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Thomas Jourdan, Aurélien Masseboeuf, Alain Marty, Frederic Lançon, Pascale Bayle-Guillemaud.
Magnetic bubbles in FePd thin films near saturation. Journal of Applied Physics, 2009, 106, pp.073913.
10.1063/1.3243318 . hal-00424242

HAL Id: hal-00424242

<https://hal.science/hal-00424242>

Submitted on 14 Oct 2009

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Magnetic bubbles in FePd thin films near saturation

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(Dated: October 14, 2009)

The structure of domain walls delimiting magnetic bubbles in L1₀ FePd thin layers is described on the basis of Lorentz transmission electron microscopy (LTEM) and multiscale magnetic simulations. Images obtained by high resolution LTEM show the existence of magnetization reversal areas inside domain walls, called vertical Bloch lines (VBL). Combining these observations and multiscale simulations on various geometries, we can identify the structure of these VBL, notably the presence or not of magnetic singularities.

I. INTRODUCTION

FePd alloys with L1₀ structure deposited in thin layers have attracted much attention because of their very high perpendicular anisotropy, which is a key property for magneto-optical recording and for high density magnetic storage. Recently, alloys with a perpendicular anisotropy have been used in spin-valves, where they are used as the polarizer and as the free layer that should be reversed¹. It has been shown that in such devices² or in magnetic tunnel junctions, the reversal of the free layer occurs through the nucleation of a reversed domain followed by the propagation of a domain wall.

Near the saturation, the band domain structure in FePd layers transforms into a lattice of magnetic bubbles³, which remains stable at high fields. In some bubbles the Bloch-like walls have different polarities, separated by segments called vertical Bloch lines (VBL). In the present work we analyze the role of VBL on the shape of these magnetic bubbles.

VBL were much studied in the 1980s in garnets, both experimentally and numerically. Typical parameters for these garnets are $K = 10^3 \text{ J.m}^{-3}$, $M_s = 1.4 \times 10^4 \text{ A.m}^{-1}$ and $A = 1.3 \times 10^{-12} \text{ J.m}^{-1}$, so that the domain wall width is $\delta = \pi \sqrt{A/K} \approx 0.1 \text{ } \mu\text{m}$. This large value, compared to the domain wall width in FePd of around 8 nm, makes possible the optical observation of domain walls and VBL⁴. In FePd, a higher resolution is necessary to probe the sample, which can be reached by Lorentz transmission electron microscopy (LTEM). Extensive analytical and numerical studies have also been performed in VBL in garnets^{5,6,7,8}. Given the high value of the quality factor $Q = 2K/(\mu_0 M_s^2) \approx 8$, a common assumption in the models is $Q \gg 1$, which notably permits to use a local approximation of the demagnetizing field and thus simplifies the calculations. This assumption is *a priori* not valid in the case of FePd, which exhibits smaller values of Q of the order of 1.6.

In the present work we performed high resolution imaging of domain walls in magnetic bubbles in FePd thin layers, using Lorentz microscopy, to highlight their magnetic configuration. In particular we describe the influence of VBL on the shape of the bubbles. We also show the

results of multiscale simulations that provide an explanation for these observed shapes.

II. OBSERVATION OF MAGNETIC BUBBLES IN FEPD THIN FILM

Lorentz microscopy is now a well established method that enables magnetic imaging with a resolution better than ten nanometers. The simplest mode of LTEM is the observation of the overlapping of electrons experiencing different Lorentz forces in magnetic domains. The contrasts obtained by simply defocalizing the lens used for imaging are called Fresnel contrasts⁹. In a classical in-plane magnetization configuration, Fresnel contrasts appear on the domain walls position due to the overlapping of electrons coming from two opposite domains. In the particular case of FePd, where magnetization is mainly out-of-plane, the contrasts can be obtained by tilting the sample¹⁰. This enables the magnetization inside the domain to act on the electron beam and to produce traditional Fresnel contrasts located on the domain walls. Otherwise contrasts can be produced by the domain walls themselves if the layer is thick enough and if the amount of in-plane magnetization in the wall is large enough¹¹ (*i.e.* to reach the LTEM sensitivity of about 10 nm.T). This was the case for our samples, so we have performed Fresnel observations of Bloch walls without tilting the FePd layers. The microscope used in these observations was a JEOL 3010 fitted in with a Gatan imaging filter for contrast enhancement¹². The images displayed in this letter have been also filtered by a Fourier approach to enhance the contrasts localized on the domain walls. The magnetization was performed using the objective lens, calibrated with a Hall Probe. The sample was prepared by Molecular Beam Epitaxy on MgO [001] substrate. The magnetic stacking is decomposed in two layers: a “soft” layer of 17 nm FePd₂ having a vanishing anisotropy is deposited before a 37 nm-L1₀ layer of FePd. Details can be found in Ref. 13. The sample was prepared for TEM observation with a classical approach: mechanical polishing and ion milling.

Fig. 1 shows two different areas of the foil during

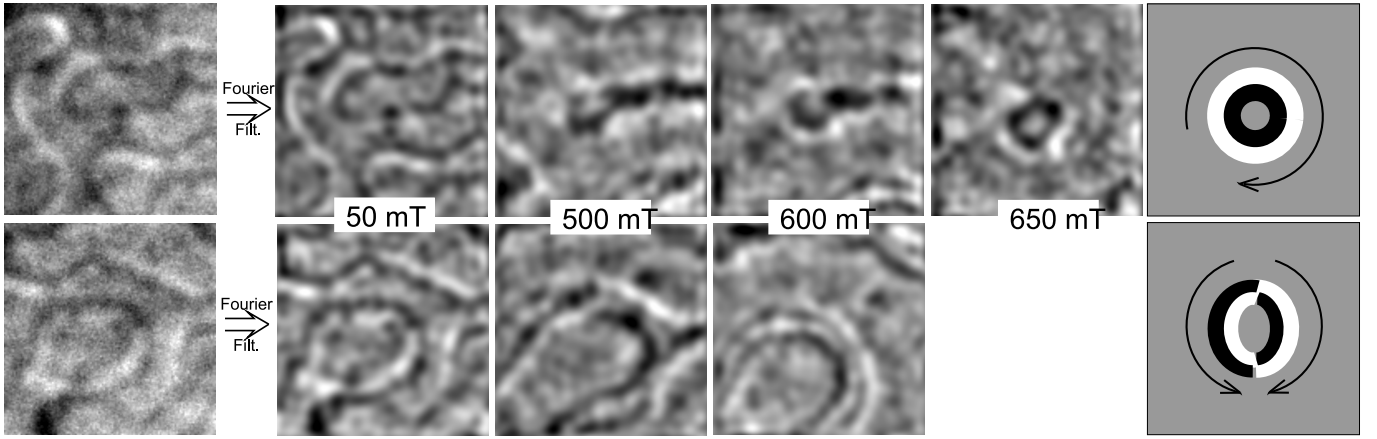


FIG. 1: Magnetization process on FePd thin film. The two rows present two different areas in the film. Both of them present a magnetic bubble state just before saturation. Left images are raw datas while the other images are enhanced by Fourier filtering. Right images are simple schemes to highlight the contrast observed in the last step of magnetization process. Arrows point out the direction of magnetic induction in bubbles. Images are 500×500 nm.

the magnetization process. We observe couples of black and white contrasts corresponding to the Bloch walls¹¹. These pictures have been obtained for increasing applied fields. We should notice that upon 500 mT the quality of the images decreases due to the action of the objective lens on the image formation. Nevertheless it is possible to follow the shape of the domains during the magnetization process (enhanced here by Fourier filtering). We observe in both cases that a magnetic domain collapses to a bubble state. Attention can thus be paid on the chirality of the Bloch wall. The chirality (sense of the magnetization inside the Bloch wall) is directly linked to the Fresnel contrast: the wall chirality of a black/white contrast and the chirality of a white/black contrast are opposite. Knowing this, the observation of the two magnetic bubbles presented in the right images of Fig. 1 gives some information on the magnetization inside the domain walls of the bubbles. The first bubble presents a continuous domain wall, swirling all around the bubble, whereas the other one exhibits two different parts with the same magnetization orientation. In the latter configuration, the magnetization inside the domain wall experiences two rotations of 180° localized at the top and the bottom of the bubble. These switching areas are known as vertical Bloch lines (VBL). One can notice the main difference in the two bubble shapes: the first one is almost round while the second bubble seems to be slightly elongated along the vertical direction.

To confirm the role of VBL on the bubble shape, we have thus simulated the inner structure of domain walls containing VBL.

III. SIMULATION OF DOMAIN WALLS WITH VERTICAL BLOCH LINES

The numerical simulation of magnetic bubbles is not a tractable problem with standard codes. Indeed it requires to handle large systems whose size is related to the size of the bubbles, but with regions where the magnetization varies rapidly in space, such as domain walls and all their substructures. Considering all regions with the same level of refinement is clearly not well adapted to such a multiscale problem and leads to a high computational effort. The same level of accuracy can be reached with a coarser mesh in uniformly magnetized regions.

In this work we used a multiscale code (Mi- μ Magnet) based on an adaptive mesh refinement technique, as well as on a mixed atomistic-micromagnetic approach, to achieve both precision and computational efficiency¹⁴. Given the large size of the systems we envisage here, the code was only used in its micromagnetic mode. It has been recently shown that micromagnetic calculations can be applied to singularities appearing in VBL, called Bloch Points (BP)¹⁵. In all calculations the mesh step is kept lower than half the exchange length.

Parameters are chosen in agreement with experimental measurements¹⁶: the saturation magnetization, anisotropy constant and exchange stiffness are $M_s = 10^6$ A.m⁻¹, $K = 10^6$ J.m⁻³, and $A = 7 \times 10^{-12}$ J.m⁻¹. With such parameters, the exchange length is $l_{ex} = \sqrt{2A/(\mu_0 M_s^2)} = 3.3$ nm.

Two types of computations have been carried out. First we investigate the properties of a straight domain wall containing a VBL. Secondly we study the role of VBL on the shape of the magnetic bubbles in FePd layers.

A. Vertical Bloch lines in straight domain walls

The system used here contains a domain wall with a single VBL (Fig. 2). The lateral size of the system is $110 \text{ nm} \times 110 \text{ nm}$ and the thickness of the layer is varied between 11.3 to 37.6 nm. No periodic boundary conditions are used, because this would involve a second VBL along y and a second domain wall along x . A view of such a system is given in Fig. 3.

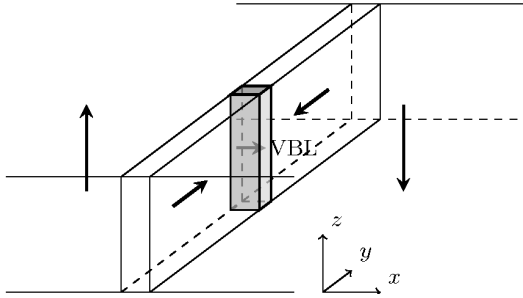


FIG. 2: Schematic representation of the system used to study the structure of VBL in domain walls.

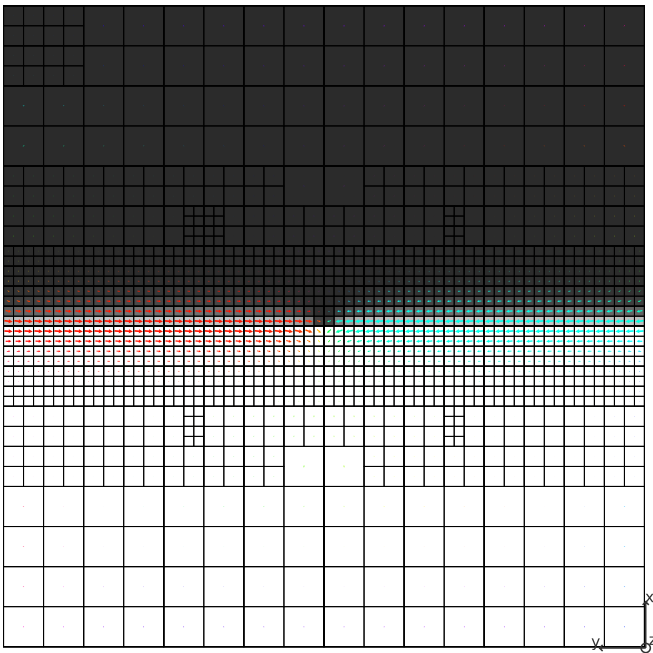


FIG. 3: (Color online) Cross-section of the whole system obtained from a multiscale simulation (Bloch wall, with a vertical Bloch line containing a Bloch point). The orientation of the magnetization for the in-plane component is given by the arrows and the color wheel, and by the grayscale for the out of plane contribution (along z). The norm of the arrows is proportional to the in plane magnetization. The lateral size of the system is $110 \text{ nm} \times 110 \text{ nm}$.

For all the values of the thickness h that we envisage, we consider configurations with and without a BP (Fig. 4 and 5). The configuration without a BP can be

stabilized only for a thickness lower than 15 nm and becomes energetically favorable below a critical thickness of around 13 nm (Fig. 6). For thicknesses larger than 15 nm, well defined Néel caps are present due to the dipolar field created by the domains and a BP nucleates on the surface where the magnetization rotates of nearly 360° (Fig. 5, at $z = 0$). It must be noted that the critical thickness is around $4 l_{ex}$, which is significantly lower than the value $7.3 l_{ex}$ found by a variational method¹⁷. Indeed this method, based on a local approximation of the dipolar field, is well justified if $Q \gg 1$ but does not hold in our case.

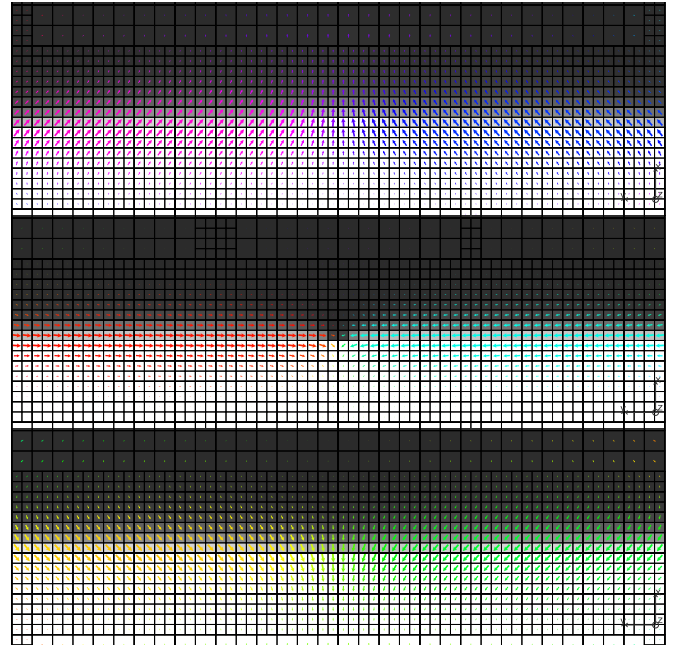


FIG. 4: (Color online) Cross-section along the planes $z = h$, $z = h/2$ and $z = 0$ (from top to bottom) with a VBL containing a BP.

An interesting feature of the domain wall containing a VBL without a BP is the so-called buckling of the magnetization near the line. This buckling was already described in garnets, where it was ascribed to the diminution of the magnetic charges created by the variation of the magnetization in the direction orthogonal to the wall⁸ (which we note x here). These charges are called π charges or dipolar charges, in analogy with π orbitals, because positive charges are associated with negative charges.

Analytical models, based on the assumption $Q \gg 1$, and two dimensional simulations, predict a much smaller value of the buckling⁸. Given our material parameters, it would be less than 2 nm, whereas it is around 10 nm for all the thicknesses we have considered. Three dimensional simulations for $Q = 7.7$ and thick garnet layers ($h \approx 50 l_{ex}$) also give a tiny buckling. In this case, a tilt of the wall is observed in the $x - z$ plane that provides a compensation for the charges associated with the variation of the magnetization along y (called σ or monopolar

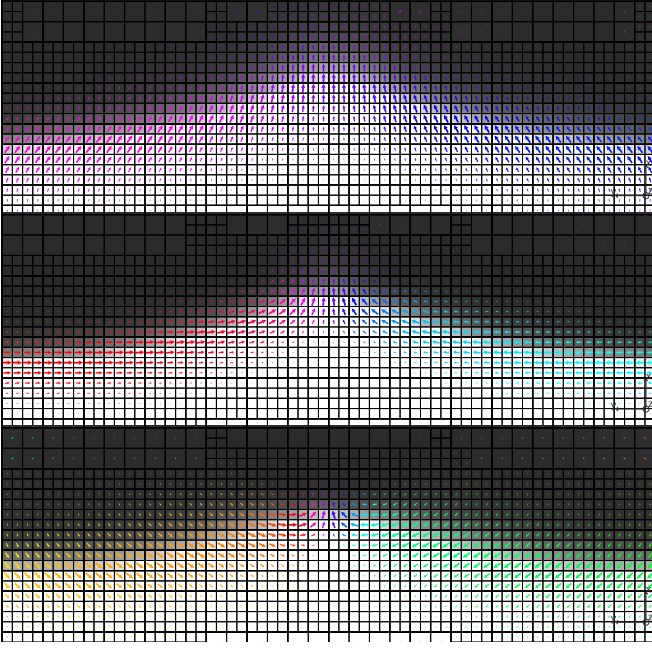


FIG. 5: (Color online) Cross-section along the planes $z = h$, $z = h/2$ and $z = 0$ (from top to bottom) with a VBL containing no BP.

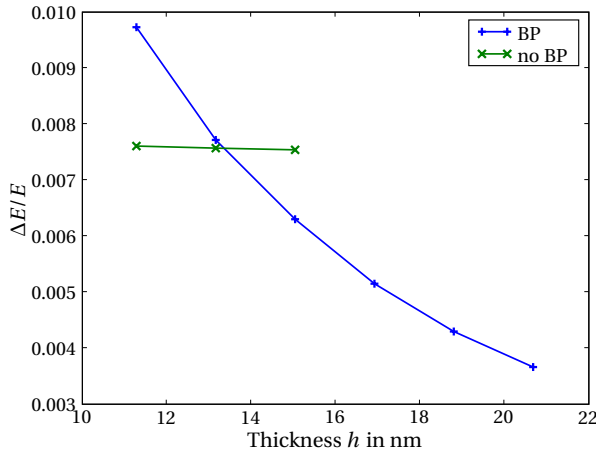


FIG. 6: (Color online) Relative energy of a system with and without a BP in the VBL. The reference energy is given by the system containing a domain wall without a VBL. The decrease of the energy when a BP is present is mainly due to the dipolar term.

charges)¹⁸.

Such a deformation is not present in our simulations. As shown in Fig. 7 and 8, the compensation of the σ charges is achieved by the buckling itself. It can be noted that this buckling is due to the dipolar term, although a small decrease of the exchange energy is also observed in the presence of buckling. Indeed, we have represented in Fig. 7 the magnetic charges $-\partial m_x/\partial x$ (π charges), $-\partial m_y/\partial y$ (σ charges) and the total charges after a trans-

formation $\phi \rightarrow -\phi$ on the configuration of Fig. 5. The angle ϕ refers to the orientation of the magnetization in the plane of the layer. The configuration after the minimization of the energy is shown in Fig. 8. The deformation of the domain wall has reversed, whereas the exchange energy was invariant under the transformation. This indicates that the deformation must be ascribed to the compensation of π and σ charges, which cannot really be distinguished, given the moderate value of Q . Incidentally, the name “ σ charges” is not really adapted to our case given that positive charges are associated with negative charges along y .

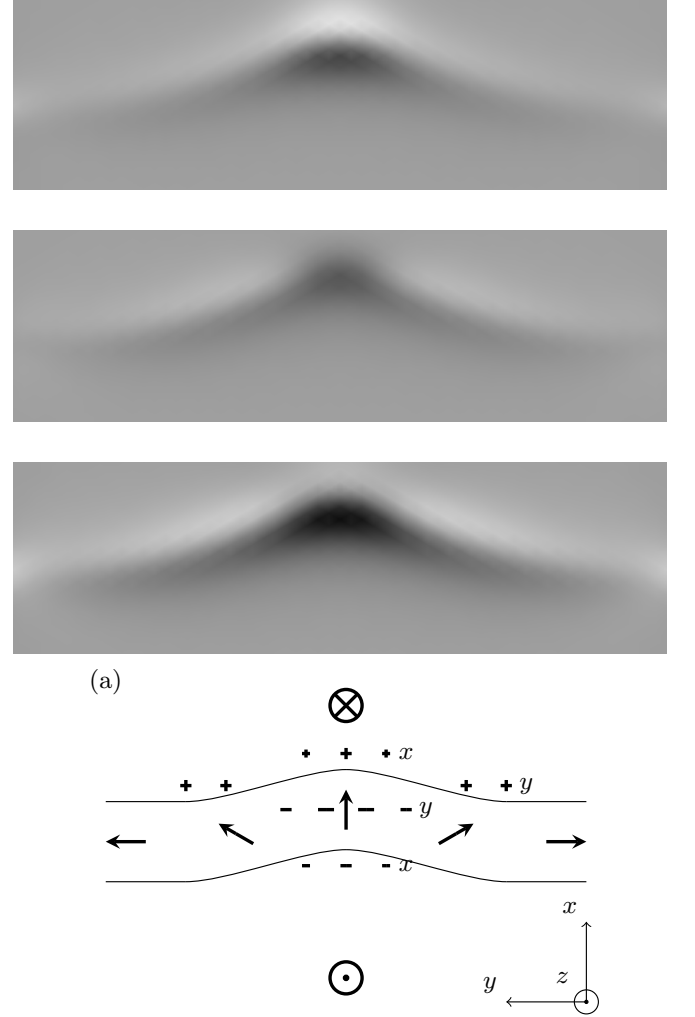


FIG. 7: Magnetic charges in the plane $z = h/2$ corresponding to the configuration of Fig. 5 when the transformation $\phi \rightarrow -\phi$ is performed on the magnetization and the configuration is left unrelaxed. From top to bottom: charges associated with the variation of m_x , m_y and total charge. The charges due to the variation along z are the same before and after the transformation and are not represented. Positive and negative charges are represented respectively by light and dark gray tones. On the schematic the letters x and y refer to charges due to the variation of m_x and m_y .

Unfortunately, no conclusions can be made from exper-

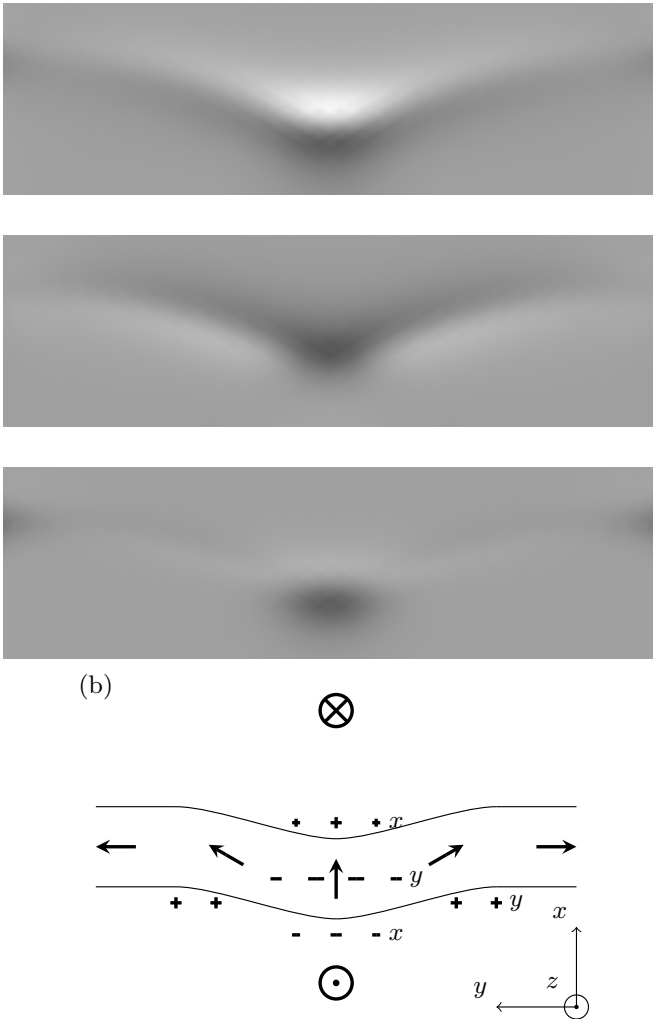


FIG. 8: Magnetic charges in the plane $z = h/2$ after the relaxation of the configuration in Fig. 7.

imental datas on the straight wall. As seen in Fig. 1, at low fields the domain walls in this sample are not straight enough. On the contrary, it is possible to simulate entire bubbles and thus to reproduce the geometry of domain walls near saturation.

B. Magnetic bubbles

It seems reasonable to think that the deformation observed in straight domain walls can be responsible for the distorted shape of the magnetic bubbles. However, the curvature of the magnetic bubbles is such that the domain wall cannot be considered as a straight object. The presence of two VBL in a bubble, that bear opposite σ charges and thus attract themselves, may also affect the distortion. Therefore it is necessary to perform the simulation of entire magnetic bubbles.

The system considered in these simulations contains a magnetic bubble centered in a square of length 218 nm.

Three thicknesses are envisaged: 15 nm, 20.7 nm and 37.6 nm. Periodic boundary conditions are used along x and y to simulate an array of bubbles. The distance between the bubble's centers is thus 218 nm and is close to the experimental value of about 250 nm. The use of an adaptive mesh refinement technique permits to decrease the number of variables by a factor of around 8.

Stability of bubbles is achieved for applied fields between two critical values: if the field is too high, the bubble collapses, and if the field is too low, the bubble transforms into a stripe domain pattern¹⁹. For a thickness of 37.6 nm, we find that the collapse field is between 0.6 and 0.7 T, close to the experimental value of 0.8 T.

For thicknesses of 20.7 nm and 37.6 nm, it is not possible to stabilize the configuration with two VBL without a BP. As observed for straight domain walls, two BP nucleate because of the dipolar field. The bubbles with VBL containing BP are found to be almost circular (Fig. 9). The small distortion may be ascribed either to the interaction between the two VBL which possess opposite charges, or to a local stiffness due to the presence of the BP.

For a thickness of 15 nm, the configuration containing VBL with BP is not stable and the two BP migrate towards the two opposite surfaces of the system. The two regions that exhibit high spatial variations of the magnetization (360° rotation for straight domain walls) are thus located on opposite sides of the system (Fig. 10). This disappearance of the two BP is associated with a deformation of the domain wall, in agreement with the one found on straight domain walls in the previous section and with experimental results. Likewise the charges are minimized and the exchange energy decreases.

It is worth noting that the magnetization in the two lines is oriented in the same direction. This is called the winding configuration³. Lines with opposite orientations of the magnetization constitute the unwinding configuration, and have found to be unstable: the two lines annihilate and the bubble is circular. Indeed, in order to minimize charges in both VBL the bubble would have a "heart"-like shape, which is not favorable. The orientation in the two lines is close to the orientation in the rest of the domain wall at $z = h/2$.

A further step can be made towards the comparison between simulated and experimental configurations by simulating Fresnel contrasts that would be obtained from the multiscale calculations. They are given in Fig. 11. Beside the result corresponding to Fig. 10, we report the results for a bubble without a BP. It can be seen that the position of the contrast and the shape of the bubble agree fairly well.

Despite the good agreement on the shape of bubbles, the transition from the BP-free to the BP configuration does not occur at the same thickness experimentally and in the simulations. Indeed, the configurations without BP are not stable in our simulations for a thickness of 37.6 nm (and even 20.7 nm), whereas according to the deformation of the bubbles observed in the samples, VBL

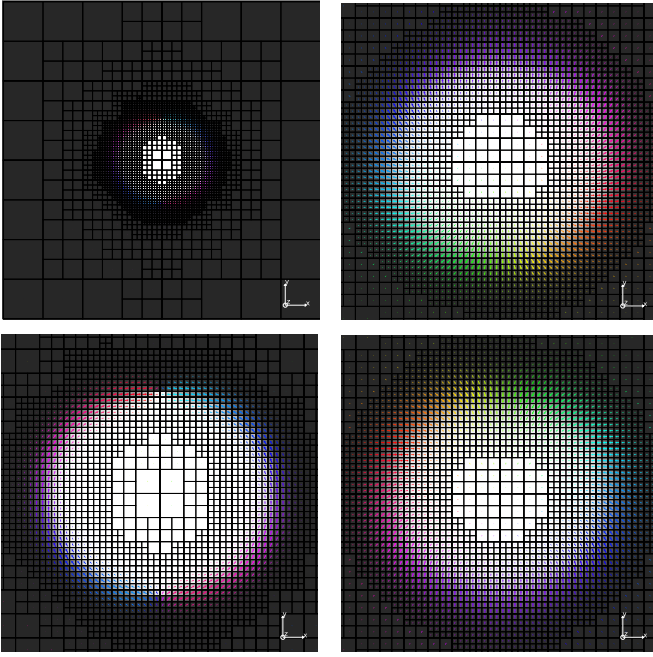


FIG. 9: (Color online) Cross-sections of a system containing two VBL with a BP. From top left to bottom right: whole system at $z = h/2$ (lateral size $218 \text{ nm} \times 218 \text{ nm}$), zoom at $z = h$, $z = h/2$ and $z = 0$ (lateral size $90 \text{ nm} \times 90 \text{ nm}$). The system is 20.7 nm thick and a field of 0.3 T is applied. The largest cell lateral size is 27.3 nm , while the smallest is 1.7 nm .

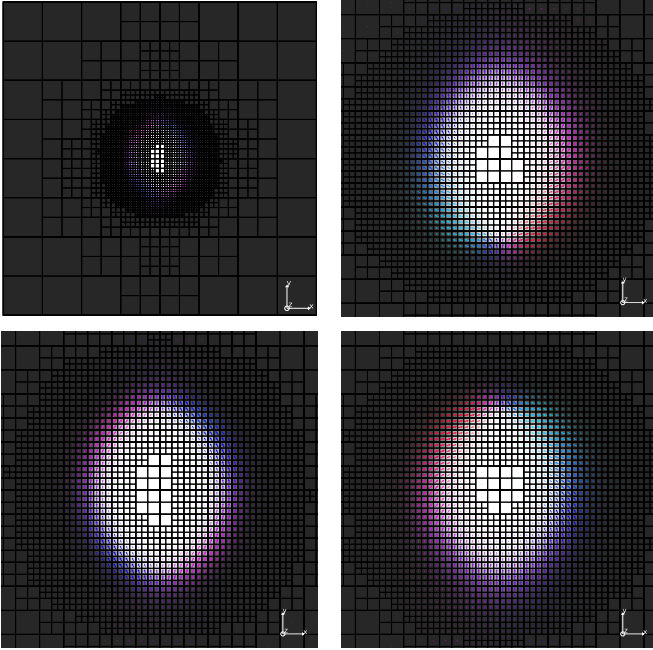


FIG. 10: (Color online) Cross-sections of a system containing two VBL with no BP. From top left to bottom right: whole system at $z = h/2$ (lateral size $218 \text{ nm} \times 218 \text{ nm}$), zoom at $z = h$, $z = h/2$ and $z = 0$ (lateral size $90 \text{ nm} \times 90 \text{ nm}$). The system is 15 nm thick and a field of 0.25 T is applied.

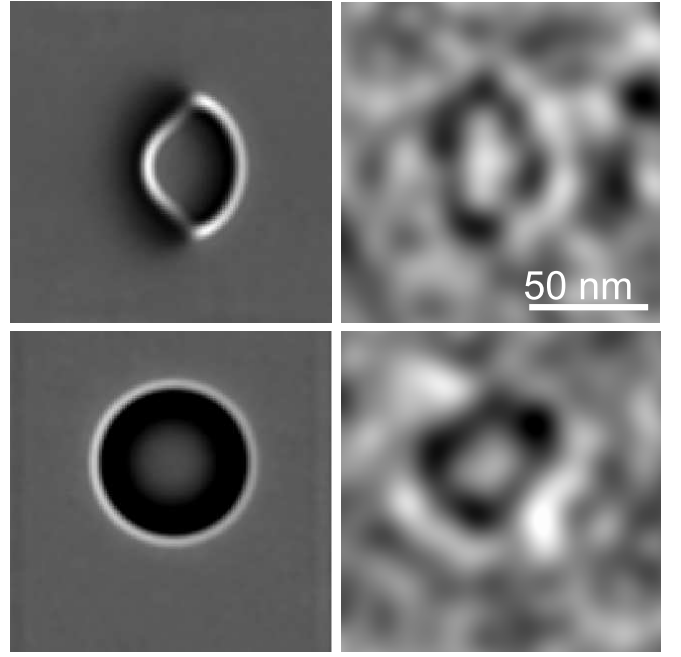


FIG. 11: Comparison of simulated Fresnel contrasts and experimental contrasts for the two type of bubbles observed. The defocalisation used is $100 \mu\text{m}$.

contain a BP at this thickness. One reason for this discrepancy may be the presence of the soft layer on which the $L1_0$ layer is deposited. The exchange and demagnetizing contributions to the energy are modified due to the different closure of the magnetic flux. The thickness of the bottom Néel cap increases¹³, which induces a dissymmetry in the system and could favor the configuration without BP.

IV. CONCLUSION

Using Lorentz transmission electron microscopy on FePd samples and multiscale simulations, we have shown that it is possible to determine the magnetic structure of domain walls as thin as 8 nm . The presence of vertical Bloch lines in some bubbles has been demonstrated by microscopy. Bubbles containing two vertical Bloch lines exhibit a distortion of the classical circular shape. The simulation of entire bubbles has been possible thanks to the multiscale approach and has revealed that the deformation observed experimentally is a signature of the absence of Bloch points inside the vertical Bloch lines. For straight domain walls in FePd, we predict a larger buckling than previously reported for other materials.

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