



HAL
open science

Structural phase transition in ZrO₂ induced by swift heavy ion irradiation at high pressure

B. Schuster, M. Lang, R. Klein, C. Trautmann, R. Neumann, A. Benyagoub,
F. Fujara

► **To cite this version:**

B. Schuster, M. Lang, R. Klein, C. Trautmann, R. Neumann, et al.. Structural phase transition in ZrO₂ induced by swift heavy ion irradiation at high pressure. The Seventh International Symposium on Swift Heavy Ions in Matter, Jun 2008, Lyon, France. hal-00251589

HAL Id: hal-00251589

<https://hal.science/hal-00251589>

Submitted on 11 Dec 2008

HAL is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers.

L'archive ouverte pluridisciplinaire **HAL**, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d'enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.

Structural phase transition in ZrO_2 induced by swift heavy ion irradiation at high pressure

B. Schuster^{*a,b}, M. Lang^c, R. Klein^a, C. Trautmann^a, R.
Neumann^a, and A. Benyagoub^d

^aGesellschaft für Schwerionenforschung (GSI), Planckstr.1, 64291 Darmstadt, Germany

^bTechnische Universität Darmstadt, Hochschulstrasse 1, Germany

^cUniversity of Michigan, 1100 N. University Avenue, 48109 Ann Arbor, USA

^dCIMAP, rue Claude Bloch, BP 5133, F-14070 Caen Cedex, France

Abstract

Exposing pressurized crystals to GeV heavy ions reveals unexpected structural changes. Irradiated at ambient conditions, natural zirconia (ZrO_2) transforms from the monoclinic structure to its tetragonal (high-temperature) phase. For this process the required fluence must exceed 5×10^{12} ions/cm² for Pb and U, and become even significantly higher for lighter ions. If samples are pressurized during irradiation using diamond anvil cells, the required fluence drops at least by one order of magnitude. The efficiency of the monoclinic to tetragonal phase transition becomes larger with increasing pressure.

Keywords: Zirconia, phase transition, high pressure, irradiation effect, Raman spectroscopy

*Corresponding author. Tel.: +49-6159-71-2033

E-Mail: bea.schuster@gsi.de

1 Introduction

The initial motivation in combining high-pressure and irradiation with swift heavy ions has its origin in geophysics, where fission tracks are analyzed for dating purposes [1][2]. Also in materials science, the question how solids respond to the simultaneous exposure to pressure and extreme energy input has raised large interest [3]. The creation of new phases otherwise not accessible may become possible for manufacturing novel substances with particular characteristics. In this study, we are concentrating on zirconia, which is known for its thermal and mechanical resistance and its radiation hardness. Because of these properties, ZrO_2 is often used as inert fuel matrix in nuclear reactors [4] [5] or as a container for radioactive e.g. plutonium waste [6].

At ambient temperature and pressure conditions, ZrO_2 is a white powder with a monoclinic structure. ZrO_2 has two known high-pressure polymorphs (orthorhombic I and orthorhombic II [7]) and two different high-temperature polymorphs (tetragonal and cubic) (see Fig 1).

At about 1200°C , a martensitic-type phase transition occurs to a tetragonal derivative of the fluorite structure, and at around 2400°C another structural phase transition leads to a true cubic fluorite structure that is stable up to the melting temperature of $\sim 2700^\circ\text{C}$ [8]. These high-temperature phases cannot be quenched, but they can be stabilized at room temperature by replacing up to 15% of Zr with larger cations (e.g. La, Mg, Y) [9]. Stoichiometric as well as stabilized zirconia has previously been irradiated with a wide spectrum of energetic particles in the nuclear [10][11][12][13] and electronic stopping regime [8][14][15]. Both compound shows a remarkable resistance against amorphization if the grain size is not on the nm scale [16]. The polymorphism of zirconia plays an important role for the evolution of the radiation damage. ZrO_2 is one of the few materials that undergoes, instead of amorphization, a crystalline-crystalline (displacive) phase transition to the higher symmetry phase, i.e. monoclinic ZrO_2 transforms either to the tetragonal or the cubic structure. Once this has occurred, the new crystalline phase persists without further transitions as observed e.g.

with 800-keV Bi ions dominated by nuclear stopping [12] and with various swift heavy ions for which the electronic stopping exceeded 13 keV/nm [17][8][18][19]. However, even at high fluences, the transformation into the tetragonal polymorph is not complete but saturates below 100% [17]. In contrast to tetragonal ZrO₂ created by heating, the radiation induced tetragonal phase does not return to the monoclinic form. Including pressure as an additional parameter, new aspects of the structural transformations can be investigated.

2 Experimental

For pressurization of samples, Diamond Anvil Cells (DAC) of the Merrill-Bassett type [20] were used. Several grains (size $\sim 0.2 \mu\text{m}$) of monoclinic ZrO₂ powder (from Goodfellow company) were inserted in the central aperture (diameter of $150 \mu\text{m}$) of a stainless steel gasket mounted between two diamonds each with a culet face with a diameter of $600 \mu\text{m}$ and a thickness of 2.64 mm. In addition, the sample chamber was filled with a 16:3:1 mixture of methanol:ethanol:water as optically transparent pressure medium to insure hydrostatic conditions [21]. Pressure was applied by squeezing the gasket by two diamonds up to a pressure between 4.3 and 13.4 GPa. The pressure was measured by adding a small ruby crystal into the sample chamber and measuring the wavelength of the red fluorescence line [22]. The pressurized ZrO₂ samples embedded in the DAC were exposed to relativistic ions from the SIS heavy-ion synchrotron of GSI (Darmstadt). The experiments were performed at room temperature using either ²³⁸U or ²⁰⁸Pb ions with kinetic energies of about 200 MeV per nucleon (MeV/u) (see Table 1).

The energy is high enough so that the ions traverse the first diamond and the entire sample (Fig 2). Moreover, the energy loss inside the sample is close to the Bragg maximum. Details of the irradiation parameters are summarized in Table 1. The irradiation experiment at the SIS was performed with a pulsed beam of 0.3 spills/s (extraction time of each spill ~ 1.4 seconds) containing

about $\sim 2\text{-}5 \times 10^8$ ions/cm² per spill. The flux was monitored via a calibrated secondary-electron emitting detector (SEETRAM) [23]. The fluence applied ranged between 1 and 2.5×10^{11} ions/cm². Fluence cross-checks were performed by counting tracks in an irradiated mica sample inspected by atomic force microscopy [24]. As reference, an additional ZrO₂ sample was irradiated behind a diamond applying identical conditions, but without external pressure. At the SIS, the beamtime and thus possible maximum fluences are limited therefore we performed additional reference irradiations to test radiation effects at high fluences (fluence series of 5×10^{10} to 5×10^{12} ions/cm²) with ²³⁸U and ²⁰⁸Pb ions at the UNILAC (Universal Linear Accelerator) of GSI. Because the maximum available energy at the UNLIAC is only 11.1 MeV/u (corresponding to a range of ~ 60 μ m for U-ions), the ZrO₂ samples were irradiated without using the DAC. The energy loss was 42 keV/nm for ²⁰⁸Pb and 53 keV/nm for ²³⁸U, respectively. Both values are well above the threshold of 13 keV/nm necessary for a transformation from the monoclinic to the tetragonal phase [8]. After irradiation, the samples were analyzed by confocal Raman spectroscopy using a HR800-180° system (Horiba Jobin Yvon) with a 20 mW He-Ne-Laser and an excitation wavelength of 632.8 nm.

3 Results and Discussion

The Raman spectra of ZrO₂ before and after irradiation with ²⁰⁸Pb ions at the UNILAC are displayed in Fig. 3. Up to a fluence of 1×10^{12} ions/cm² for Pb as well as U, the Raman spectra hardly change and predominantly show the bands typical of monoclinic ZrO₂ at 180, 190, 334, 348, 380, 476, 617, and 638 cm⁻¹ [13][25]. At fluences around 2×10^{12} ions/cm², small modifications become visible. At 5×10^{12} ions/cm² and above, the spectra exhibit new broad bands located at 148, 260, 465, and 640 cm⁻¹ which are assigned to the tetragonal phase of ZrO₂ [8][25][26]. For both ion species, this transformation occurs in a rather small fluence regime between 2×10^{12} and 5×10^{12} ions/cm². Further

fluence increase to 7.5×10^{12} Pb-ions/cm² does not induce additional changes. This effect is similar to the observations made earlier by Benyagoub for Ni (135 MeV) and Kr (300 MeV) ions [27], but saturation for these lighter ions requires fluences higher than 2.5×10^{13} ions/cm². Also in agreement with [27], Pb- or U-irradiated ZrO₂ does not fully transform into its tetragonal phase but still exhibits some Raman bands of the monoclinic phase (e.g. at 180 and 190 cm⁻¹.)

The samples irradiated at high pressure show quite different results. In a first approach, Raman spectra of irradiated ZrO₂ samples still pressurized in the DAC to 10.9 GPa were recorded, but the quality of the spectra is quite poor, because the Raman light has to pass the diamond twice before reaching the detector. Therefore, the intensity of the signal is extremely weak, and the background increases enormously so that recording a single spectrum may take several hours. In the few spectra collected, no significant change could be observed, which was not too surprising given the low fluences available for the DAC irradiation. Another complication for a comparative study resulted from the fact that at a pressure of 10.9 GPa, the ZrO₂ sample is no longer monoclinic but has transformed to its high-pressure orthorhombic I phase.

The Raman investigations concentrated on samples removed from the DAC, i.e. at ambient conditions. With the mapping function, several areas of the sample were examined using the microscope objective with the largest magnification (100x) with a spatial resolution of about 1 μm.

Surprisingly, the Raman spectra varied strongly with the position where the spectrum was acquired. Fig. 4 shows four representative Raman spectra of the sample irradiated with 2.5×10^{11} Pb-ions/cm² at 10.9 GPa. Spectrum (a) resembles that of an unirradiated monoclinic ZrO₂ sample and was found at about 50% of the tested sample areas. Spectrum (b) is dominated by deformed monoclinic ZrO₂ bands with a small trace of tetragonal ZrO₂ indicated by bands at 148 and 260 cm⁻¹. Spectrum (c) shows a mixture of monoclinic and tetragonal ZrO₂ while spectrum (d) resembles that of almost pure tetragonal ZrO₂. A nearly complete transformation appeared at about 7% of the tested sample

areas while all other spectra showed different mixtures of monoclinic and tetragonal ZrO_2 . Comparing the relative intensities of the monoclinic bands at 180 and 190 cm^{-1} and the tetragonal band at 148 cm^{-1} , the amount of monoclinic ZrO_2 (X_m) in the two-phase mixture can be estimated from the intensities of the monoclinic (I_m) and tetragonal (I_t) bands [25].

$$X_m = \frac{I_m(180 \text{ cm}^{-1}) + I_m(190 \text{ cm}^{-1})}{I_m(180 \text{ cm}^{-1}) + I_m(190 \text{ cm}^{-1}) + I_t(148 \text{ cm}^{-1})} \quad (1)$$

Eq. (1) yields $X_m = 99\%$ for spectrum 4a, $X_m = 97\%$ for 4b, $X_m = 70\%$ for 4c, and $X_m = 13\%$ for 4d. This significantly exceeds the tetragonal content of the sample irradiated with 7.5×10^{12} Pb-ions/ cm^2 at the UNILAC where the amount of monoclinic ZrO_2 did not surpass 28%.

To exclude that pressure alone could have caused this strong transformation, a reference sample was pressurized to 11 GPa remaining in the DAC for two weeks. A second reference sample which was irradiated through a single diamond without being pressurized was also analyzed. The Raman spectra of both reference samples showed no deviation from a virgin sample (see Fig 5).

To determine the effect of pressure, ZrO_2 samples were irradiated at various pressures (ambient, 4.3, 7.6, 11.1, and 13.6 GPa) while keeping energy and fluence (1.5×10^{11} U-ions/ cm^2) constant (see Table 1). As stated before, the transformation of the sample is not homogeneously distributed, and thus it is difficult to directly compare the Raman spectra of the different samples. Therefore, for each sample the spectrum that showed the largest amount of tetragonal phase was selected (see Fig. 6). Up to 7.6 GPa, no strong transformation of ZrO_2 can be observed, but only the merging of the two lines at 305 cm^{-1} and the appearing band at 260 cm^{-1} . The latter belongs to the tetragonal phase and indicates the beginning of the phase change. In the sample irradiated at 11.1 GPa, areas that underwent a pronounced transformation to the tetragonal phase were found. For a pressure of 13.6 GPa, the transformation is slightly stronger than for 11.1 GPa. The originally prominent bands of the monoclinic phase at 180 and 190 cm^{-1} vanished almost completely. The degree of transformation is even more advanced than for the irradiation with 5×10^{12} U-ions/ cm^2 at the

UNILAC, where the electronic energy loss was about 40% higher compared to the irradiation at the SIS. The pressure applied during the irradiation obviously plays an important role in the mechanism of the monoclinic-tetragonal transformation since at higher pressures the tetragonal phase reaches a higher fraction. Reference samples showed that neither pressure nor irradiation with 1.5×10^{11} U-ions/cm⁻² alone could induce this degree of transformation, and therefore the effect must be linked to the combination of both.

Several authors tried to explain the mechanism of the monoclinic-tetragonal transformation induced by high-energetic as well as low-energetic heavy ions [28][29][12][15]. Investigation of the development of the tetragonal phase as a function of fluence showed a sigmoidal curve with saturation at about 85% tetragonal ZrO₂ [14][15][12]. This sigmoidal behavior suggests a mechanism for the phase transition which needs two ion impacts [30][8][12]. The first ion is assumed to create defects, mainly oxygen vacancies, that generate an important strain field in the neighborhood in the monoclinic phase of ZrO₂ and might be responsible for lowering the transition temperature [27][12]. These local strain fields are necessary to trigger the phase transformation by the second ion impact.

This might be an explanation why a pressurized sample displays this transformation already at fluences more than one magnitude lower than needed at ambient pressure conditions. A sufficiently high pressure induces a strain that otherwise has to be created by the first ion impact. In this case, the first ion takes over the part of the second ion in triggering the phase transition, and a double impact is no longer necessary. This could also explain why the transformation appears only locally within the sample. At 1.5×10^{11} ions/cm², about 10% of the sample area has been covered by ion hits assuming a track diameter of about 10 nm. It should be noted, that Raman Spectroscopy allows us to detect this displacive phase transition only if the area reaches an extension of several μm^2 .

4 Conclusion

This work presents first results of effects in ZrO_2 irradiated at various pressures with relativistic Pb and U ions. We demonstrated that the phase transition from monoclinic to tetragonal ZrO_2 , which is known to occur under high-fluence irradiation [14][27][12], begins at fluences more than one magnitude lower if the pressure exceeds 11 GPa. Reference experiments showed that neither pressure nor irradiation alone can provoke such a strong transformation. Obviously the combination of both parameters plays a crucial role in the transformation mechanism. The pressure must exceed a certain value in order to initiate the transformation and further pressure increase enhances this process. The phase transformation is not homogeneous throughout the sample volume. In contrast to irradiation at ambient conditions it appears only locally and the degree of the transformation varies strongly depending on the measured position. This may indicate that under pressure the structural transition does not require a double hit process, because pressure takes over the role of the first ion by inducing a strain field within the sample.

Acknowledgement: M. Lang gratefully acknowledges financial support from the German Science Foundation (DFG).

References

- [1] G.A. Wagner and P. van den Haute. *Fission-Track Dating*, volume 6 of *Solid Earth Sciences Library*. Kluwer Academic Publishers, 1992.
- [2] R.L. Fleischer and P.B. Price. *Journal of Geophysical Research*, 69:331–339, 1964.
- [3] U.A. Glasmacher, M. Lang, H. Keppler, F. Langenhorst, R. Neumann, D. Schardt, Ch. Trautmann, and G.A. Wagner. *Physical Review Letters*, 96(195701), 2006.
- [4] V.M. Oversby, C.C. McPheeters, C. Degueldre, and J.M. Paratte. *Journal of Nuclear Materials*, 245:17–26, 1997.
- [5] C. Degueldre. *Journal of Nuclear Material*, 320:96–105, 2003.
- [6] W.L. Gong, W. Lutze, and R.C. Ewing. *Journal of Nuclear Materials*, 278:73–84, 2000.

- [7] O. Ohtaka, H. Fukui, T. Kunisada, and T. Fujisawa. *Physical Review B*, 63(174108), 2001.
- [8] A. Benyagoub. *Physical Review B*, 72(094114), 2005.
- [9] A. Meldrum, L. A. Boatner, and R. C. Ewing. *Nuclear Instruments and Methods B*, 207(1):28–35, 2003.
- [10] K.E. Sickafus, H. Matzke, K. Yasuda, J.A. Valdez, P. Chodak III, M. Nastasi, and R.A. Verrall. *Journal of Nuclear Materials*, 274:66–77, 1999.
- [11] M. Tang, J.A. Valdez, and K.E. Sickafus. *Applied Physics Letters*, 90, 2007.
- [12] D. Simeone, G., D. Gosset, and S. Le Caër. *Nuclear Instruments and Methods B*, 250:95–100, 2006.
- [13] J. A. Valdez, M. Tang, Z. Chi, M. I. Peters, and K. E. Sickafus. *Nuclear Instruments and Methods B*, 218:103–110, 2004.
- [14] A. Benyagoub, F. Couvreur, S. Bouffart, et al. *Nuclear Instruments and Methods B*, 175-177:417–421, 2001.
- [15] G. Baldinozzi, D. Simeone, D. Gosset, I. Monnet, S. Le Caër, and L. Mazerolles. *Physical Review B*, 74(132107), 2006.
- [16] A. Meldrum, L.A. Boatner, and R.C. Ewing. *Physical Review Letters*, 88(025503), 2002.
- [17] A. Benyagoub, F. Levesque, F. Couvreur, C. Gibert-Mougel, C. Dufour, and E. Paumier. *Applied Physics Letters*, 77(20), 2000.
- [18] A. Benyagoub. *The European Physical Journal B*, 34:395–398, 2003.
- [19] C. Gibert-Mougel, F. Couvreur, J.M. Constantini, S. Bouffard, et al. *Journal of Nuclear Materials*, 295:121–125, 2001.
- [20] L. Merrill and W.A. Bassett. *Rev. Sci. Instruments*, 45:290–294, 1974.
- [21] R. Miletich, D.R. Allan, and W.F. Kuhs. In R.M. Hazan and R.T. Downs, editors, *High-Temperatur and High-Pressure Crystal Cemistry*, volume 41. Mineralogical Society of America, 2000.
- [22] H.K. Mao, J. Xu, and P.M. Bell. *Journal of Geophysical Research*, 91:4673–4676, 1986.
- [23] B. Jurado, K.-H. Schmidt, and K.-H. Behr. *Nuclear Instruments and Methods A*, 483:603–610, 2002.

- [24] J. Ackermann, N. Angert, R. Neumann, C. Trautmann, M. Dischner, T. Hagen, and M. Sedlacek. *Nuclear Instruments and Methods B*, 107:181–184, 1996.
- [25] B. Kim, J. Hahn, and K. R. Han. *Journal of Materials Science Letters*, 16, 1997.
- [26] J.-M. Costantini, A. Kahn-Harari, F. Beuneu, and F. Couvreur. *Journal of Applied Physics*, 99(123501), 2006.
- [27] A. Benyagoub. *Nuclear Instruments and Methods B*, 218:451–456, 2004.
- [28] A. Benyagoub. *Nuclear Instruments and Methods B*, 206:132–138, 2003.
- [29] A. Benyagoub. *Nuclear Instruments and Methods B*, 245:225–230, 2006.
- [30] W.J. Weber. *Nuclear Instruments and Methods B*, 166-167:98–106, 2000.
- [31] J.F. Ziegler and J.P. Biersack. *The stopping and the Range of Ions in Matter SRIM-2006*. <http://www.srim.org>, 2006.

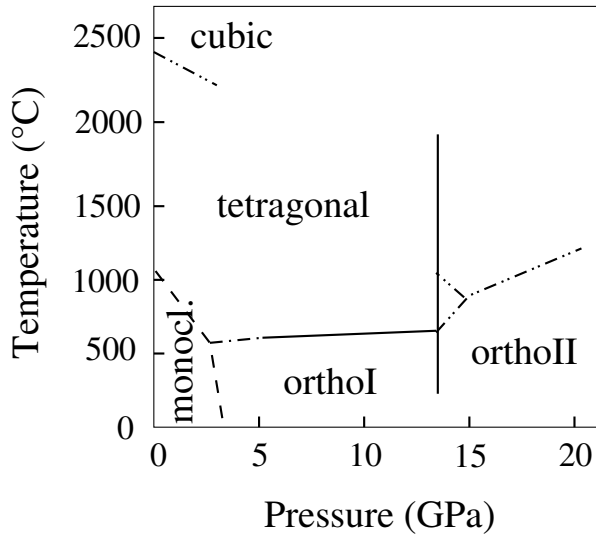


Figure 1: Phase diagram of ZrO_2 [7]

Sample	Ion	Pressure [GPa]	Fluence [ions/cm ²]	E_i [GeV]	E_f [GeV]	dE/dx [keV/nm]
SIS beam						
ZrO ₂ 01	²⁰⁸ Pb	10.9	2.5×10^{11}	204	35	30
ZrO ₂ R1	²⁰⁸ Pb	ambient	2.5×10^{11}	204	35	30
ZrO ₂ 02	²³⁸ U	4.3	1.5×10^{11}	228	39	37
ZrO ₂ 03	²³⁸ U	7.6	1.5×10^{11}	228	39	37
ZrO ₂ 04	²³⁸ U	11.0	1.5×10^{11}	228	39	37
ZrO ₂ 05	²³⁸ U	13.6	1.5×10^{11}	228	39	37
ZrO ₂ R2	²³⁸ U	ambient	1.5×10^{11}	228	39	37
ZrO ₂ 06	²³⁸ U	11.2	1.5×10^{11}	228	39	37
UNILAC beam						
ZrO ₂	²³⁸ U	ambient	$5 \times 10^{10} - 5 \times 10^{12}$		2.64	53
ZrO ₂	²⁰⁸ Pb	ambient	$5 \times 10^{10} - 7.5 \times 10^{12}$		2.31	42

Table 1: Parameters of irradiation experiments. E_i denotes the initial kinetic ion energy and E_f the ion energy when reaching the sample. dE/dx is the mean electronic energy loss deposited in the sample. The values were calculated with the SRIM 2006 code [31] taking into account the different in-beam components (detector foils, exit window, air gap, and anvil).

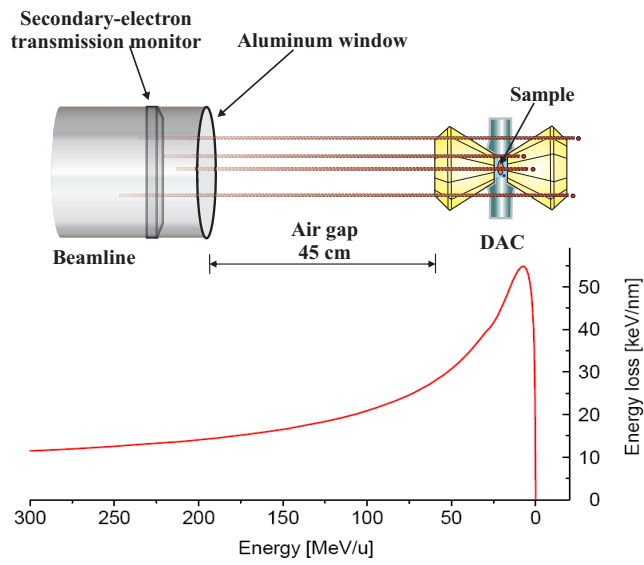


Figure 2: Schematics of sample irradiation at the heavy-ion synchrotron SIS (not to scale) [3]. Before reaching the sample, the beam exits the beamline through an Al-window and travels through 45 cm air and 2.6 mm diamond. The energy, energy loss, and ion range is estimated by calculation with the SRIM06 code [31].

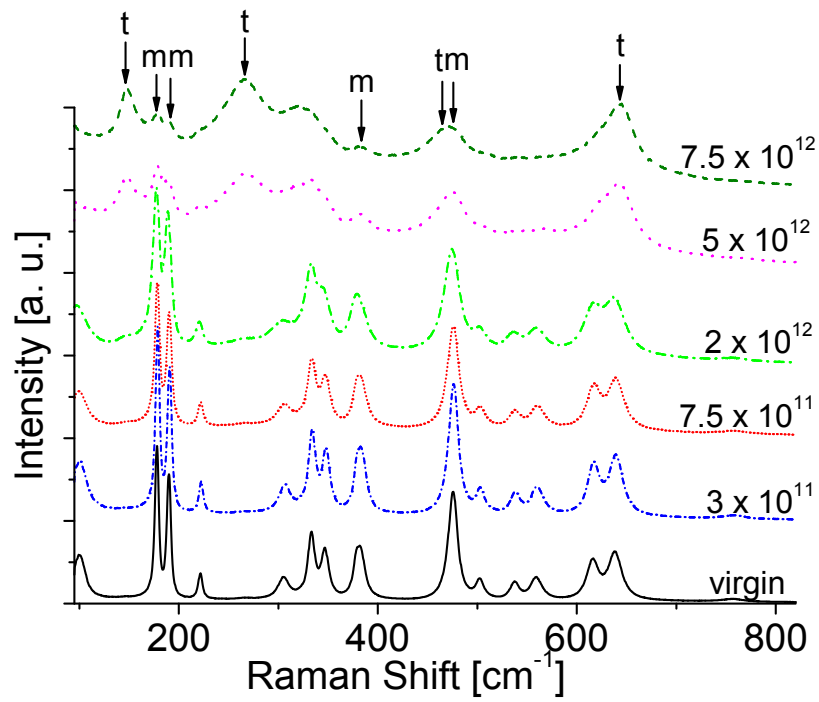


Figure 3: Raman spectra of ZrO_2 irradiated with Pb-ions (2.3 GeV) of various fluences ions/cm²

. Bands labeled as t (m) are ascribed to the tetragonal (monoclinic) phase.

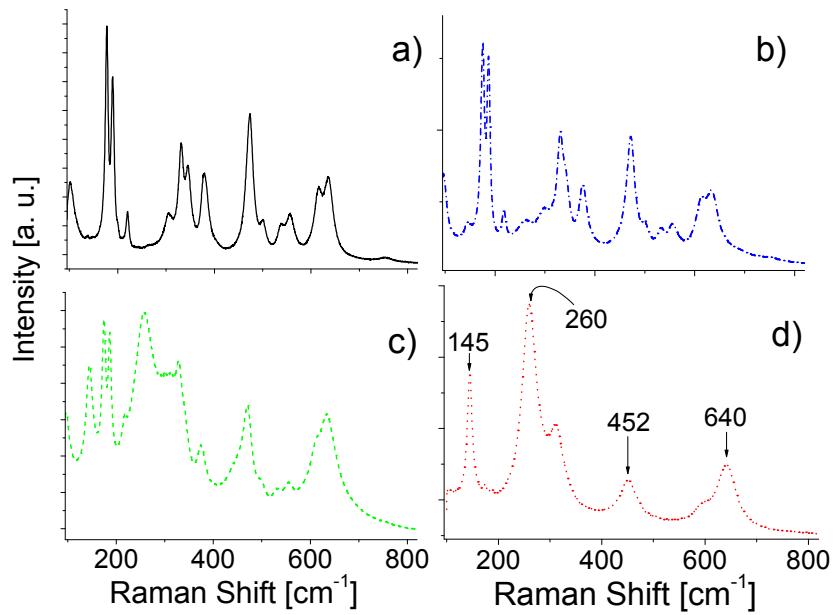


Figure 4: Raman spectra recorded outside the DAC at different positions of sample ZrO₂-01 (see Table 1) irradiated at 10.9 GPa with 2.5×10^{11} Pb-ions/cm².

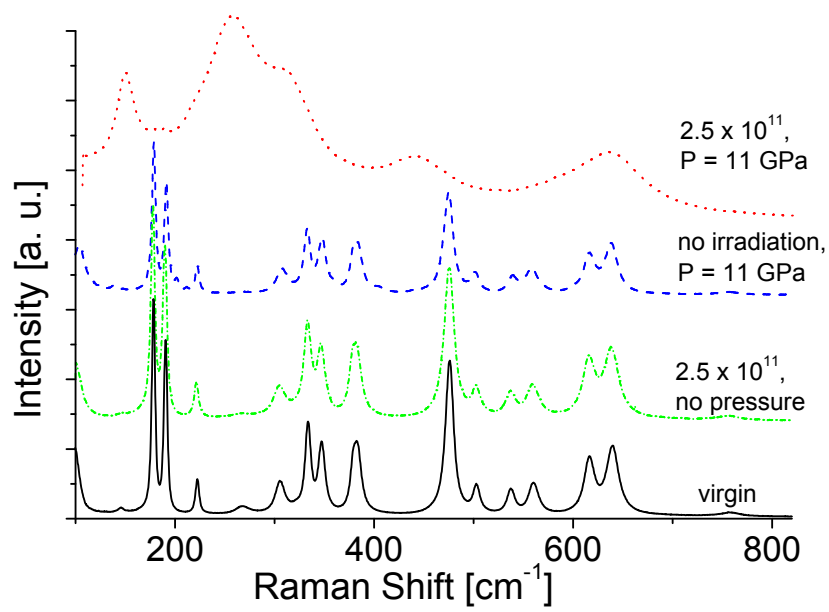


Figure 5: Raman spectrum of sample ZrO₂-01 (dotted line, irradiated under pressure) compared to reference samples only exposed to either irradiation (dash-dotted) or pressure (dashed) and virgin ZrO₂.

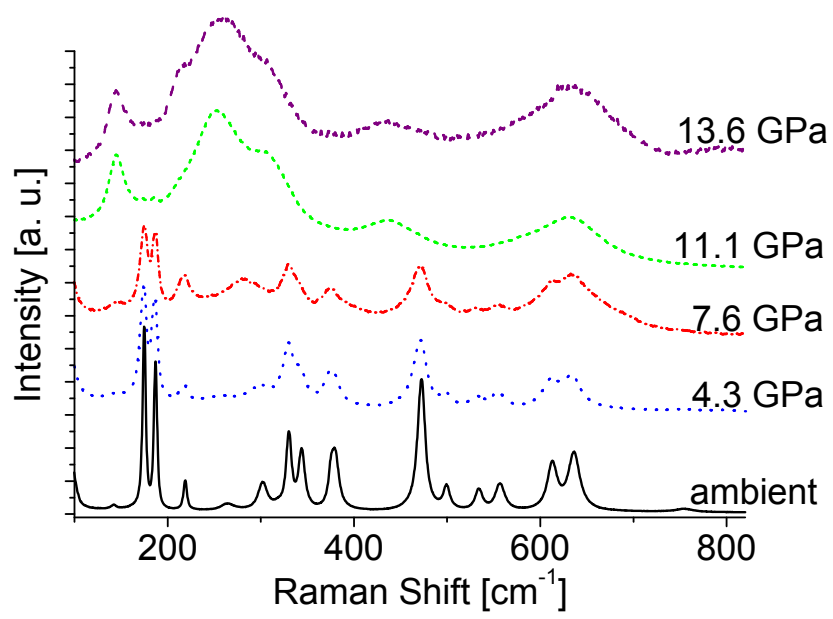


Figure 6: Raman spectra recorded outside the DAC of samples ZrO₂ 02 - 05 and R2 irradiated with 1.5×10^{11} U-ions/cm² at various pressures.