a closer look into the crystal structures of Er₃OFS₃ and Er₃OF₃S₂ proves that further similarities are only moderate.

Nevertheless, the relationship between ErFS and Er₃OFS₃ can not be denied. In the crystal structure of the sulfur-rich Er₃OFS₃ (space group *I*4/*mmm*: a = 374.86(2), c = 2068.41(9) pm, Z = 2) light-anion centred tetrahedral $[ZEr_4]^{10.5+}$ units (Z = 50% O²⁻ + 50% F⁻) (Fig. 3) form condensed $\frac{2}{\infty} [ZEr_{4/4}]^{1.5+} (\equiv \frac{2}{\infty} [OFEr_2]^{3+})$ layers by sharing four common edges (Fig. 1). These layers are strongly related to the $\frac{2}{\infty}$ [FEr_{4/4}]²⁺ sheets that form the PbFCl-type crystal structure of A-ErFS, together with S²⁻ anions. In Er₃OFS₃ the $\frac{2}{\infty}$ [OFEr₂]³⁺ building units spread out parallel to the (001) plane. In addition, they alternate along [001] with $_{\infty}^{2}$ [ErS_{4/4}S_{2/1}]³⁻ (= $\frac{2}{\infty}$ [ErS₃]³⁻) layers to complete the three-dimensional arrangement (Fig. 2). This second sort of layers is built up by octahedral $[ErS_6]^{9-1}$ units, sharing four common trans-oriented edges (Fig. 1). In the tetragonal crystal system an octahedral [ErS₆]⁹⁻ building block has more degrees of freedom than a polyhedron with ideal octahedral symmetry 4/m 3 2/m (O_h) (Fig. 3). This is reflected in slightly variable interatomic distances between cations and anions in the $[\text{ErS}_6]^{9-}$ units: $d((\text{Er1})^{3+}-(\text{S1})^{3+}) = 265.1 \text{ pm } (4\times) \text{ and } d((\text{Er1})^{3+} (S2)^{2-}$ = 265.4 pm (2×) (Table 3). The mean value of approximately 265 pm is in good agreement with interatomic distances found in other erbium(III) sulfides, e. g. D-type Er_2S_3 (d($\text{Er}^{3+}-\text{S}^{2-}$) = 261 – 283 pm for the Er^{3+} cations with CN = 6) [19]. As a result of the different two-dimensional building blocks, two crystallographically unique Er^{3+} cations need to be present: the above described $(Er1)^{3+}$ cations, with six adjacent S^{2-} anions, and $(Er2)^{3+}$ cations, which participate in the $\frac{2}{\infty}$ [OFEr₂]³⁺ layers and hold a coordination number of eight. The latter are surrounded by four S²⁻ anions and four light Z^{n-} anions $(Z^{n-}$ = 50% O^{2-} + 50% $F^{-})$ in the shape of a square antiprism (Fig. 3). Thus, the coordination sphere of the $(Er2)^{3+}$ cations resembles the surrounding area of the Er³⁺ cations in PbFCltype ErFS, although, compared to ErFS where the Er³⁺ cations show a coordination number of nine, the capping S^{2-} anion is missing in Er_3OFS_3 . The interatomic distances for Er^{3+} cations and Z^{n-} anions $(d((\text{Er2})^{3+}-\text{Z}^{n-}) = 231.7 \text{ pm}, 4\times)$ within these $[\text{ErZ}_4\text{S}_4]^{11-}$ polyhedra $(Z^{n-} = 50\% \text{ O}^{2-} + 50\% \text{ F}^{-})$ lie in-between the values found in erbium(III) fluoride sulfide $(d(\text{Er}^{3+}-\text{F}^{-}) = 244 \text{ pm in A-ErFS [8]})$ and erbium(III) oxide sulfide $(d(\text{Er}^{3+}-\text{O}^{2-}) = 225 - 229 \text{ pm in})$ Er₂O₂S [7]). This indicates that the light-anion site is simultaneously occupied by O²⁻ as well as F⁻ anions, which is also necessary to achieve electroneutrality in this compound. Therefore, the ratio of the mixed occupation was fixed at 50% O^{2-} : 50% F⁻, although, a free refinement of the occupancy factors of O²⁻ against F⁻ results in a ratio of 36% O^{2-} : 64% F⁻. Keeping in mind that the deviation is about 14% and a refinement of O²⁻ against F⁻ from X-ray data is not very reliable, the value for the refined ratio ranges within the

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tolerance. Furthermore, the ratio of one oxygen and one fluorine atom per formula unit is also confirmed by wavelength dispersive X-ray analysis (WDX), which results in the formula Er3.08O0.98F0.65S2.80. Therefore, the empirical formula Er3OFS3 is approved within the error of measurement. The coordination sphere for $(Er2)^{3+}$ is finally completed by four S²⁻ anions, which are 289.2 pm afar. This value is slightly larger than the space between erbium and sulfur for the sixfold coordinated (Er1)³⁺ cations, but still in the range of general interatomic $d(\text{Er}^{3+}-\text{S}^{2-})$ distances [19]. So far, this structure type, in which $\frac{2}{\infty}[ZM_{4/4}]$ and $\frac{2}{\infty}[MX_{4/4}X_{2/1}]$ sheets (M = metal cations; $Z = O^{2-}$, F^- ; $X = S^{2-}$, CI^-) pile up, is only known for compounds with two different cationic species, e. g. Bi_xCa_yO₂S₃, $Bi_xCd_yO_2Cl_3$ [27, 28] and $U_2ErO_2S_3$ [29]. Thus, Er_3OFS_3 is the first example for this structure type with only one kind of metal cations, but a combination of two sorts of light anions.

The second erbium(III) oxide fluoride sulfide $Er_3OF_3S_2$ crystallizes in the orthorhombic crystal system (space group Ccce) with a = 538.31(2), b = 1892.03(9) and c = 538.27(2) pm. As in the crystal structure of the aforementioned compound, edge-sharing light-anion centred $[ZEr_4]^{10.75+}$ units (Z = 25% O²⁻ + 75% F⁻) form condensed $\frac{2}{\infty}$ [ZEr_{4/4}]^{1.75+} layers. However, two of these twodimensional building blocks are further linked via more common edges to form ${}_{\infty}^{2}[\operatorname{ZEr}_{2/8}\operatorname{Er}_{2/4}]^{+} (\equiv {}_{\infty}^{2}[\operatorname{OF}_{3}\operatorname{Er}_{3}]^{4+})$ double layers (Fig. 4). Therefore, one sort of Er^{3+} cations, $(Er1)^{3+}$, is coordinated exclusively by eight of the light anions, whereas the other one, $(\text{Er2})^{3+}$, carries five S²⁻ as well as four Zⁿ⁻ anions (Z = 25% O²⁻ + 75% F⁻) in the shape of a monocapped distorted square antiprism, alike the coordination for Er^{3+} cations in PbFCl-type ErFS (Fig. 5). As the space group Ccce, a subgroup of I4/mmm (Scheme 1), indicates, the coordination polyhedra in $Er_3OF_3S_2$ have lower symmetries as the ones in Er_3OFS_3 . Thus, the tetrahedral $[ZEr_4]^{10.75+}$ units in $\text{Er}_3\text{OF}_3\text{S}_2$ with $d(\mathbb{Z}^{n-}-\mathbb{E}r^{3+}) = 228.9, 229.8, 230.3$ and 286.1 pm are better described as trigonal pyramids or even monocapped triangles (Fig. 5), which is a quite common coordination feature for fluoride anions. In addition, the nine interatomic distances (CN(Er2) = 7 + 2) in the capped antiprisms $[(\text{Er2})Z_4S_5]^{12-}$ range from $d(\text{Er}^{3+} Z^{n-}$ = 228.9 (2×), 286.1 (2×) to $d(Er^{3+}-S^{2-})$ = 278.9 (4×), 283.4 pm (1x), whereas in the $[(Er1)Z_8]^{7-}$ polyhedra the distances unite $(d(\text{Er}^{3+}-\text{Z}^{n-}) = 229.8 \text{ (4x)} \text{ and } 230.3 \text{ pm (4x)}, \text{ Table 3)}, \text{ but the}$ shape of the polyhedron is somewhere between a square antiprism and a square prism (Figs. 5 and 6) with a rotation angle of $\alpha_t = 20.2^\circ$ between the two square planes (ideal prism: $\alpha_t = 0^\circ$, ideal antiprism: $\alpha_t = 45^\circ$). Like in Er₃OFS₂ the light-anion site (Z) in Er₃OF₃S₂ (= $Er_3Z_4S_2)$ is occupied by $O^{2-}\mbox{ as well as }F^-\mbox{ anions.}$ To obtain electroneutrality in this compound one out of four light anions (Z) has to be oxygen. Therefore, the occupancy of the mixed O^{2-}/F^{-} site was fixed at $Z^{n-} = 25\% \text{ O}^{2-}$: 75% F⁻, which is in very good agreement with an O^{2-}/F^{-} ratio of 27% : 73% experimentally determined by wavelength dispersive X-ray analysis (WDX). Compounds with the same composition M3X4S2 are already known for M = Sm, Eu, Yb [18, 30], but in these cases the light-anion site (Z) is exclusively occupied by $F^{\scriptscriptstyle -}$ anions and electroneutrality is achieved by a mixed valence (M^{2+}/M^{3+}) of the metal cations. Thus, M^{3+} cations (M = Sm, Eu, Yb) represent the central atom within the $[MF_4S_5]^{11-}$ units, similar to $(Er2)^{3+}$ in $Er_3OF_3S_2$, whereas M^{2+} cations are surrounded by eight F- anions in the shape of an ideal square prism (Fig. 6), alike the coordination of Ca^{2+} in the fluoritetype structure of CaF₂ [31]. These mixed-valent rare-earth metal compounds crystallize in the tetragonal crystal system with space 60 group *I*4/*mmm* (*e. g. a* = 384.61(3), *c* = 1884.2(4) pm for M = Yb), in which the light anions (F) occupy the Wyckoff site 8g. For Er₃OF₃S₂, which bears strictly trivalent cations along with a mixture of two different anions (O²⁻/F⁻) on the light-anion site, a related superstructure in space group I4/mmm with a = 380.59(3), c =1892.03(9) pm is observed, but in this symmetry the light anions (O^{2-}/F^{-}) are assigned to the Wyckoff site 16n. Thus, the light-anion site has to be half-occupied only as well as mixed allocated. Former attempts by different research groups to properly refine the crystal structures of various M₃OF₃S₂ representatives in orthorhombic space groups without any face centering of the translation symmetry (e. g. $M_3OF_3S_2$, M = La, Ce in Pnmm [17]) or even without inversion symmetry (e. g. $M_3OF_3S_2$ with M = Y, Ho and Tm in P222 [18, 32]) remained unsatisfying. However, a symmetry reduction to space group Ccce, according to the Bärnighausen tree [33] seen in Scheme 1, provides a better description of the crystal structure, in which at least the underoccupation is resolved. It can be easily understood, why there has to be a loss in symmetry for Er^(III)₃OF₃S₂ in contrast to Yb^(III)₂Yb^(II)F₄S₂, by comparing the two different anionic surroundings for the eightfold coordinated (Er1)³⁺ and (Yb1)²⁺ cations, respectively (Fig. 6). In the square prisms $[Yb^{(II)}F_8]^{6-}$ the gap between F⁻ anions amounts to $d(F^- - F^-) = 272$ pm (4x) within the quadratic area and 311 pm (4x) within the rectangular faces. Substitution of the Yb²⁺ cations with the smaller Er^{3+} cations causes a shrinking of the thickness of the $\frac{2}{\infty} [\mathrm{OF}_3 \mathrm{Er}_3]^{4+}$ double layers, which reduces the $Z^{n-}...Z^{n-}$ distance considerably. Although, the spacing $d(Z^{n-}...Z^{n-})$ (Z = 25% O²⁻ : 75% F⁻) within the square plane of the [Er^(III)X₈]⁷⁻ polyhedra levels off (273 pm), the interspacing of Zⁿ⁻...Zⁿ⁻ in the rectangular faces would amount only to about 242 pm, which is far smaller than twice the anionic radius of $r(O^{2-}) = 138$ pm or $r(F^{-}) = 131$ pm [34, 35]. Hence, due to electrostatic repulsion the quadratic faces are twisted, and the shape of the $\left[\text{ErZ}_{8}\right]^{7-}$ polyhedra (Z = 25% $O^{2-}:$ 75% $F^{-})$ is no longer a square prism, but a hybrid between square prism and square antiprism, in which a distance $d(Z^{n-}...Z^{n-})$ of 259 pm is maintained. Although, this is still a little bit smaller than double the ionic radii, such short distances between two fluoride anions are also known for other compounds, e. g. ErF_3 with $d(F^- \cdots F^-) = 250$ pm [36] or $\text{ErF}[\text{MoO}_4]$ with $d(F^- \cdots F^-) = 255$ pm [37]. In the special case of edge-fused planar [FM₃]⁸⁺ triangles as found in B-type YFS and Ctype LuFS $d(F^{-}...F^{-})$ can even range between 230 and 235 pm [38]. Nevertheless, for $Er_3OF_3S_2$ the coordination geometry of the Er^{3+} cations can no longer be supported in the tetragonal space group I4/mmm, thus, the symmetry in $\text{Er}_3\text{OF}_3\text{S}_2$ has to be reduced. Since this is merely triggered by the light anions, the tetragonal superstructure is highly prominent and the real orthorhombic symmetry (Ccce) can be simply overlooked. Therefore, it is suspected that the tetragonal mixed-cationic representatives of the Yb₃F₄S₂-type structure, which contain alkaline-earth as well as rareearth metals on one crystallographic position in space group I4/mmm (e. g. $EAM_2F_4S_2$ with EA = Ca, Sr and M = Sm, Ce [39]) and in which the light anions (F⁻) also half-occupy the Wyckoff site 16n, are actually orthorhombic. Besides, it is assumed that the compounds described in [39] therefore contain some oxygen along with fluorine. Usually the cell determination on a single-crystal diffractometer results in the tetragonal supercell, which makes it difficult to refine the cell parameters from single-crystal data. Unfortunately, there are not enough crystallites of Er₃OF₃S₂ to properly determine the unit cell from X-ray powder diffraction data so far.

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Figure Captions and Scheme

- **Figure 1** View at the condensed $\frac{2}{\infty}$ [OFEr₂]³⁺ layers (*left*) and $\frac{2}{\infty}$ [ErS₃]³⁻ sheets (*right*), which spread out parallel to the (001) plane in the crystal structure of Er₃OFS₃. Dotted lines indicate the linking edges. The light-anion site is occupied by 50% O²⁻ + 50% F⁻.
- Figure 2 Part of the crystal structure of Er₃OFS₃, showing the stacking of the $\frac{2}{\omega}$ [OFEr₂]³⁺ and $\frac{2}{\omega}$ [ErS₃]³⁻ layers (O²⁻/F⁻ = 50% O²⁻ + 50% F⁻).
- Figure 3 Coordination polyhedra around the two crystallographically different Er^{3+} cations (*top*) and the diverse anions (*bottom*) in Er_3OFS_3 .
- Figure 4 Section of the crystal structure of $Er_3OF_3S_2$, showing the stacking of the $\frac{2}{\omega}[OF_3Er_3]^{4+}$ double layers $(O^2/F^- = 25\% O^{2-} + 75\% F^-)$.
- Figure 5 Coordination polyhedra around the two crystallographically different Er^{3+} cations (*top*) and the S²⁻ and Zⁿ⁻ anions (Zⁿ⁻ = 25% O²⁻ + 75% F⁻) (*bottom*) in Er₃OF₃S₂.
- **Figure 6** Light-anion surrounding of the $(\text{Er1})^{3+}$ cations, as viewed onto the quadratic plane (*left*) and onto the rectangular faces (*mid*) of the polyhedron, along with a comparison of the two different anionic surroundings for the eightfold coordinated $(\text{Er1})^{3+}$ cations in $\text{Er}_3\text{OF}_3\text{S}_2$ (*mid*) and $(\text{Yb1})^{2+}$ cations in $\text{Yb}_3\text{F}_4\text{S}_2$ (*right*).

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3 ⊿	Table 1. Crystallogra	phic data for Er
4 5		
6	Crystal system	
7	Formula units	
8	Space group	
9	Lattice parameters,	a in pm
10		b in pm
11		c in pm
12	Calculated density, D	x in g·cm ⁻³
13	Molar volume, V_m in	cm ³ ·mol ⁻¹
14	F(000)	
15	Index range, $\pm h / \pm k$	±l
16	Theta range, θ_{max}	
17	Absorption coefficien	nt, μ in mm ⁻¹
18	Data corrections	
19		
20		
21	Reflections, collected	l / unique
22	R_{int} / R_{σ}	
23	Structure solution and	1 refinement
24		
25		
26	R1 for n reflections w	$ith F_o \geq 4\sigma(F_o)$
27	R_1 / wR_2 for all reflect	ctions
28	Goodness of Fit (Goo	oF)
29	Extinction, g	
30	Residual electron der	isity,
31	ρ in e ⁻ ·10 ⁻⁶ pm ⁻³ , min	. / max.
32		
33		
34	Table 2. Atomic coo	rdinates and equ
35	$U_{eq}^{a)}$ for Er_3OFS_3 and	$Er_3OF_3S_2$.
36	FraOFSa	x/a
37	E1. (2.)	
38	Er1 (2a)	0
	1 (() a)	~ ~ ~ ~ ~ ~ ~ ~ ~ ~ ~ ~ ~ ~ ~ ~ ~ ~ ~ ~

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ta for Er₃OFS₃ and Er₃OF₃S₂.

Er₃OFS₃

tetragonal

Z = 2

= *a*

7.232

87.516

4 / 4 / 26

1725 / 139

0.097 / 0.028

0.029 (n = 130)

0.031 / 0.057

1.155

0.0023(3)

-1.78 / 2.29

538

27.95

43.89

I4/mmm

374.86(2)

2068.41(9)

Er₃OF₃S₂

Z = 4

Ccce

538.31(2)

1892.03(9)

538.27(2)

7.741

82.536

1084

23.20

46.22

background, polarization, and Lorentz factors; numerical absorption correction with the program HABITUS [20]

program package SHELX-97 [21], scattering factors according to International Tables, Vol. C [22]

6/24/7

4624 / 333

0.042 / 0.019

0.031 (n = 248)

0.041 / 0.096

-2.36 / 2.21

1.157 0.0075(6)

orthorhombic

Table 3. Selected interatomic distances (d/pm, esd = ± 0.1) and bond angles (\triangleleft/deg , esd = ± 0.1) for Er₃OFS₃ (*top*) and Er₃OF₃S₂ (*bottom*).

Er ₃ OFS ₃					
distances			angles		
Er1 - S1	4×	265.1	S1-Er1-S1	4×	90
Er1 - S2	$2 \times$	265.4	S2-Er1-S1	8×	90
			S2-Er2-S2	4×	80.8
Er2 – O/F	4×	231.7	S2 - Er2 - O/F	8×	73.2
Er2 - S2	4×	289.2	O/F - Er2 - O/F	4×	69.8
			Er2 - O/F - Er2	4×	110.2
			Er2 - O/F - Er2	2×	108.0
$Er_3OF_3S_2$					
distances			angles		
Er1 – O/F	4×	229.8	O/F - Er1 - O/F	4×	68.1
Er1 – O'/F'	4×	230.3	O/F - Er1 - O/F	4×	92.7
			S-Er2-S	4×	86.1
Er2 – O/F	2×	228.9	S-Er2-S'	8×	74.9
Er2 – O'/F'	2×	286.1	O/F - Er2 - O/F	1×	85.0
Er2 - S	2×	278.9	$O^\prime\!/F^\prime - Er2 - O^\prime\!/F^\prime$	1×	105.4
Er2 - S'	2×	278.9	$O/F-Er2-O^{\prime}/F^{\prime}$	4×	63.5
Er2 - S''	1×	283.4	Er1 - O/F - Er1'	1×	111.6
			Er1 - O/F - Er2	1×	121.0
			Er1' - O/F - Er2	1×	120.8
			Er2 - O/F - Er2'	$1 \times$	94.7
			Er1 - O/F - Er2'	$1 \times$	100.8
			Er1' - O/F - Er2'	1×	100.7

Table 4. Motifs of mutual adjunction [40-42] for Er₃OFS₃ (top) and Er₃OF₃S₂ (bottom).

and equivalent isotropic thermal displacement parameters S₂.

Er ₃ OFS ₃	x / a	y / b	z / c	U_{eq} in pm ²
Er1 (2 <i>a</i>)	0	0	0	178(5)
Er2 (4 <i>e</i>)	0	0	0.31582(5)	194(4)
$O/F^{b)}$ (4d)	0	¹ / ₂	¹ / ₄	109(25)
S1 (2b)	0	0	¹ / ₂	253(19)
S2 (4 <i>e</i>)	0	0	0.1284(3)	143(10)
$Er_3OF_3S_2$	x / a	y / b	z / c	U_{eq} in pm^2
Er1 (4 <i>a</i>)	0	¹ / ₄	¹ / ₄	198(4)
Er2 (8f)	0	0.59419(4)	¹ / ₄	101(4)
O/F ^{c)} (16 <i>i</i>)	0.1996(9)	0.6848(3)	0.0512(9)	284(14)
S (8f)	0	0.4444(2)	¹ / ₄	55(6)

 $^{a)}$ U_{eq} is defined as one third of the trace of the orthogonalized U_{ij} tensor [20];

^{b)} fixed mixed occupation: O = 50%; F = 50%;

^{c)} fixed mixed occupation: O = 25%; F = 75%.

Er ₃ OFS ₃	O/F	S1	S2	CN
Er1	0 / 0	4/4	2 / 1	6
Er2	4/4	0/0	4/4	8
CN	4	4	5	
Er ₃ OF ₃ S ₂	O/F	S		CN
Er1	8/2	0/0		8
Er2	2+2 / 1+1	5/5		7+2
CN	3+1	5		

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View at the condensed [OFEr2]3+ layers (left) and [ErS3]3- sheets (right), which spread out parallel to the (001) plane in the crystal structure of Er3OFS3. Dotted lines indicate the linking edges. The light-anion site is occupied by 50% O2- + 50% F-. 170x79mm (600 x 600 DPI)



Part of the crystal structure of Er3OFS3, showing the stacking of the [OFEr2]3+ and [ErS3]3 layers (O2–/F– = 50% O2– + 50% F–). 170x219mm (600 x 600 DPI)



ZAAC

Coordination polyhedra around the two crystallographically different Er3+ cations (top) and the diverse anions (bottom) in Er3OFS3. 170x168mm (600 x 600 DPI)





Section of the crystal structure of Er3OF3S2, showing the stacking of the [OF3Er3]4+ double layers (O2-/F- = 25% O2- + 75% F-). 170x158mm (600 x 600 DPI)





Coordination polyhedra around the two crystallographically different Er3+ cations (top) and the S2and Zn- anions (Zn- = 25% O2- + 75% F-) (bottom) in Er3OF3S2. 170x133mm (600 x 600 DPI)



Light-anion surrounding of the (Er1)3+ cations, as viewed onto the quadratic plane (left) and onto the rectangular faces (mid) of the polyhedron, along with a comparison of the two different anionic surroundings for the eightfold coordinated (Er1)3+ cations in Er3OF3S2 (mid) and (Yb1)2+ cations in Yb3F4S2 (right). 170x54mm (600 x 600 DPI)





Bärnighausen symmetry tree, displaying the group-subgroup relationship between the tetragonal supercell and the real orthorhombic unit cell of Er3OF3S2. 170x211mm (600 x 600 DPI)