



HAL
open science

Theoretical study of the recombination of Frenkel pairs in irradiated silicon carbide

Guillaume Lucas, Laurent Pizzagalli

► **To cite this version:**

Guillaume Lucas, Laurent Pizzagalli. Theoretical study of the recombination of Frenkel pairs in irradiated silicon carbide. *Journal of Physics: Condensed Matter*, 2007, 19, pp.086208. 10.1088/0953-8984/19/8/086208 . hal-00170794

HAL Id: hal-00170794

<https://hal.science/hal-00170794>

Submitted on 10 Sep 2007

HAL is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers.

L'archive ouverte pluridisciplinaire **HAL**, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d'enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.

Theoretical study of the recombination of Frenkel pairs in irradiated silicon carbide

Guillaume Lucas and Laurent Pizzagalli

E-mail: Laurent.Pizzagalli@univ-poitiers.fr

Laboratoire de Métallurgie Physique, CNRS UMR 6630, Université de Poitiers,
B.P 30179, F-86962 Futuroscope Chasseneuil Cedex, France

PACS numbers: 61.72.Cc,61.72.Ji,61.80.Jh

Submitted to: *J. Phys.: Condens. Matter*

Abstract. The recombination of Frenkel pairs resulting from low energy recoils in 3C-SiC has been investigated using first principles and Nudged Elastic Band calculations. Several recombination mechanisms have been obtained, involving direct interstitial migration, atoms exchange, or concerted displacements, with activation energies ranging from 0.65 eV to 1.84 eV. These results are in agreement with experimental activation energies. We have determined the lifetime of the $V_{Si} + Si_{TC}$ Frenkel pair, by computing phonons and the Arrhenius prefactor. The vibrational contributions to the free energy barrier have been shown to be negligible in that case.

1. Introduction

Silicon carbide has been extensively studied and used in various applications, due to its unique physical, chemical and mechanical properties [1]. In electronics, SiC is a possible replacement for silicon in high temperature, high power and high frequency devices. In radioactive environment, possible uses concern fusion reactors, confinement matrices or spatial electronics. A good knowledge of the crystalline SiC behaviour during and after irradiation is a prerequisite for all these applications. In fact, during irradiation, lattice atoms are displaced, resulting in the formation of structural defects such as interstitials and vacancies. The damage is accumulating in the material, possibly leading to mechanical and electrical properties deterioration, and even amorphization in the case of large doses or low temperatures [2]. Also, only ion implantation is a viable option to dope SiC-based electronic devices, since dopants have high migration energies in silicon carbide, preventing the use of conventional thermal diffusion techniques. The accumulation of damage in SiC due to ion implantation has been largely studied [3, 4].

In order to anneal these defects and recover a good crystalline quality, thermal treatments have been applied during or after irradiation. The defects concentration

will change according to temperature and irradiation parameters, as a dynamic process of defect creation and recombination. To better understand the materials behaviour or the crystal recovery, after irradiation or annealing treatment, knowledge of the stability and mobility properties of defects is required. In particular, important quantities are the activation energies for recombining vacancies and interstitials from the Frenkel pairs formed in irradiated materials. Experimentally, it has been shown that distinct recovery stages exist as a function of temperature and of the level of damage, for 4H-SiC [5]. In the case of 6H-SiC, isochronal and isothermal annealings indicate three recovery stages as a function of temperature, with activation energies 0.3 ± 0.15 eV (150-300 K), 1.3 ± 0.25 eV (450-550 K), and 1.5 ± 0.3 eV (570-720 K) [6, 7]. This last result is in accordance with a study of dynamic annealing in 4H-SiC, yielding an activation energy of 1.3 eV for recovery with a temperature ranging from 350 K to 430 K [8]. Finally, in the high temperature domain, Itoh *et al.* have shown that irradiation-induced centers in 3C-SiC are fully annealed above 1020 K, with a measured activation energy of 2.2 ± 0.3 eV [9].

The available informations are not complete enough to draw a full and comprehensible picture of the defects annealing and crystal recovery of silicon carbide. It is worth to note that cascade simulations reveal that most of the formed defects due to irradiation are Frenkel pairs with very few clustered defects and antisites [10, 11]. Two mechanisms are then mainly responsible for defect annealings, the first being short range recombination of Frenkel pairs, and the second long range migration of point defects. The latter has been largely studied, and the activation energies for single point defects have been calculated using quantum mechanical approaches [12, 13, 14]. Less knowledge has been accumulated for the former process, principally because of the large number of possible Frenkel pair configurations. Using classical molecular dynamics, Gao and Weber have determined that activation energies for Frenkel pairs recombination range from 0.22 eV to 1.6 eV [15]. Another study yields an energy barrier of 1.16 eV for the recombination of V_{Si} and $SiSi$ dumbbell [16, 17]. These results suffer from the poor accuracy of classical potentials for describing transition states. Using more accurate Density Functional Theory (DFT) methods and transition states calculation techniques, Bockstedte *et al.* [18] and Rauls [13] have investigated the recombination mechanisms and the associated activation energies. However, the few selected configurations have been built by associating an interstitial with a vacancy in an arbitrary way.

Using first principles DFT calculations, we have investigated the recombination of Frenkel pairs with short interstitial-vacancy distances in 3C-SiC. The recombination mechanism and the associated activation energies have been determined with the Nudged Elastic Band (NEB) method [19, 20]. The Frenkel pairs considered in this work have been obtained during *ab initio* molecular dynamics of low energy recoils [21]. These configurations are then supposed to provide a realistic description of Frenkel pairs occurring in irradiated SiC, in the case of a low density of defects. After a short report of our results on the stability of Frenkel pairs, the calculated mechanisms and energy barriers for recombination are described. The lifetime of one Frenkel pair has been estimated, taking into account vibrational entropy contributions. Finally we discuss

our results in relation with experiments and previous studies.

2. Methods

We employed the plane-wave pseudopotential code PWscf included in the Quantum-Espresso package [22]. Our calculations have been performed in the framework of Density Functional Theory [23, 24] and the Generalized Gradient Approximation (GGA-PBE) as parametrized by Perdew, Burke and Ernzerof [25]. Vanderbilt ultrasoft pseudopotentials have been used for describing electron-ion interactions. We found that with these pseudopotentials, a plane wave basis with an energy cutoff of 25 Ry is enough to get difference energies converged to 0.01 eV. The silicon carbide lattice parameter is $a_0 = 4.382$ Å. Defect configurations are modelled with periodically repeated supercells. Single defects and Frenkel pairs stability have been calculated in cells encompassing 216 and 64 atoms. The Brillouin zone sampling was performed using the Γ point only for large cells, and a $4 \times 4 \times 4$ special k-points mesh for small cells [26]. For recombination investigations, smaller cells including 64 or 96 atoms have been considered, as well as $4 \times 4 \times 4$ (64 atoms) or $2 \times 2 \times 2$ (96 atoms) special k-points meshes, in order to keep reasonable computational times. In this work, we considered defects in their neutral charge state.

Two different approaches may be used for studying the recombination of Frenkel pairs. The first, based on molecular dynamics and Arrhenius plots, allows to determine recombination without prior knowledge of the path. Unfortunately, energy barriers for Frenkel pairs recombination in silicon carbide are expected to be large. The probability to obtain a successful event is then low, and very long molecular dynamics runs at high temperature are then required, preventing first principles molecular dynamics calculations. It is also not clear whether the harmonic approximation remains valid for high temperatures, or whether the right recombination mechanisms are activated. The second method, employed in this work, is the determination of the minimum energy path for recombination. We used the NEB technique [19], with three images between initial (Frenkel pair) and final (perfect crystal) states. The climbing image algorithm was also used to determine precisely the transition state [20]. While such a technique allows a precise calculation of the recombination mechanism as well as the associated activation energy, a guess of the initial path is required. Non-trivial mechanisms are then difficult to obtain.

3. Frenkel pairs stability

The Figures 1-5 shows the considered Frenkel pairs. Structural properties of these defects have already been described in a previous publication [27]. In the Table 1 the defect formation energies are reported, calculated according to the usual formalism [28], for two different cell sizes. Among all Frenkel pairs, $V_{C+CSi_{[100]}}$ has the lowest formation energy. More generally, Frenkel pairs including C vacancies have the lowest formation

	d_{FP}	216 atoms		64 atoms	
		E_f	ΔE	E_f	ΔE
$V_C + CC_{[100]}$	0.85	9.90	-0.03	10.69	-0.17
$V_C + CSi_{[100]}$	0.5	6.69	-3.24	7.06	-3.80
$V_C + CSi_{[100]}$	0.95	9.96	0.03	10.86	0.00
* $V_{Si} + Si_{TC}$	1.5	14.08	-0.44	14.92	-1.56
$V_{Si} + Si_{TC}$	0.9	13.46	-1.06	13.26	-3.22

Table 1. Calculated formation energy E_f (in eV) of several Frenkel pairs, for two cell sizes. d_{FP} (in a_0) is the interstitial-vacancy distance, and ΔE is the formation energy difference between a Frenkel pair and individual point defects. A larger cell (96 atoms) has been considered for the configuration * due to the large vacancy-interstitial distance.

energies, as expected regarding the low formation energies of C defects (both vacancy and interstitials) compared to Si defects [27]. It would be interesting to compare these energies, obtained in the framework of GGA, with previous calculations performed using Local Density Approximation (LDA). Unfortunately, the Frenkel pair formation energies are often not reported in the available literature. Rauls points to energies of 10-11 eV for $V_C + CC$ pairs depending on the distance [13], in agreement with our results.

In the Table 1 is also reported the difference between the formation energy of Frenkel pairs and the sum of the formation energies of individual defects. This quantity allows to compare the Frenkel pair stability against dissociation. Almost all differences are negative, suggesting that Frenkel pairs are more stable than single defects. Also, large energy differences are mostly associated with short vacancy-interstitial distances, indicating an attractive interaction between vacancies and interstitials [27].

One important aspect is the influence of cell size on the stability of Frenkel pairs. In the Table 1 is reported the formation energies for cells including 216 or 64 atoms. Differences as large as 0.9 eV are obtained. We have observed a similar effect for single point defects, with larger formation energies for the carbon vacancy, and interstitials. The biggest difference is obtained for silicon interstitials, suggesting that a 64 atoms cell is too small to allow the full lattice distortion around the defect. Although present, this effect is less pronounced for Frenkel pairs, since there is a better accommodation of strains thanks to the vacancy. Using small cells, one may then expect an overestimation of the energy barrier for creating the Frenkel pair from the perfect crystal. Nevertheless, it is reasonable to assume much smaller differences for the recombination energy barrier, since similar lattice distortions are expected for the Frenkel pair configuration and the transition state. At most, lattice distortion should be larger for the transition state, leading to a slight overestimation of the recombination barriers.

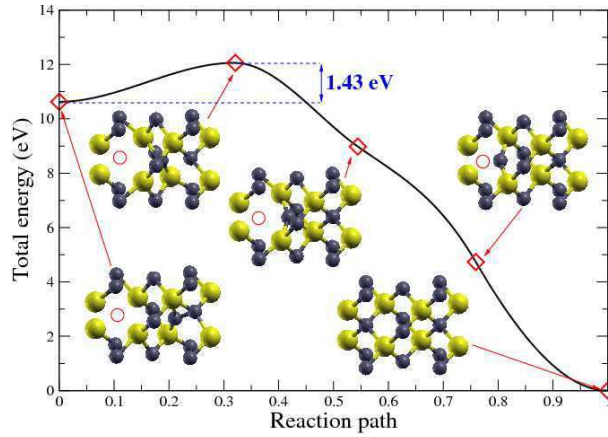


Figure 1. Minimum energy path for the recombination of the Frenkel pair $V_C + CC_{[100]}$ ($d_{FP} = 0.85a_0$). The insets show the relaxed configurations along the minimum energy path. Light (dark) spheres are silicon (carbon) atoms, while the empty circle marks the position of the vacancy.

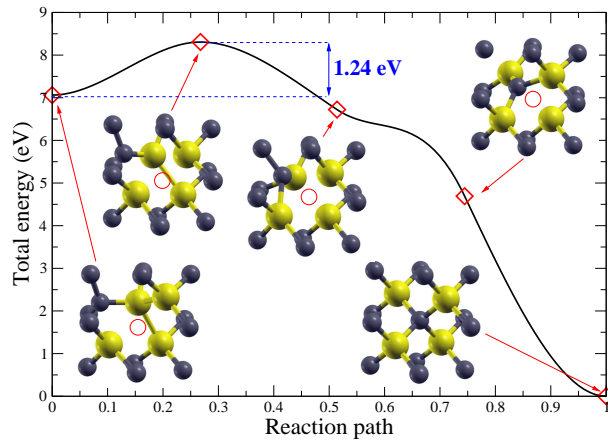


Figure 2. Minimum energy path for the recombination of the Frenkel pair $V_C + CSi_{[100]}$ ($d_{FP} = 0.5a_0$). See legend of Figure 1 for further details.

4. Frenkel pairs recombination

The Figure 1 shows the recombination path and the energy barrier for the Frenkel pair $V_C + CC_{[100]}$ ($d_{FP} = 0.85a_0$). The carbon interstitial migrates along the [100] direction until it recombines with the vacancy. The energy barrier is 1.43 eV, which is in the upper part of the experimental data.

The second studied Frenkel pair is $V_C + CSi_{[100]}$ ($d_{FP} = 0.5a_0$). In that case, the recombination path is not direct, and involves the concerted displacement of the carbon interstitial together with the silicon atom forming the dumbbell (Figure 2). The calculated energy barrier is 1.24 eV, again in the upper part of the available experimental data. This calculated path and energy barrier are in agreement with a previous calculation, yielding a similar mechanism and an energy barrier of 1.4 eV for

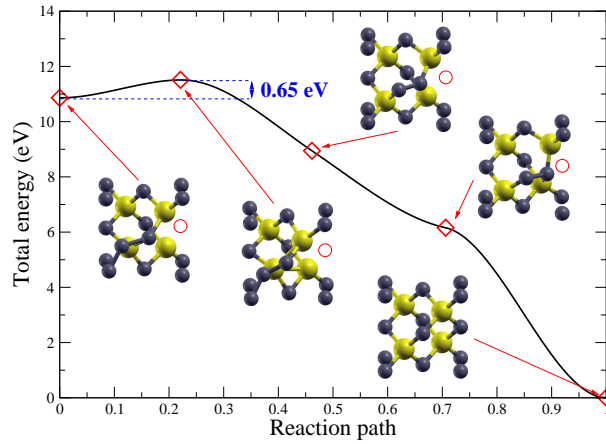


Figure 3. Minimum energy path for the recombination of the Frenkel pair $V_C + CSi_{[100]}$ ($d_{FP} = 0.95a_0$). See legend of Figure 1 for further details.

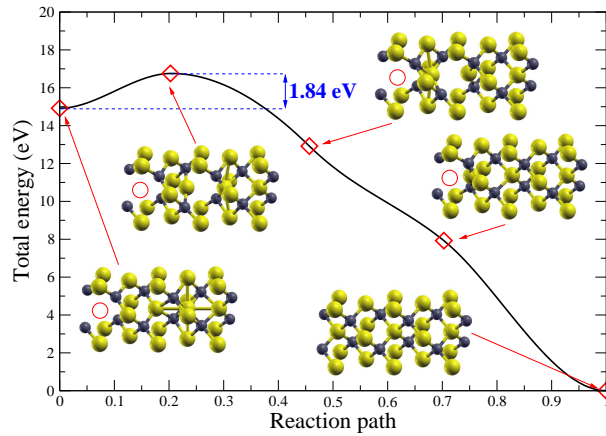


Figure 4. Minimum energy path for the recombination of the Frenkel pair $V_{Si} + Si_{TC}$ ($d_{FP} = 1.5a_0$). See legend of Figure 1 for further details.

an equivalent Frenkel pair configuration [18].

The third configuration, $V_C + CSi_{[100]}$ is close to the previous one, with a larger interstitial-vacancy distance ($d_{FP} = 0.95a_0$). Surprisingly, the energy barrier is much lower in this case, with 0.65 eV (Figure 3). This energy is required for the first step of the recombination process, with the transformation of the CSi dumbbell to a CC dumbbell, very close to the vacancy. This CC defect then recombines very easily with the vacancy. Such a result is in agreement with a previous calculation showing a very low recombination energy (0.2 eV) for this last step [18].

The Figure 4 shows the recombination of the Frenkel pair $V_{Si} + Si_{TC}$ ($d_{FP} = 1.5a_0$) according to an exchange mechanism. The silicon interstitial replaces another silicon atom, located between the vacancy and the interstitial original position. This silicon atom is filling the vacancy. The whole recombination process is associated with a large energy barrier of 1.84 eV. It is likely that another mechanism, more complex and less