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Large increase of the Curie temperature by orbital ordering control

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Using first principle calculations we showed that the Curie temperature of manganites thin films can be increased by far more than an order of magnitude by applying appropriate strains. Our main breakthrough is that the control of the orbital ordering responsible for the spectacular T_C increase cannot be imposed by the substrate only. Indeed, the strains, first applied by the substrate, need to be maintained over the growth direction by the alternation of the manganite layers with another appropriate material. Following these theoretical findings, we synthesized such super-lattices and verified our theoretical predictions.

Ferromagnetic super-lattices with very high T_C present obvious interests for electronic applications. This is the first step required to design and control magnetism in more elaborate structures. Perovskite manganese oxides of generic formula $A_{1-x}B_xMnO_3$ (A being a trivalent cation and B a divalent one) are metallic and ferromagnetic in the approximative range of doping values $0.17 < x < 0.5$ [5]. These properties are directly related to the $3d$ shell of the manganese atoms and can be easily explained by the double-exchange mechanism. The Curie temperature is thus directly related to the delocalization within the e_g orbitals of the Mn^{3+} / Mn^{4+} ions [2], that is to the manganese ions orbital order. Finding a way to design such materials with the desired $3d$ orbital order would thus open the way to the design of artificial materials with the desired magnetic properties, and thus to novel magneto-electronic applications, including spin-valve devices or non-volatile magnetic memory working far above room temperature.

The manganese e_g shell is doubly degenerated (in a regular octahedron environment) however only partly occupied. The Mn^{3+} ions are thus Jahn-Teller active and small distortions of the oxygen octahedra can stabilize one of the e_g orbitals with respect to the other. If one can impose a specific geometry around the Mn ions (for instance using synthesis in thin films), it should thus be possible to choose between the $d_{3z^2-r^2}$ and the $d_{x^2-y^2}$ orbitals which one will be occupied and which one will be empty. If the $d_{3z^2-r^2}$ is occupied, the double exchange will take place essentially along the \vec{c} direction, that is between (\vec{a}, \vec{b}) planes (see figure 1). On the contrary if the $d_{x^2-y^2}$ is stabilized, the intra-layer double exchange will be very strong and the inter-layer weak. In an essentially 2-dimensional (2-D) system such as thin films, only the in-plane interactions are important and the control of the orbital ordering should thus allow the control of magnetic exchange and the associated Curie temperature, T_C . According to the above argument, the Curie temperature of 2-D thin films will be maximized if one

(i) stabilizes the $d_{x^2-y^2}$ orbital over the $d_{3z^2-r^2}$ and (ii) maximizes the in-plane effective exchange integral, between the $d_{x^2-y^2}$ orbitals, by preventing the tilt of the octahedra observed in the bulk [6]. Indeed, in the double exchange mechanism, T_c scales as

$$T_c \sim J \sim t_{dp}^2 \sim S_{dp}^2 \quad (1)$$

(J : effective exchange between adjacent Mn ions, t_{dp} : (Mn) d -(O) $2p$ transfer integral and S_{dp} : overlap) and the octahedra tilt lowers the S_{dp} overlap between the Mn $d_{x^2-y^2}$ and the O $2p_\lambda$ bridging orbitals.

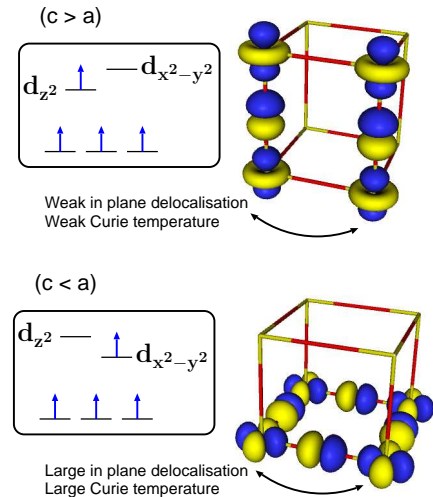


FIG. 1: (Color online) Schematic representation of orbital ordering induced by cell distortions in a two dimensional system. ($c > a$) The elongation in the \vec{c} direction favors the $d_{3z^2-r^2}$ orbital occupation and weak in-plane ferromagnetic coupling, the delocalisation being essentially in the \vec{c} direction. ($c < a$) The contraction of the c parameter favors the $d_{x^2-y^2}$ orbital occupation, strong in-plane ferromagnetic coupling and large Curie temperature.

$A_{1-x}B_xMnO_3$ thin films deposited on a cubic substrate presenting $a = b$ parameters slightly larger than the bulk manganite value should thus insure an increase of the manganite in-plane parameters and, in order to approximately conserve the cell volume, a decrease of the c parameter. With $c/a < 1$ the $d_{x^2-y^2}$ orbital occupation

*Deceased february 2009

and large T_c should be favored, as desired.

Several authors synthesized such thin films as for instance $\text{La}_{2/3}\text{Sr}_{1/3}\text{MnO}_3$ (LSMO) thin films deposited on the SrTiO_3 cubic substrate [7]. They found that for a small number of LSMO unit cells in the \vec{c} direction, the magnetization as well as the Curie Temperature are strongly decreased compared to the 370K bulk value [8]. From twelve deposited unit cells however the bulk behavior is approximately recovered with just a slight T_c reduction. It is clear that the expected behavior and thus orbital ordering is not realized in these films. In fact the quick recovering of the bulk magnetic behavior tells us that the strains imposed by the substrate are essentially very quickly relaxed as a function of the film thickness. This is most probably the case for the octahedra tilt present in the LSMO bulk. One should thus find a way to maintain the desired strains (in-place cell parameter elongation and quadratic symmetry) over the whole film. For this purpose one can think to alternate LSMO layers with layers of an adequate cubic material as in $[(\text{La}_{2/3}\text{Sr}_{1/3}\text{MnO}_3)_n(\text{BaTiO}_3)_p]_q$ super-lattices. We thus explored this possibility using first principle calculations.

We studied, using periodic density functional theory [4], the $(\text{La}_{2/3}\text{Sr}_{1/3}\text{MnO}_3)_3(\text{BaTiO}_3)_3$ super-lattice deposited on SrTiO_3 (001) substrate (see figure 2). The effect of the substrate was modeled by imposing the in-plane lattice constants to $a = b = 3.0905 \text{ \AA}$. The c parameter was optimized as well as atomic positions.

The super-lattice is built from metallic LSMO layers and insulating BaTiO_3 (BTO) layers. It is thus of crucial importance for the reliability of the results to be able to accurately position the LSMO Fermi level referring to the BTO gap. For this purpose we used hybrid functionals known to correctly reproduce insulating gaps, that is B3LYP [13] and the very recent B1WC [14] specifically designed to obtain the correct properties on ferroelectric materials such as BaTiO_3 . We thus used the CRYSTAL package [12] with the basis sets and effective core pseudo-potentials (ECP) of reference 15. Another important technical point is the treatment of the La/Sr disorder. Indeed, in periodic calculations it is not possible to properly mimic the ionic disorder. We thus made a whole set of calculations associated with the two extreme cases, that is with the Sr ions totally ordered (3 different positions) and with average La/Sr ions. These average ions were modeled either using ordered La/Sr ECPs but with averaged effective nuclear charges (3 possibilities) or by using only La ECPs with average effective charges. All 7 different approximations of the ionic disorder were computed with the two functionals. All of them yielded equivalent results and physical properties, even if one finds slight numerical differences. One set of numerical values (corresponding to B3LYP functional, ordered ECPs with the strontium in the middle layer and

averaged charges) are presented as a matter of example.

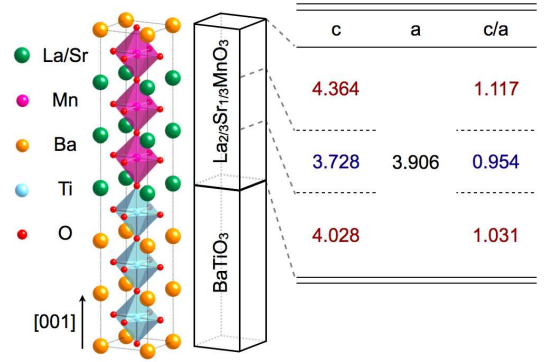


FIG. 2: (Color Online) Schematic representation of the super-lattice and calculated lattice parameters (in \AA) for the LSMO layer.

Considering the fact that the LSMO layers are extended in the (\vec{a}, \vec{b}) directions compared to the bulk ($a = b = c = 3.87 \text{ \AA}$), our results (figure 2) are at first quite surprising. Indeed, the two LSMO layers at the interface with the BaTiO_3 are not only extended in the \vec{a} and \vec{b} directions, but also in the \vec{c} direction. It results a strong extension of the unit cell volume with respectively 66.57 \AA^3 — for the interface with a Ba-O layer — and 61.44 \AA^3 — for the interface with a Ti-O layer — to be compared with the bulk pseudo-cubic unit cell volume of 58.3 \AA^3 [6]. On the contrary, the central LSMO layer is strongly contracted in the \vec{c} direction resulting in a reduction of the unit cell volume (56.87 \AA^3). One sees immediately that for the interface layers the $c/a > 1$ anisotropy strongly favors the stabilization of the $d_{3z^2-r^2}$ orbital. This stabilization is indeed associated in our calculations with a larger occupation of the $d_{3z^2-r^2}$ orbital compared to the $d_{x^2-y^2}$ one. The anisotropy is reversed for the central layer ($c/a < 1$) and thus the $d_{x^2-y^2}$ orbital is stabilized and presents a larger occupation number. The magnetism is essentially two-dimensional, the BaTiO_3 layers remaining essentially non magnetic. We can thus predict two magnetic behaviors in these super-lattices. The first one is related to the interfacial layers and is associated with a weak in-plane double-exchange and thus a weak Curie's temperature (T_{c1}). The second one is related to the central layers and is associated with a strong in-plane double-exchange and thus a strong Curie's temperature (T_{c2}). One can thus predict three different phases, the first one at very low temperature with all three LSMO layers participating to the magnetization, the second one at larger temperature with only the central layer participating to the magnetism and finally, at high temperature, the paramagnetic phase.

As mentioned earlier in a double-exchange magnetic coupling, the exchange integral is directly proportional to the delocalization between adjacent magnetic ions and thus to the square of the transfer integral between the

magnetic $3d$ orbital and the ligand $2p$ bridging orbital. We did an analytical estimation of the ratio between the (Mn) $3d$ -(O) $2p$ transfer integral for the $d_{3z^2-r^2}$ and $d_{x^2-y^2}$ magnetic orbital. This ratio can be undervalued as

$$\frac{T_{c1}}{T_{c2}} \sim \left| \frac{t_{d_{3z^2-r^2};p_x}}{t_{d_{x^2-y^2};p_x}} \right|^2 \simeq \left| \frac{S_{d_{3z^2-r^2};p_x}}{S_{d_{x^2-y^2};p_x}} \right|^2 < \frac{1}{(4\sqrt{3})^2} = \frac{1}{48}$$

predicting more than an order of magnitude between T_{c1} and T_{c2} .

Another important result is the fact that the BaTiO_3 triple-layer does not preserve its insulating character but exhibit a weak, however non null, density of states at the Fermi level in the dominant LSMO spin orientation. This electronic delocalization through the BaTiO_3 triple-layer insures the ferromagnetic coupling between the different LSMO layers by what we could call a long-range double-exchange mechanism.

As a conclusion, first principles calculations predict for the super-lattice

- two different ferromagnetic phases, associated with two Curie's temperature differing by nearly two orders of magnitude,
- an optimized high Curie temperature expected to be larger than the bulk one (no octahedra tilt).

We thus synthesized $(\text{La}_{2/3}\text{Sr}_{1/3}\text{MnO}_3)_3(\text{BaTiO}_3)_3$ super-lattices using laser-MBE deposition [16] on a $[001]$ -oriented SrTiO_3 substrate.

The pulsed laser deposition was performed in low pressure ($5 \cdot 10^{-4}$ mbar) to enable control of the growth using the Reflexion High Energy Electron Diffraction (RHEED). While the growth of the structures was carried out in low ozone content (0.1% volume), to promote a maximum oxidation of the super-lattice without favorizing a strong diffusion between the different layers, the structures were rapidly cooled in high ozone concentration (7% volume). The deposition of each material, namely BTO and LSMO, was measured and calibrated by growing them independently under the same conditions (620°C , pressure $5\% \times 10^{-4}$ mbar, ozone concentration 0.1% volume) using an in situ RHEED control for a precise control of the growth of each layer. Figure 3 shows two RHEED patterns, taken along the $[110]$ azimuth of the substrate. These two pictures evidence a streaky-like diffraction pattern in agreement with a 2D growth of the super-lattice. The comparison between the two patterns taken at the end of the deposition (A) and at the end of the cooling (B) evidences two features. First, there is no drastic modification of the pattern after the cooling process. Second the in-plane structure of the super-lattice exhibits the same lattice parameters as the SrTiO_3 substrate and thus can be described in a perovskite-like subcell. The B pattern also presents Kikuchi lines indicating the high quality of the deposited film. These lines are not

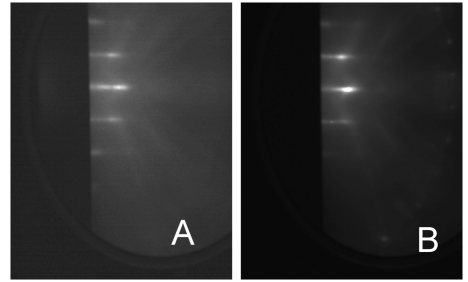


FIG. 3: RHEED patterns along the $[110]$ azimuth of the SrTiO_3 substrate. A: 620°C , 5×10^{-4} mbar ($\text{O}_2 + \text{O}_3$). B: 120°C , 5×10^{-7} mbar.

observed in pattern A, due to the important scattering of the electrons at the deposition pressure.

To achieve a better structural characterization of the super-lattices an X-ray diffraction study was carried out using a 4-circle diffractometer. Two different set of reflections were studied: the out of plane reflections and the so-called asymmetric reflections. From the first set of reflections, the value of the out-of-plane lattice parameter could be calculated for the $[(\text{La}_{2/3}\text{Sr}_{1/3}\text{MnO}_3)_3(\text{BaTiO}_3)_3]_{25}$ super-lattice. We found an out-of-plane lattice parameter of 4.04 \AA , corresponding almost to the value determined from the theoretical calculations. This value is larger than the expected value, calculated from the average out-of-plane parameters of films of the two starting materials. A Williamson-Hall analysis carried out on the $[00l]$ reflections shows that the coherence length of the diffraction along that direction corresponds to 600 \AA , that is the thickness of the deposited film. The same analysis carried out on asymmetrical reflections ($[011]$, $[022]$, ...) indicates that the in-plane lattice parameter is $3,9048 \text{ \AA}$, in agreement with the substrate lattice parameter, and that the coherence length of the diffraction along an in-plane direction is 1000 \AA , in agreement with the size of the terraces observed on the SrTiO_3 substrates.

The magnetic measurements (figure 4, red squares), carried out in the temperature range 4-350 K, evidence two ferromagnetic phases, as predicted. The first one at low temperature ($T < T_{c1} \simeq 25 \text{ K}$) and the second one on the rest of the measurement temperature range, indicating a second Curie temperature T_{c2} far above 350 K. Indeed, the magnetisation remains roughly constant between 100 and 350 K. The Curie temperature of LSMO bulk [8] or thin films [9] is 370 K, unfortunately our SQUID magnetometer does not allow measurements above 350 K and thus check whether the 370 K transition temperature is actually overcome by T_{c2} . We thus synthesised similar super-lattices using the isoelectronic calcium manganite: $\text{La}_{2/3}\text{Ca}_{1/3}\text{MnO}_3$ (LCMO). Indeed, the LCMO presents a bulk [10] and thin films [11] Curie temperature of only 250-260 K, while one can expect sim-

ilar orbital ordering effects as predicted in the strontium super-lattice. Figure 4 reports the magnetisation curve (green circles) of a $[(\text{La}_{2/3}\text{Ca}_{1/3}\text{MnO}_3)_4(\text{BaTiO}_3)_4]_{20}$ super-lattice. As for the LSMO super-lattice, one observes a low Curie temperature (50 K) expected to correspond to the two interfacial manganite layers. In addition one can verify that the magnetic moment does not go to zero even above the Curie temperature of the bulk material, as predicted by theoretical calculations. The

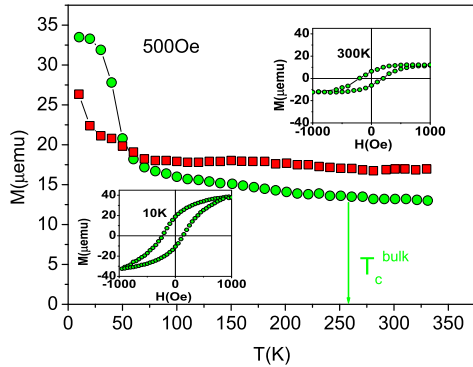


FIG. 4: (Color Online) Magnetization (500 Oersted) for the $[(\text{La}_{2/3}\text{Sr}_{1/3}\text{MnO}_3)_3(\text{BaTiO}_3)_3]_{25}$ (red squares) and $[(\text{La}_{2/3}\text{Ca}_{1/3}\text{MnO}_3)_4(\text{BaTiO}_3)_4]_{20}$ (green circles) super-lattices. Hysteresis loops of the calcium super-lattice, recorded at 10 K and 300 K, are presented in the inserts. The arrow marks the bulk Curie temperature for the calcium compound. The surface of both samples is 16mm^2 .

Curie temperature of the central layer is not observed in this temperature range, the theoretical calculations assumed that this temperature must be higher than 350 K ($48 \times 50 = 2400$). The two inserts of figure 4 show that magnetic loops can be registered both at 10 K and 300 K, the latter temperature being well above the Curie temperature of the bulk material. It means that a ferromagnetic behavior still exists in the super-lattices even at this temperature, in agreement with the theory.

In summary we showed that it is possible to increase the Curie temperature of a manganite ferromagnetic thin film far above the bulk Curie temperature, by the control of orbital ordering. The key idea to reach this result is that the strains applied by the substrate are not sufficient by themselves and that they must be maintained using the regular intercalation of adequate layers. Our findings open new opportunities for the design and control of magnetism in artificial structures and pave the way to novel magneto-electronic applications, including non-volatile magnetic memory working far above room temperature.

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