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Cracking of titania nanocrystalline coatings

Marie Mahé, Jean-Marc Heintz, Jürgen Rödel and Peter Reynders

1. Introduction

Titanium oxide-coated platelet-like substrates are largely used in the industry of pearlescent pigments. The most common substrate used for the precipitation of the titanium oxide layers in pigments is the muscovite mica, due to its availability. Synthetic substrates, like SiO₂ platelets, Al₂O₃ platelets, glass platelets and synthetic mica have been developed during the last two decades. These substrates allow, for example, a better control of the thickness and/or of the impurities. Nevertheless, TiO₂-coated natural mica remains commercially by far the most important family because of low cost of the mica substrate in comparison to synthetic substrates.[1] and [2] TiO₂-coated mica studied in this work was obtained using the chloride process in which a TiOCl₂ solution is slowly added to a mica platelet suspension at low pH and temperatures of 60–90 °C. The synthesis of TiO₂-coated platelet-like substrates produced by Merck has already been optimised in terms of TiO₂ precipitation, lustre and chroma. Nevertheless, the presence of cracks can still be observed at the surface of some of the pigments. In order to continuously improve the quality of the pigments, crack formation at the surface of inorganic pigments needs to be understood.

Cracking of inorganic films has repeatedly been reported in the literature.[3], [4], [5], [6], [7], [8], [9], [10], [11], [12], [13] and [14] Depending on the systems studied, several reasons have been proposed for crack formation. The drying of porous material, like sol-gel bodies[3], [4], [8] and [13] or granular ceramics[5], [6], [7], [9], [10], [11], [12] and [14] is often seen as a decisive step for the appearance of cracks at the surface of these materials. In these cases, drying stresses develop which are linked to the tension forces that exist in the liquid inside the pores of the drying body. Crack formation is also reported to occur during sintering of ceramic films for cases with differential sintering leading to strain incompatibilities.[5], [6], [7] and [11] During sintering, cracks favourably develop from intrinsic defects. Finally, problems of crack formation and decohesion can also develop during cooling of annealed ceramic films deposited on a substrate due to differences between the thermal expansion coefficients of the substrate and the coated film.[15] and [16]

An improved understanding of the cracking of TiO₂ nano-layers deposited on platelet-like substrates is presented in this study. The effects of the drying step and the calcination step on cracking are treated. Moreover, the influence of layer thickness as well as of the underlying substrate has been examined, with respect to their acido-basicity, crystallinity and roughness.

2. Experimental

2.1. Synthesis of the samples

Titanium oxide layers were deposited on different platelet-like substrates via an aqueous liquid deposition process (LDP). [1], [2] and [17] It corresponds to controlled precipitation, at low pH (~ 2), of titanium chloride within an aqueous suspension containing the substrates. The TiO_2 rutile structure leads to a higher refractive index than the TiO_2 anatase structure.² Therefore, a process has been developed at Merck to create a TiO_2 rutile layer on mica by precipitating first a very thin SnO_2 layer on the substrate, which acts as a template for the TiO_2 rutile. The preparation of the samples used in this publication has been described in the literature.¹⁷ The thickness of the TiO_2 layers can be accurately controlled (± 5 nm) by varying the amount of titanium chloride solution. The precipitates were filtered off, washed completely with deionized water and dried at 110 °C for 12 h. The dried pigments were then sintered at 850 °C for 30 min.

The TiO_2 layer thickness deposited on the substrates were equal to 60 nm, 140 nm and 300 nm after sintering. Natural mica, SiO_2 platelets and Al_2O_3 platelets were utilized as substrates. To describe the samples studied in this work, the following type of notation will be used: "Ti-140-mica" with "Ti" for TiO_2 layer, "140" for TiO_2 -thickness layer equal to 140 nm after sintering and "mica" for the substrate used.

2.2. Substrates

The titanium oxide layers were precipitated on three different substrates: mica platelets (Merck, diameter 10–50 μm , thickness range 300–800 nm, mean thickness 400 nm), SiO_2 platelets (Merck, diameter 10–50 μm , thickness 450 ± 10 nm) and Al_2O_3 platelets (Merck, diameter 10–50 μm , thickness 200 ± 10 nm). The mica employed is the natural dioctahedral muscovite $\text{KAl}_2[\text{AlSi}_3\text{O}_{10}](\text{OH})_2$.¹⁸ The surfaces of the mica muscovite platelets are oriented along the (0 0 1) plane.¹⁹ The synthetic SiO_2 platelets are produced by a web-coating process.²⁰ These substrates are amorphous. Al_2O_3 platelets ($\ast\text{-Al}_2\text{O}_3$, corundum) were produced using a controlled crystal growth process in molten sodium sulfate.²⁰ The surface of the platelets is preferentially orientated according to the (0 0 1) plane of $\ast\text{-Al}_2\text{O}_3$ corundum. There is no correlation between the lattice parameters of mica or Al_2O_3 platelets (corundum) with the lattice parameters of SnO_2 and TiO_2 rutile or anatase. The points of zero charge (PZC) values, given in the literature, for the oxides constituting the substrates are the following: $\text{PZC}_{\text{SiO}_2} \sim 2$ and $\text{PZC}_{\text{Al}_2\text{O}_3(0\ 0\ 1)} \sim 5\text{--}6$. [18] and [21] Since the main plane developed at the surface of mica is close to SiO_2 , the PZC of mica can be considered close to the SiO_2 value.

Most of the experiments have been carried out with the three previously quoted substrates: natural mica, SiO_2 and Al_2O_3 platelets. Nevertheless, some experiments have been performed using synthetic mica and glass platelets. In fact, the surfaces of SiO_2 , Al_2O_3 and glass platelets are

smoother than those of natural and synthetic mica. Also, the chemical nature of synthetic and natural mica (respectively fluorophlogopite and muscovite) is different.

2.3. Characterization of pigments

The layers consist of oxide particles, which are called grains throughout this publication to make a clear distinction from the whole pigment particles themselves that consist of these films attached to substrates. In a given sample, not all pigment particles contain cracks. Therefore, the average percentage of pigment particles exhibiting cracks in a sample has been determined, using light microscopy. For ease of reading, the term "crack percentage" is used in the following instead of "average percentage of pigment particles". Micrographs of the samples were taken using an Eclipse ME 600 (Nikon GmbH) under reflection conditions. It has also been used to determine the evolution of the cracks between the drying and the sintering steps. For this purpose, dried pigment particles were deposited on a sapphire holder on which markers had been inscribed. The dried particles were observed through the optical microscope; then the system composed of the dried particles on the sapphire holder was sintered at 850 °C for 30 min. Thanks to the markers on the sapphire holder, the pigments previously observed in their dried state were identified and characterized after sintering.

The morphology of the TiO₂ layers and of the substrates was studied by scanning electron microscopy (SEM) (LEO 1530 Gemini). SEM was also used to measure the layer thickness on micrographs of fractured pigments. The pigments were mixed with a lacquer and dried. The lacquer, containing the pigments, was then broken in such a way that some of the particles were also fractured which allowed the thickness of the coating film to be measured.

The Brunauer–Emmett–Teller (BET) specific surface areas and the pore volumes were measured using a Micromeritics ASAP 2400 apparatus with nitrogen at 77 K. Assuming that the surface area of the substrates is negligible and that the TiO₂ grains attached to them are spherical, the particle size G_{BET} of dried TiO₂ layers was evaluated. (The specific area of mica, SiO₂ platelets and Al₂O₃ platelets are 2.8, 3.05, and 1.8–3.0 m²/g, respectively.) The weight percentage of TiO₂ in the pigments is taken into account for determination of the grain size:

$$G_{\text{BET}}(\text{nm}) = \frac{6000 f_{\text{TiO}_2}}{S_{\text{BET}} \times \rho_{\text{TiO}_2}}$$

Here f_{TiO_2} is the TiO₂ weight oxide fraction, S_{BET} is the BET surface area (m²/g) and ρ_{TiO_2} is the theoretical density of titanium oxide (4.26 g/cm³ (for rutile) and 3.85 g/cm³ (for anatase)). The relative densities ρ_{rel} of the TiO₂ layers were also deduced from nitrogen adsorption measurements. Finally, the average pore radius \bar{r}_p of the sample is also calculated from the nitrogen adsorption analyses according to the following formula (Eq. (1)):

(1)

$$\bar{r}_p = \frac{2 \times V_{tot}}{S_{BET}}$$

The average pore radius \bar{r}_p is different from the most frequent pore size deduced from the pore size distribution but provides a rough estimate of the pore size in a sample.

X-ray analyses (Bruker AXS D5000 θ/θ diffractometer) were carried out to determine the crystallite size of the titanium oxide precipitated on the substrates. The radiation used was Cu K α 1. The crystallite size (DXRD) was estimated from the full width at half-maximum of diffraction peaks using the Scherer equation (Eq. (2)):

(2)

$$D_{XRD} = \frac{0.9\lambda_{Cu\ K\alpha_1}}{\sqrt{B^2 - b^2} \cos \theta}$$

Here θ is the scattering angle, B is the peak width at half height and b the peak width at half height of the reference substance (LaB $_6$). The wavelength of the X-ray is $\lambda_{Cu\ K\alpha_1}$ =0.1541 nm.

Pycnometry analyses were performed in ethanol at 20 °C for SiO $_2$ platelets dried at 110 °C, and sintered at 600 or 850 °C.

The AFM results reported in this paper were obtained in air at room temperature using a Park Scientific Instruments Autoprobe CP scanning probe microscope (SPM) operating in the non-contact mode.

3. Results

3.1. Substrate morphology

SEM observations show that surfaces both from natural and synthetic mica platelets exhibit defects (Fig. 1). The presence of step-like defects at their surface is related to the easy cleavability of mica. These defects can be detected at the surface of natural mica as well as synthetic mica. According to the SEM micrographs, SiO $_2$, Al $_2$ O $_3$ and glass platelets are smoother.

AFM images were recorded on three small defect-free areas (edge length about 2–5 μ m) that were selected from different regions. For each substrate, two platelets were analysed and the average roughness determined. The synthetic mica exhibits a rather high average roughness (.9 nm) in contrast to natural mica where it is rather low (.1 nm). The average roughness of glass platelets (.5 nm) and SiO $_2$ platelets (.4 nm) is smaller than the value for synthetic mica but larger than the average roughness of natural mica. Finally, the average roughness of the Al $_2$ O $_3$ surface is very small (<1 nm).

Mica substrate surfaces exhibit steps, especially close to the edges of the mica particle. The height of the steps of synthetic and natural mica was determined: average values for these two types of mica are around 20–100 nm (Fig. 2). Occasionally, rare step-like defects on Al₂O₃ or SiO₂ can also be found.

3.2. Thermal evolution of the substrates

Mica and alumina platelets, which are subjected to temperatures higher than 850 °C before their use in the LDP process, do not exhibit any significant modification as regards to their size and mass during the sintering of the pigments. In contrast, pigments based on SiO₂ can shrink in the lateral direction during the heat treatment. Optical microscopy was employed to determine the in-plane shrinkage of non-coated SiO₂ platelets during the sintering process. Between 110 and 850 °C, shrinkage is about 6–8%. These values are close to the ones that have been measured on pigment particles (-6%).

The volume shrinkage of SiO₂ platelets was also investigated using pycnometry analyses in ethanol. SiO₂ platelets dried at 110 °C or sintered at 600 °C or 850 °C were measured. The densities deduced from these experiments and the corresponding volume shrinkages are provided in Table 1. Densification occurs mostly between 600 and 850 °C. Moreover, qualitative agreement is obtained between 2D observations and pycnometry analyses since the final experimental value of the volume shrinkage ($\Delta V/V = 14\%$) is about the same order of magnitude of the value that can be calculated from the linear shrinkage determined by optical microscopy ($\Delta V/V = (1-\Delta l/l)^3 \approx 17\%$).

A thermogravimetric analysis (TGA) of SiO₂ platelets previously dried at 110 °C was also performed from 25 to 900 °C. The weight loss is equal to 7.3% at 700 °C. About 3.8% of the weight loss occurs before 200 °C and 3.4% of the weight is lost from 200 to 700 °C. The major weight loss occurs before 600 °C, before the volume shrinkage of SiO₂ platelets takes place.

3.3. Pigment particles

The pigments obtained after the LPD process and drying are presented in Fig. 3. A pigment sample is composed of thousands of pigment particles. The diameter of a pigment particle is about 10–50 µm and TiO₂ layer thicknesses are about 50–300 nm.

3.4. TiO₂ layers

Pore and grain sizes from the different dried TiO₂ layers are reported in Table 2. The average pore radius \bar{r}_p is about 2 nm. This value is quite consistent with a theoretical radius of a pore (r_{th}) surrounded by 3 crystallites of radius R, that can be calculated from the following equation¹⁴ :

$$(R + r_{th})\cos 30^\circ = R.$$

In our case, R is taken from BET measurements: $R = \text{GBET}/2$ and $10 < \text{GBET} < 30$ nm, providing a theoretical pore radius ranging from 0.8 to 2.3 nm.

TiO₂ grains are confirmed as monocrystalline since values given by BET and X-ray diffraction measurements are very similar. The grain size seems to increase with layer thickness according to both types of measurements.²⁶

The SEM micrographs of the microstructure of dried TiO₂ layers in Ti-140-Al₂O₃ and in Ti-300-Al₂O₃ demonstrate that the TiO₂ granular layer is precipitated homogeneously and regularly on the substrate (Fig. 4). The film thickness in a given pigment remains constant. Grain size of the dried layers is in the nanometric range as shown by BET and XRD analyses.

3.5. Crack percentage

As already mentioned, the preparation of inorganic layer-substrate pigments is composed of several steps. After precipitation, the pigment particles are dried and then sintered at 850 °C. The presence of cracks is detected after the drying step (Fig. 5), although the cracks are then barely visible.

After sintering, cracks are more easily detected with light microscopy or SEM and the crack percentage was determined. In most cases, several samples corresponding to the same type of pigments have been studied to check the reproducibility of the results. The influence of different parameters (layer thickness, substrate nature, precipitation conditions) on crack percentage after sintering is reported in Table 3. It can be noted that there is a critical thickness beyond which the crack percentage increases (see Ti-60-mica and Ti-140-mica or Ti-140-Al₂O₃ and Ti-300-Al₂O₃). This critical thickness is smaller when TiO₂ layers are precipitated on mica (between 60 and 140 nm) than when they are precipitated on Al₂O₃ and SiO₂ platelets (>140 nm).

3.6. Crack evolution during sintering of pigments

The evolution of cracks between the drying and the sintering steps has been analysed for the different pigments. Examples of mica based pigments are presented in Fig. 6. Cracks present at the surface of sintered pigment particles are more visible than cracks present at the surface of as-dried pigments. Nevertheless, the crack percentage of dried and sintered samples is about the same, which indicates that cracks are formed during the drying step.

Moreover, SEM micrographs performed on the same area of a cracked pigment particle show that the crack opening displacement within dried TiO₂ layers increases during the sintering process (Fig. 7).

As shown in Table 3 and Fig. 5, the Ti-300-Al₂O₃ pigment also exhibits cracking problems. The evolution of cracks between the drying and the sintering steps has been observed for several Ti-300-Al₂O₃ pigment particles. Again, cracks at the surface of sintered pigment particles are more

visible than cracks present at the surface of dried pigments. Some cracks are widening but the crack percentage of dried and sintered samples remains similar. The same conclusion as for mica based pigments can then be drawn: the main cracks are formed during the drying step. In contrast, the case of pigments based on silica substrates is different from the other substrates. For example, the crack percentage in Ti-140-SiO₂ samples is very low. However, a very small amount of cracks can be detected in the dried sample and the evolution of these cracks during sintering can be observed using light microscopy. Cracks initiated during the drying step were either eliminated or rendered less visible after sintering.

4. Discussion

4.1. Cracking during drying

As has been pointed out in the results, cracking of TiO₂ layers can be attributed to the drying step of the pigments. The drying of TiO₂ layers is performed in this study at 110 °C. At this temperature, the thermal expansion of the substrates should be limited and the difference in thermal expansion between the film and the substrate should not cause crack formation. To confirm this hypothesis, a Ti-140-mica sample was dried at room temperature in a desiccator. Its crack percentage was found comparable to a sample dried in the oven at 110 °C. Thus, it can be concluded that the difference in thermal expansion between the TiO₂ layer and mica substrate is not governing the cracking of TiO₂ layers.

The influence of the drying step on the cracking of inorganic materials like granular ceramics or gels[8], [22], [23], [24] and [25] has been mainly linked to the presence of the liquid in the pores during the drying of the body. At the onset of drying, water evaporates from the pores at the surface of the material. This leads to the substitution of a solid/liquid interface by a solid/vapour interface, if the water is not replenished from the interior of the body. To prevent exposure of the solid phase, the meniscus of the liquid adopts a curved liquid/vapour interface. A capillary pressure (ΔP) develops across the curved interface liquid/gas and reaches its maximum value (ΔP_M) when the radius of the meniscus is small enough to fit into the pores. This leads to tension in the liquid, balanced by compressive forces in the solid phase which causes the network to shrink. The maximum value of the capillary pressure (ΔP_M), which corresponds to the maximum of the drying stress, is given by Eq. (3)8:

(3)

$$\Delta P_M = \frac{2\gamma_{LV} \cos\theta}{a}$$

Here γ_{LV} is the liquid/vapour interfacial energy, θ is the contact angle between the liquid and the solid phase, and a is the pore radius.

In Eq. (3), pores are considered cylindrical and the pore radius (a) is the radius of the pores in the drying material when ΔP_M is reached. The exact value of a for the present products was not precisely determined since the layer characterization was performed after the drying and not during the drying. Nevertheless, a can be approximated by \bar{r}_p , the average pore radius deduced from the nitrogen adsorption measurements since most of the shrinkage occurs before ΔP_M is reached.⁸ Then, for $\cos \theta \approx 1$, $\gamma_{LV} = 7 \times 10^{-3} \text{ J m}^{-2}$ (surface tension of water) and $a \approx \bar{r}_p \approx 2 \text{ nm}$, the maximal capillary pressure ΔP_M is about 70 MPa. It is possible to show with this simple estimation of ΔP_M that there is a significant capillary tension in the liquid at the critical point. The capillary pressure ΔP causes contraction of the solid phase during drying. When the solid phase is a film deposited on a rigid substrate like a gel film [24] and [25] or a granular ceramic film, [9], [10], [12] and [13] the contraction of the film imposes compressive stresses on the substrate and corresponding tensile stresses in the film. These can lead to the initiation of cracks at stress concentrators like pores.

4.2. Cracking during sintering

Sintering does not promote cracking in mica and alumina based pigments. Specifically, sintering of constrained Ti-140-mica and Ti-300- Al_2O_3 films leads to a widening of the cracks and also to a higher densification near the crack faces since there are no geometrical constraints close to the cracks.²⁶

The sintering behaviour of the SiO_2 platelets leads to a different scenario: for the Ti-140- SiO_2 pigment, the occasional cracks formed during drying are eliminated or partially healed during the sintering step. The increase in density of the amorphous SiO_2 platelets from 1.770 to 2.019 g cm^{-3} between 600 and 850 °C is attributed to a real densification. This increase in density is not correlated to the SiO_2 platelet weight loss, which occurs mainly between 25 and 600 °C. The weight loss is first due to water removal from the surface of the SiO_2 platelets (until 200 °C) and then due to the release of water resulting from the dehydroxylation of silanol groups on the surface of SiO_2 platelets. The obtained density at 850 °C (2.019 g cm^{-3}) is in agreement with values found in the literature for amorphous silica of around 2.200 g cm^{-3} .²⁷ Therefore, the densification of SiO_2 substrates leads to shrinkage of the platelets, which can lead to partial or complete crack healing.²⁸

4.3. Influence of layer thickness on the cracking of dried TiO_2 layers

Cracking behaviour is observed to depend on layer thickness: for the same substrate, cracking of TiO_2 layers increases with the thickness of the layer (Table 3 and Fig. 5).

According to linear elastic fracture mechanics, a constrained film, subject to stress, will crack when the strain energy released in the process exceeds the energy required to form the crack. [29] and [30] For a film and a substrate having the same elastic moduli, a critical thickness for film cracking

(hf) can be expressed at a given level of tensile stress (σ_x) by the following equation Eq. (4)30:

(4)

$$h_f = \frac{2 \Gamma_f E_f}{\pi \cdot c_e^2 \sigma_x^2}$$

Here c_e is a dimensionless constant (close to 1), E_f is the elastic modulus of the film and Γ_f is the energy required to form two new crack surfaces per unit area.

In our case, it is not possible to calculate precisely this critical cracking thickness since E_f and Γ_f are not known for such granular nanometric films. However, due to the nanometric character of the microstructure, the development of high drying stresses in these nanocrystalline films is very likely (see the high value of capillary pressure estimated in Section 4.1). It can account for a submicrometer critical cracking thickness as it is often observed in the case of granular films obtained through sol-gel processes ($<0.1 \mu\text{m}$). [31], [32] and [33] Experimentally, the critical cracking thicknesses of the studied films fit reasonably well with this analysis. They can be estimated to be of the order of 100 nm when TiO_2 is deposited on mica and >140 nm when deposited on alumina or silica (Table 3).

4.4. Influence of the substrate on the cracking of dried TiO_2 layers

Considering Ti-140-mica, Ti-140- Al_2O_3 and Ti-140- SiO_2 green pigments, the TiO_2 layers exhibit the same properties with respect to porosity, grain size and layer thickness but the cracking behaviour of these layers differs. Therefore, the substrates on which the layers are precipitated exert an influence on the cracking of the TiO_2 layers.

As already mentioned, the difference in thermal expansion between the different substrates is not the main cause of the difference in cracking behaviour of TiO_2 layers since cracking occurs at relatively low temperature (110 °C).

The surface charge and the surface structure of the different substrates also cannot explain the different cracking behaviour of the TiO_2 layers. In the precipitation conditions ($\text{pH} < 2$), all substrates present a relatively positive surface charge ($\text{pH} < \text{PZC}$), leading to the same TiO_2 precipitation.

Further, no correlation could be established between the substrate lattice parameters and the SnO_2 ($\equiv \text{TiO}_2$ rutile) lattice parameters, so that template effects are not expected. Besides, TiO_2 anatase layers were precipitated on different substrates, and their cracking behaviour depends on the underlying substrates in the same way as for TiO_2 rutile layers.

The roughness of the different substrates should be considered in more detail since this parameter can control the adhesion of the film on the substrate.

The values obtained for the roughness of the different substrates are, however, not conclusive. 140 nm TiO₂ layers only exhibit cracks when deposited on synthetic or natural mica, which exhibit an average roughness of either 9 nm or 1 nm, respectively. The average roughness of glass platelets (~5 nm) and SiO₂ platelets (~4 nm) is smaller than the average roughness of synthetic mica and larger than the average roughness of natural mica. Finally, the average roughness of Al₂O₃ is very small (<1 nm). However, a common feature of natural and synthetic mica is the steps present at the surface of the platelets.

A Finite Element Model (FEM)-stress simulation was performed to simulate the development of stresses in a drying layer deposited on a rigid substrate containing steps. For this model, constrained film shrinkage was simulated using the strain mismatch which occurs when cooling from an increased temperature with different thermal expansion coefficient for film and substrate. The temperature difference and the thermal expansion mismatch were chosen to match the strain incompatibility during drying. Mechanical and geometrical parameters are provided in Fig. 8. The results of the computation are provided with a focus on the effect of surface inhomogeneities on stress distribution (Fig. 9). The FEM simulation confirms that there is an increase of the stress when step-like defects are present at the surface of the substrate. The increase of the stress occurs in the vicinity of the steps and extends to a distance of the order of the step height. In some cases, cracks are experimentally observed along the steps at a distance that fits reasonably with the simulation. In other cases, no clear correlation between cracking and steps could be observed. However, this FEM simulation was quite simple and therefore too limited to simulate precisely what happens in a polycrystalline TiO₂ layer. For example, the different powder packing around steps and the attendant effect on local fracture energy cannot be avoided and could not be taken into account.

5. Conclusion

Cracking in thin nanocrystalline titania films deposited on different substrates has been shown to develop during drying. The subsequent sintering step leads to an increase of the crack opening displacement for dense substrates (mica and alumina) and to partial or complete crack healing for densifying substrates (silica). A critical thickness for cracking could be discerned for these layers, which was about 100 nm for mica substrates and higher than 140 nm for the alumina and silica substrates. Surface steps in the mica are suggested to be responsible for this difference.

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References

- 1 G. Pfaff and P. Reynders, Angle-dependent optical effects deriving from submicron structures of films and pigments, *Chem. Rev.* 99 (1999), pp. 1963–1981.
- 2 J. Maile, G. Pfaff and P. Reynders, Effect pigments-past, present and future, *Prog. Org. Coat.* 54 (2005), pp. 150–163.
- 3 J. Zarzycki, M. Prassas and J. Phalippou, Synthesis of glasses from gels: the problem of monolithic gels, *J. Mater. Sci.* 17 (1982), pp. 3371–3379.
- 4 T. Mizuno, H. Nagata and S. Manabe, Attempts to avoid cracks during drying, *J. Non-Cryst. Solids* 100 (1988), pp. 236–240.
- 5 T.J. Garino and H.K. Bowen, Deposition and sintering of particle films on a rigid substrate, *J. Am. Ceram. Soc.* 70 (1987), pp. C315–C317.
- 6 T. Cheng and R. Raj, Flaw generation during constrained sintering of metal-glass multilayer films, *J. Am. Ceram. Soc.* 72 (1989), pp. 1649–1655.
- 7 A. Jagota and C.Y. Hui, Mechanics of sintering thin films-II cracking due to self-stress, *Mech. Mater.* 9 (1990), pp. 107–109.
- 8 G.W. Scherer, Theory of drying, *J. Am. Ceram. Soc.* 73 (1990), pp. 3–14.
- 9 R.C. Chiu, T.J. Garino and M.J. Cima, Drying of granular ceramic films: I effect of processing variables on cracking behavior, *J. Am. Ceram. Soc.* 76 (1993), pp. 2257–2264.
- 10 R.C. Chiu and M.J. Cima, Drying of granular ceramic films: II drying stress and saturation uniformity, *J. Am. Ceram. Soc.* 76 (1993), pp. 2769–2777.
- 11 R.K. Bordia and A. Jagota, Crack growth and damage in constrained sintering films, *J. Am. Ceram. Soc.* 76 (1993), pp. 2475–2485.
- 12 J.S. Reed, *Principles of Ceramics Processing* (2nd ed.), Wiley-VCH, Weinheim (1995).
- 13 M.N. Rahaman, *Ceramics Processing and Sintering* (2nd ed.), Marcel Dekker, New York (2003).
- 14 G.K.L. Goh, S.K. Donthu and P.K. Pallathadka, Cracking and orientation of solution-deposited rutile TiO₂ films, *Chem. Mater.* 16 (2004), pp. 2857–2861.
- 15 M.S. Hu, M.D. Thouless and A.G. Evans, The decohesion of thin films from brittle substrates, *Acta Metall.* 36 (1988), pp. 1301–1307.
- 16 F. Zhao, B. Wang, X. Cui, N. Pan, H. Wang and J.G. Hou, Buckle delamination of textured TiO₂ thin films on mica, *Thin Solid Films* 489 (2005), pp. 221–228.
- 17 Bauer, G., Osterried, K., Schmidt, C., Vogt, R., Kniess, H. B., Uhlig, M. et al., Coloured and

coated platelike pigments. WO Patent 93/08237, 1991.

18 J.P. Jolivet, *Metal Oxide Chemistry and Synthesis*, Wiley-VCH, Weinheim (2000).

19 Y. Kuwahara, Muscovite surface structure imaged by fluid contact mode AFM, *Phys. Chem. Miner.* 26 (1999), pp. 198–205.

20 S. Teaney, G. Pfaff and K. Nitta, New effect pigments using innovative substrates Silica and alumina platelets extend the product variety and performance range of pearlescent pigments, *Eur. Coat. J.* 4 (1999), pp. 90–96.

21 G.V. Franks and L. Meagher, The isoelectric points of sapphire crystals and alpha-alumina powder, *Colloids Surf. A* 214 (2003), pp. 99–100.

22 D.M. Smith, G.W. Scherer and J.M. Anderson, Shrinkage during drying of silica gel, *J. Non-Cryst. Solids* 188 (1995), pp. 191–206.

23 G.W. Scherer, S. Hæreid, E. Nilsen and M.-A. Einarsrud, Shrinkage of silica gels aged in TEOS, *J. Non-Cryst. Solids* 202 (1996), pp. 42–52.

24 G.W. Scherer, Drying gels II film and flat plate, *J. Non-Cryst. Solids* 89 (1987), pp. 217–238.

25 G.W. Scherer, Drying gels VIII revision and review, *J. Non-Cryst.* 109 (1989), pp. 171–182.

26 M. Stech, P. Reynders and J. Rödel, Constrained film sintering of nanocrystalline TiO₂, *J. Am. Ceram. Soc.* 83 (2000), pp. 1889–1896.

27 A.P. Legrand, *The Surface Properties of Silicas*, John Wiley & Sons, New York (1998).

28 J. Rödel and A.M. Glaeser, High-temperature healing of lithographically introduced cracks in sapphire, *J. Am. Ceram. Soc.* 73 (1990), pp. 592–601.

29 J.M. Heintz, O. Sudre and F.F. Lange, Instability of polycrystalline bridges that span cracks in powder films densified on a substrate, *J. Am. Ceram. Soc.* 77 (1994), pp. 787–791.

30 F.F. Lange, Chemical solution routes to single-crystal thin films, *Science* 273 (1996), pp. 903–909.

31 H. Kozuka and M. Kajimura, Single-step dip coating of crack-free BaTiO₃ films > 1 μm thick: effect of poly(vinylpyrrolidone) on critical thickness, *J. Am. Ceram. Soc.* 83 (2000), pp. 1056–1062.

32 L.B. Freund and S. Suresh, *Thin Film Materials. Stress Defect Formation and Surface Evolution*, Cambridge University Press, Cambridge (2003).

33 M. Bockmeyer and P. Löbmann, Crack formation in TiO₂ films prepared by sol-gel processing: quantification and characterization, *Thin Solid Films* 515 (2007), pp. 5212–5219.

Figures

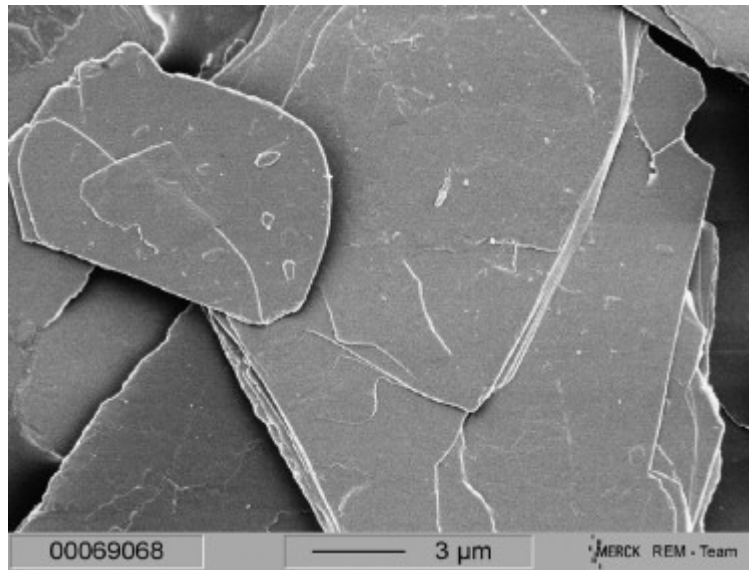


Fig. 1. SEM micrograph of mica substrates used in the LDP process.

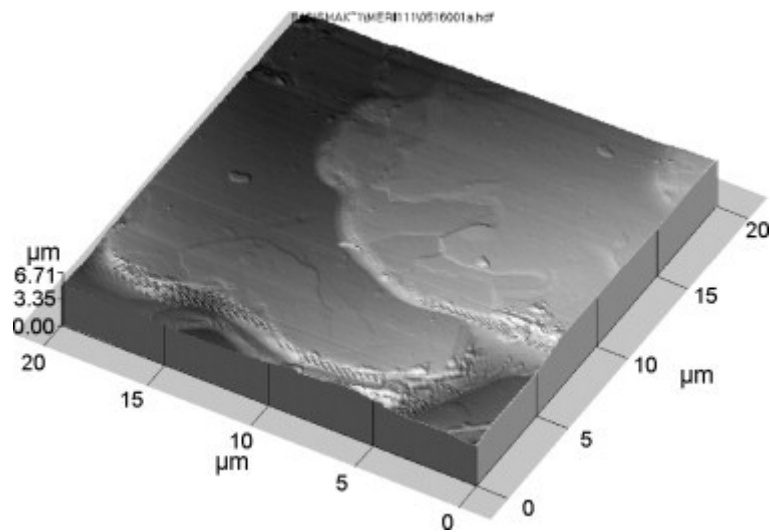


Fig. 2. AFM micrograph showing steps at the surface of natural mica.

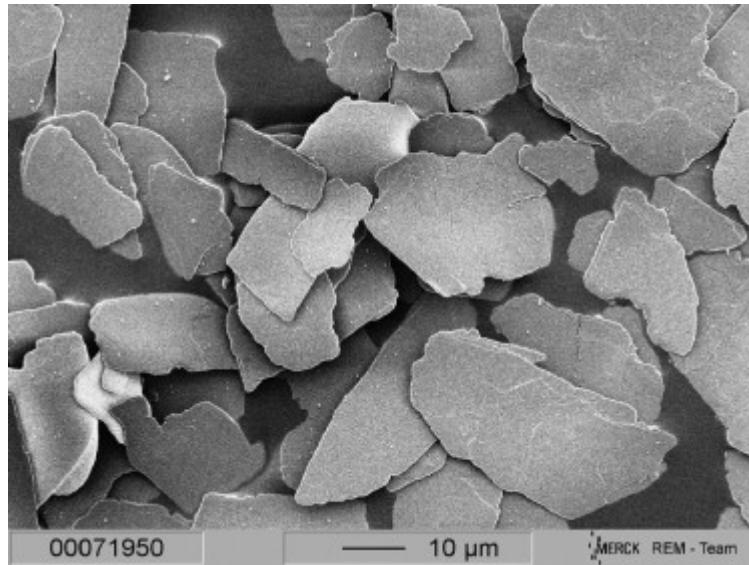


Fig. 3. SEM micrograph of Ti-140-mica pigment particles.

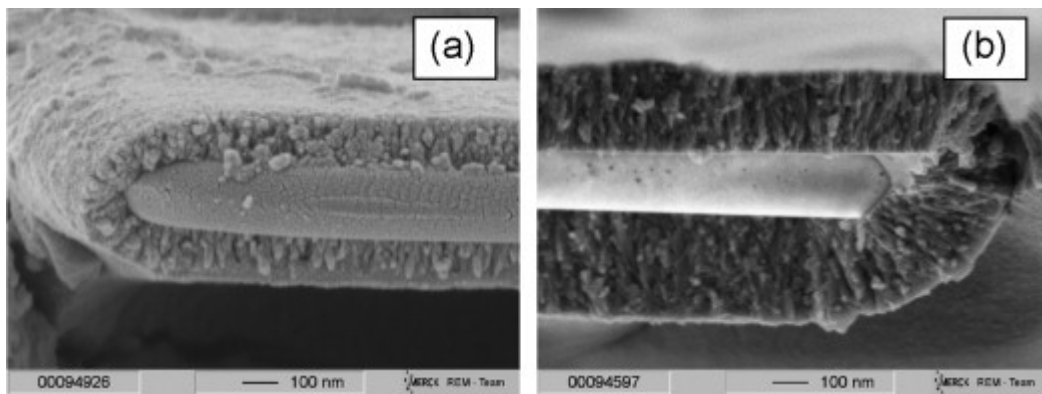


Fig. 4. SEM cross-section of dried pigment particles: (a) Ti-140-Al₂O₃; (b) Ti-300-Al₂O₃.

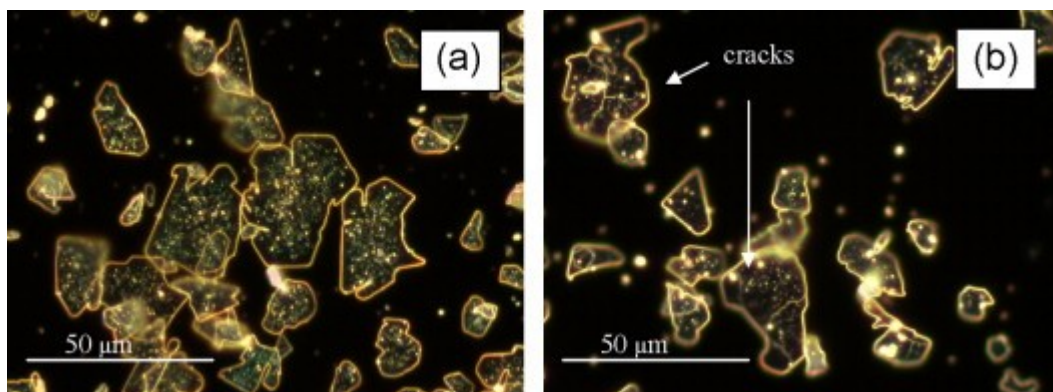


Fig. 5. Light microscopy images of (a) Ti-140-Al₂O₃ and (b) Ti-300-Al₂O₃ pigment particles in their dried state: no cracks were found in Ti-140-Al₂O₃ whereas cracks can be detected in Ti-300-Al₂O₃.

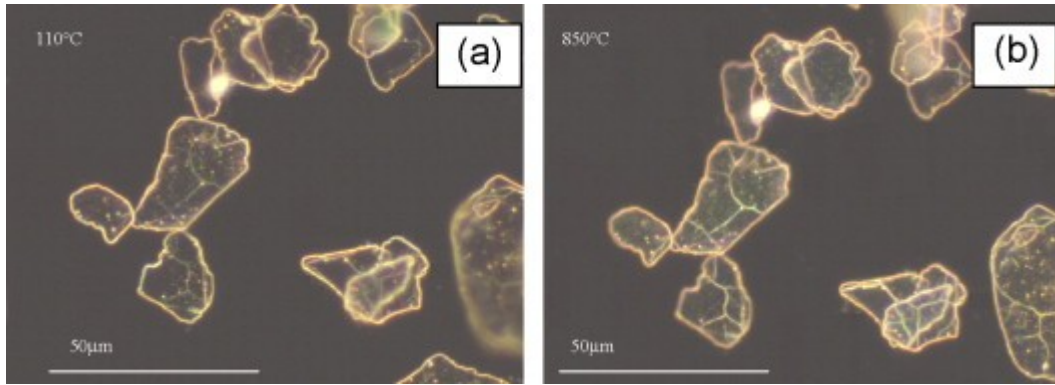


Fig. 6. Light microscopy images of Ti-140-mica pigment particles (a) before and (b) after sintering at 850 °C.

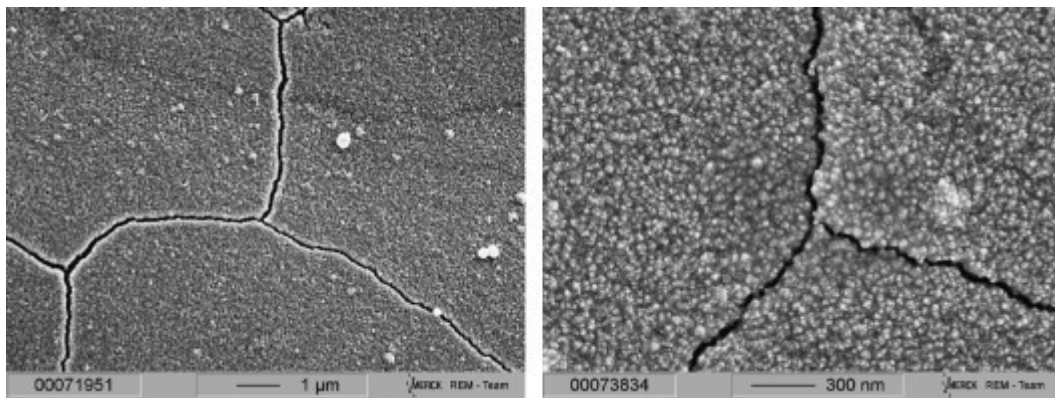


Fig. 7. SEM micrograph of a Ti-140-mica pigment particle (left) before and (right) after sintering at 850 °C.

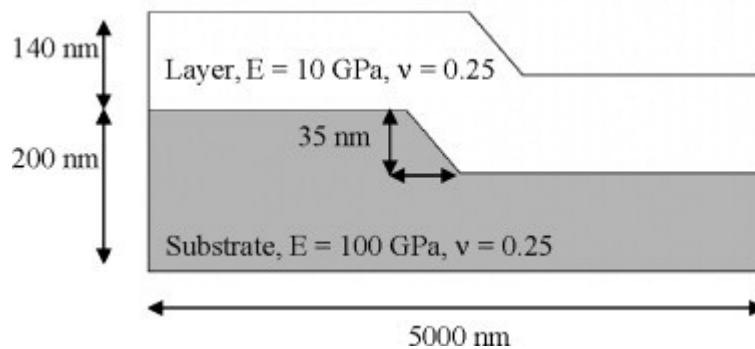


Fig. 8. Mechanical and geometrical parameters used in the FEM analysis ($\Delta T = 100 \text{ K}$, $\alpha_{\text{substrate}} = 9 \times 10^{-6} \text{ K}^{-1}$, $\alpha_{\text{layer}} = 0.976 \times 10^{-6} \text{ K}^{-1}$).

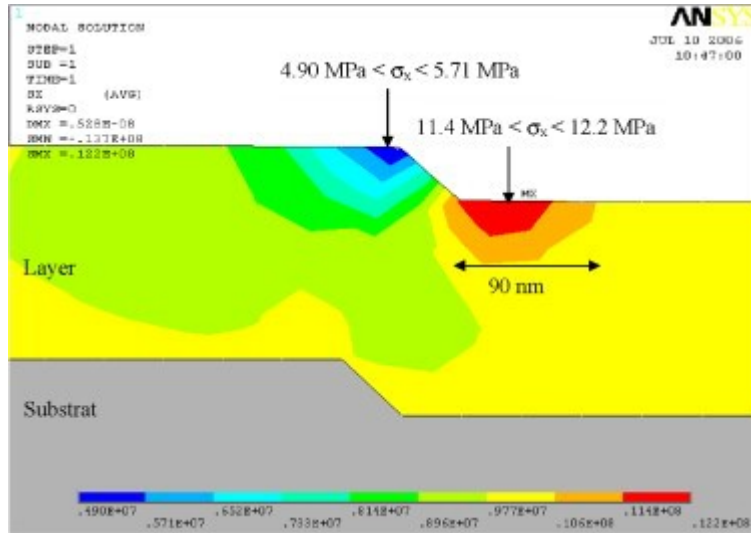


Fig. 9. FEM analysis: contour plot representing the stress distribution in a drying layer deposited on a substrate in the vicinity of a step-like defect.

Tables

Table 1.

Volume shrinkage of SiO₂ platelets deduced from pycnometry analyses

	SiO ₂ platelets 110 °C	SiO ₂ platelets 600 °C	SiO ₂ platelets 850 °C
Calculated density (g/cm ³)	1.736	1.770	2.019
Volume shrinkage	Reference	1.9%	14%

Table 2.

Properties of dried TiO₂ layers

	\bar{r}_p (nm)	G_{BET} (nm)	$D_{\text{XRD}(1\ 0\ 1)}$ (nm)
Ti-60-mica	2.2	10	–
Ti-140-mica	2.0	18	24
Ti-140-Al ₂ O ₃	2.6	26	21
Ti-140-SiO ₂	2.1	18	22
Ti-300-Al ₂ O ₃	1.6	27	29
Ti-300-SiO ₂	2.1	43	31

Table 3.

Crack percentage in the different studied pigments (percentage of pigment particles exhibiting cracks in a sample)

Samples and number of samples studied	Crack percentage
Ti-60-mica (3 samples)	Low (1–5%)
Ti-140-mica (16 samples)	High (30–40%)
Ti-140-Al ₂ O ₃ (5 samples)	Very low (1%)
Ti-140-SiO ₂ (5 samples)	Very low (1%)
Ti-300-Al ₂ O ₃ (1 sample)	High (20–30%)
Ti-300-SiO ₂ (1 sample)	Moderate (~10%) but some delamination