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# Influence of domain wall interactions on nanosecond switching in magnetic tunnel junctions

J. Vogel,<sup>1</sup> W. Kuch,<sup>2,\*</sup> R. Hertel,<sup>3</sup> J. Camarero,<sup>4</sup> K. Fukumoto,<sup>2,\*</sup> F. Romanens,<sup>1</sup>  
S. Pizzini,<sup>1</sup> M. Bonfim,<sup>5</sup> F. Petroff,<sup>6</sup> A. Fontaine,<sup>1</sup> and J. Kirschner<sup>2</sup>

<sup>1</sup>Laboratoire Louis Néel, CNRS, 25 avenue des Martyrs, B.P. 166, F-38042 Grenoble Cedex 9, France

<sup>2</sup>Max-Planck-Institut für Mikrostrukturphysik, Weinberg 2, D-06120 Halle, Germany

<sup>3</sup>Institute of Solid State Research (IFF), Research Center Jülich, D-52425 Jülich, Germany

<sup>4</sup>Dpto. Física de la Materia Condensada, Universidad Autónoma de Madrid, E-28049 Madrid, Spain

<sup>5</sup>Departamento de Engenharia Elétrica, Universidade do Paraná, CEP 81531-990, Curitiba, Brazil

<sup>6</sup>Unité Mixte de Physique CNRS/Thales, Route départementale 128,  
F-911767 Palaiseau, France, and Université Paris-Sud, F-91405 Orsay, France

We have obtained microscopic evidence of the influence of domain wall stray fields on the nanosecond magnetization switching in magnetic trilayer systems. The nucleation barrier initiating the magnetic switching of the soft magnetic Fe<sub>20</sub>Ni<sub>80</sub> layer in magnetic tunnel junction-like FeNi/Al<sub>2</sub>O<sub>3</sub>/Co trilayers is considerably lowered by stray fields generated by domain walls present in the hard magnetic Co layer. This internal bias field can significantly increase the local switching speed of the soft layer. The effect is visualized using nanosecond time- and layer-resolved magnetic domain imaging and confirmed by micromagnetic simulations.

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The active part of devices like spin-valves and magnetic tunnel junctions, used in magnetic random access memories (MRAM), consist of an ultrathin soft magnetic layer and a harder magnetic layer separated by a non-magnetic spacer layer. These devices rely on the fast switching of the magnetization of the soft layer for reading or writing separate bits of information. Micromagnetic interactions have a strong influence on this switching. Demagnetizing effects and stray fields at the edges of nanosized magnetic structures can influence the magnetic configuration and the magnetization reversal of the soft magnetic layer, but interface roughness can also play a role and induce a magnetostatic coupling with the underlying hard magnetic layer. Much larger, but more localized magnetostatic effects exist when a domain wall is present in the hard magnetic layer [1, 2]. Direct evidence of the influence of domain wall stray fields in one layer on the static domain configuration of another layer has been obtained by Kuch *et al.* [3] on Co/Cu/Ni trilayers using x-ray photoelectron emission microscopy (X-PEEM). Schäfer *et al.* [4, 5] have used Kerr microscopy to show the effect of stray fields of Bloch domain walls in a Fe whisker on the magnetization of a thin Fe film through a MgO spacer. Similar effects were recently also observed in systems with perpendicular magnetization [6]. Thomas *et al.* [7] have observed that repeated motion of domain walls in the soft magnetic layer of a soft magnetic/nonmagnetic/hard magnetic trilayer can demagnetize the hard magnetic layer, even if the coercive field of the hard layer is several times larger than the field used for the reversal. In thin films with in-plane uniaxial anisotropy the static coercivity is usually determined by the field needed for domain nucleation. In FM/NM/FM trilayers the stray field of a domain wall in the hard magnetic layer can locally decrease this quasi-

static nucleation field in the soft magnetic layer [1]. Here, we show a first direct, real-time observation of this effect by studying the magnetization reversal dynamics of the soft FeNi layer in the presence of a domain wall in the harder Co layer in Fe<sub>20</sub>Ni<sub>80</sub>/Al<sub>2</sub>O<sub>3</sub>/Co trilayers, taking advantage of the layer selectivity of X-PEEM combined with x-ray magnetic circular dichroism (XMCD-PEEM). Our micromagnetic simulations show that the stray field of domain walls in the Co layer locally tilts the magnetization of the Fe<sub>20</sub>Ni<sub>80</sub> (FeNi in the following) layer in the direction perpendicular to the easy axis, opposite to the direction of the magnetization in the core of the Co domain wall. The Co domain wall stray field acts thus locally as an effective transverse bias field [8]. This internal bias field decreases the nucleation barrier and can considerably increase the local switching speed of the soft layer.

The measurements were performed on a Fe<sub>20</sub>Ni<sub>80</sub>(4nm)/Al<sub>2</sub>O<sub>3</sub>(2.6nm)/Co(7nm) trilayer deposited on Si(111) by RF sputtering. The Si substrate was miscut by 6° along the  $\bar{2}11$  direction, which after heat treatment [9] leads to a step-bunched surface presenting terraces in the shape of ellipses with an average length of about 1  $\mu$ m and a width of about 40 nm, separated by 6 nm high steps. Before depositing the trilayer, 3 nm of Co were deposited and then oxidized to form a layer of CoO. This layer served to increase the coercivity of the Co layer and allowed doing measurements with magnetic pulses strong enough to saturate the FeNi layer without changing the Co domain pattern. The alumina layer was obtained by depositing pure Al followed by a glow discharge under a 10 Pa O<sub>2</sub> plasma. The topographic steps in the substrate are transferred to the magnetic trilayers [10, 11], leading to an in-plane

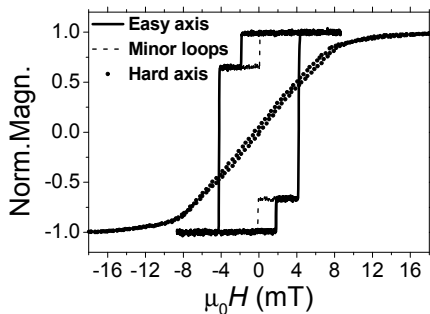


FIG. 1: Quasi-static hysteresis loops of the magnetic tunnel junction-like trilayer obtained by longitudinal Kerr effect. Loops obtained with the field applied along the easy (continuous line) and hard (dotted line) magnetization axes are shown, as well as minor loops for the FeNi layer taken along the easy axis (dashed line).

uniaxial magnetic anisotropy with the easy axis along the long axis of the terraces. The steps at the end of the terraces induce a magnetostatic Néel orange-peel coupling [11–13] between the two magnetic layers through the spacer layer. In Fig. 1 we present magnetization loops of the sample, obtained by longitudinal Kerr effect measurements, for fields applied either along (easy axis) or perpendicular (hard axis) to the steps. Minor loops for the permalloy layer show a shift of about 1 mT with respect to zero field due to the orange-peel coupling. The squareness of the loops indicates that in quasi-static conditions the reversal takes place through the nucleation of one or several reversed domains and a subsequent fast propagation of the generated domain walls. The coercivity is determined by the nucleation barrier.

Time- and layer-resolved magnetic domain images were obtained combining XMCD-PEEM and a stroboscopic acquisition mode [14–16]. The used microscope was a commercial Focus IS-PEEM and the experimental setup was identical to the one described in previous publications [17]. Magnetic pulses provided by a small coil mounted directly on the sample were synchronized with the x-ray photon pulses [14], with a repetition rate of 625 or 312.5 kHz. The measurements were performed on beamline UE56/2-PGM2 at the BESSY synchrotron in Berlin (Germany). The FeNi domain structure was imaged by tuning the x-ray energy to the Fe  $L_3$  absorption edge (707 eV), while for the Co layer the Co  $L_3$  edge energy (778 eV) was used.

In order to study the fast switching of the FeNi layer and the influence of the Co domain walls thereon, we first induced a domain structure in the Co layer using a 3 ms pulse with an amplitude of about 10 mT. The resulting domain structure is shown in Fig. 2(l). White and black regions correspond to domains with an in-plane magnetization direction pointing parallel and anti-parallel to the black arrow, respectively. Bipolar magnetic pulses

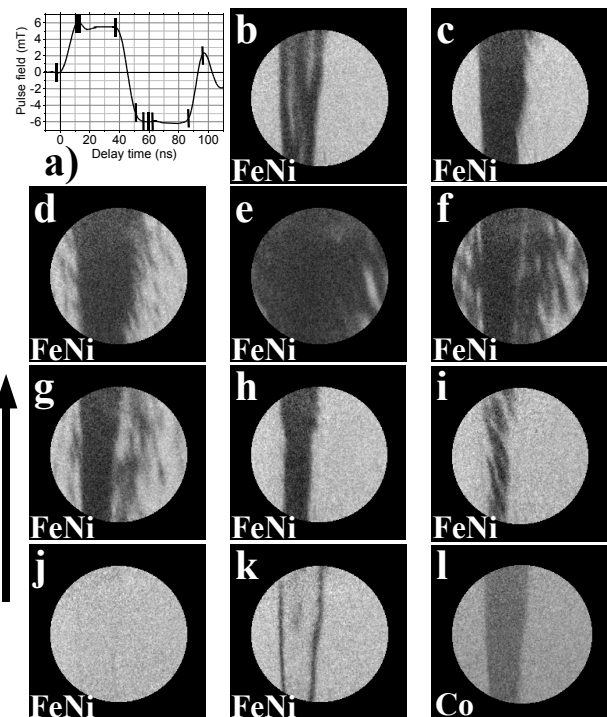


FIG. 2: Time and layer-resolved XMCD-PEEM images of the magnetization state of the FeNi [(b)-(k)] and Co layers (l). The field of view in these images is about  $100 \mu\text{m}$  and the spatial resolution  $1 \mu\text{m}$ . The projection of the x-ray incidence direction on the sample surface is pointing up in the images (parallel to the arrow) and is parallel (anti-parallel) to the direction of the field for positive (negative) pulses. The magnetization direction is in the plane of the layers and points up (parallel to the arrow) for black domains, and down for white domains. The FeNi images were taken for delays between photon and magnetic pulses of -3, 11, 13, 37, 51, 56, 60, 62, 87 and 97 ns, as indicated in (a). The Co image was taken for a delay of 60 ns.

with amplitude of about 6 mT and a length of 40 ns for both positive and negative pulses were then applied to the sample parallel to the easy magnetization axis. Images recorded for the FeNi layer in pump-probe mode, for different delays between photon and magnetic pulses, are shown in Fig. 2(b)-2(k). The corresponding positions of the photon pulses with respect to the magnetic pulses are given in Fig. 2(a). Before the pulses [Fig. 2(b)] the domain structure in the FeNi layer is strongly correlated to the one in the Co layer [Fig. 2(l)], due to the rather strong orange-peel coupling that tends to align the magnetization direction in the two magnetic layers. The small overshoot at the end of the negative pulse is not sufficient to completely align the magnetization in the FeNi layer above the black Co domain with the underlying Co magnetization before the pulse. When the field increases, propagation of the domain walls in the FeNi layer takes place first [Fig. 2(c)]. At the maximum of the positive pulse some newly nucleated reversed domains

also become visible [Fig. 2(d)]. At the end of the positive pulse, the FeNi layer is almost saturated except for some remaining white domains at the right and left bottom [Fig. 2(e)]. When the field is reversed, starting from this nearly saturated state, new white domains appear during the rising flank of the pulse [Fig. 2(f)] above the white domains in the Co layer and above the Co domain walls [Fig. 2(f,g)]. Nucleation and propagation of domain walls is faster above the white domains in the Co layer, due to the orange peel coupling leading to a higher effective field (external plus coupling fields). On the plateau of the negative pulse, the FeNi above the black Co domain switches, initially by propagation of domain walls [Fig. 2(h)] and a few nanoseconds later also by nucleation of some new white domains [Fig. 2(i)]. At the end of the negative pulse [Fig. 2(j)], the FeNi layer seems completely saturated in the white direction. Closer inspection actually shows that faint grey lines are still present at the position where domain walls are present in the Co layer. This shows that the Co domain wall stray field is strong enough to prevent saturation of the FeNi layer even for a field of 6 mT, which is about three times the static coercivity for the FeNi layer. On the other hand when the field direction is changed, preferential nucleation of reversed domains takes place above the domain wall in the Co layer, as seen in Fig. 2(k). This effect also takes place going from e) to g) but it is less clear since nucleation centers become visible only after some time, when they have expanded through domain wall propagation to a size that is large enough to be visible with our microscope. Nucleation processes in the FeNi layer may occur on top of the Co domain walls, but at higher fields also in the middle of existing Co domains. If the field is increased very quickly, the corresponding critical fields are reached almost at the same time, so that the difference between the two nucleation fields is difficult to see (cf. 2(f,g)). In 2(k), the field was increased more slowly and to lower values, making this difference in nucleation field (and time) much more visible.

Our nanosecond time- and layer-resolved domain images thus reveal the strong influence of domain walls in the Co layer on the nanosecond reversal of the FeNi layer. We did several measurement series on the same spot of the sample, with different domain patterns induced in the Co layer. Preferential nucleation always takes place above Co domain walls and is therefore not simply caused by topographic features in the sample. Other samples, with different miscuts and anisotropies showed the same qualitative results. This clearly indicates that the Co domain wall stray field locally decreases the barrier for nucleation of reversed domains in the adjacent FeNi layer, therefore increasing the local switching speed.

The effect of stray fields of domain walls in one layer on the magnetization of the other layer in FM/NM/FM trilayers has been treated quantitatively by several authors [1, 3, 5, 7, 18]. In most of these cases, the domain

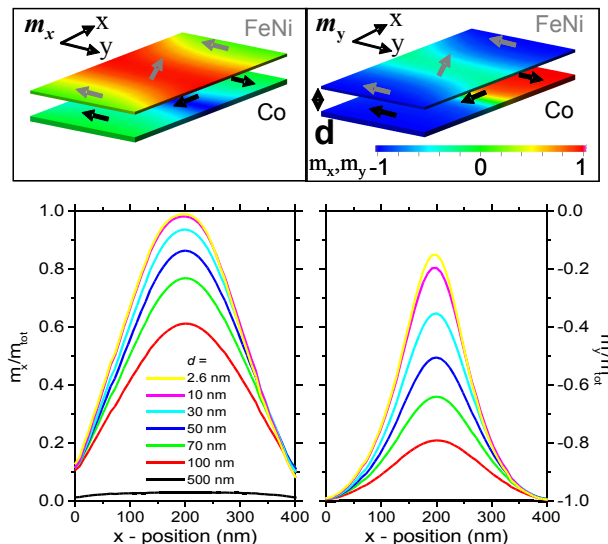


FIG. 3: Top: Simulations of a domain wall in the Co layer and its influence on the FeNi magnetization, for a spacer thickness  $d$  of 30 nm. In the top left panel, the component of the magnetization along  $x$  ( $m_x$ ) is shown color-coded for the Co (bottom) and FeNi (top) layers, while in the top right panel the  $y$ -component ( $m_y$ ) is given. The rectangles represent only a small fraction of the three-dimensional finite-element mesh, most of which has been removed to uncover the region of interest. Bottom: X-component (left) and y-component (right) of the FeNi magnetization along  $x$ , in the middle of the simulated region, for different distances  $d$  between the Co and FeNi layers.

wall was treated as a homogeneously magnetized region of width  $W_{DW}$  and with a magnetization perpendicular to the overall magnetization direction. In order to get a more precise idea of the influence of stray fields emitted by real domain walls, we have performed micromagnetic simulations using a code based on a combination of the finite element method (FEM) and the boundary element method (BEM). This FEM/BEM scheme is particularly suited to simulate magnetostatic interactions of ferromagnetic particles, as described in Ref. [19]. The magnetic structures are obtained by energy minimization. For the magnetic saturation polarization  $J_S = \mu_0 M_S$  and the exchange constant  $A$ , values were taken of 1 T and 13 pJ/m for FeNi, and 1.76 T and 30 pJ/m for Co, respectively. Experimentally determined values of 1720 J/m<sup>3</sup> (FeNi) and 11200 J/m<sup>3</sup> (Co) were used for the uniaxial anisotropy constant  $K$ . The magnetization was simulated in platelets with a width of 200 nm along the easy magnetization axis ( $y$ -axis) and 400 nm perpendicular to this axis ( $x$ -axis). To mimic extended strips, free boundary conditions along  $y$  were obtained geometrically by connecting the edges of the platelet to form a ribbon shape. Two domains with opposite magnetization directions along the easy axis were introduced in the Co layer, resulting in a Néel-type domain wall. The FeNi

layer in the simulations was initially homogeneously magnetized along  $y$ , and then its magnetization was relaxed to reach equilibrium. The final magnetization directions are indicated with grey and black arrows for the FeNi and Co layers, respectively. The results of the simulations for a 30 nm thick spacer layer are shown in the top panels of Fig. 3. In the top left panel the  $x$ -component of the magnetization ( $m_x = M_x/M_S$ , with  $M_S$  the saturation magnetization) is given, while the top right panel represents the  $y$ -component ( $m_y = M_y/M_S$ ). The color code ranges from blue ( $m_{x,y} = -1$ ) to red ( $m_{x,y} = 1$ ) for all layers, with  $m_{x,y} = 0$  given by green.

The simulated width of a domain wall in the Co layer using Lilley's definition [20] is about 70 nm. The influence of the domain wall on the magnetization of the FeNi layer decreases with increasing separation between the FM layers but is considerable up to spacer thicknesses as large as 100 nm.

The main result of the simulations is that above the Co domain wall the magnetization in the FeNi layer is strongly tilted in the direction perpendicular to the easy magnetization direction, opposite to the magnetization direction in the center of the wall. The profile of the  $x$ -component of the FeNi magnetization for different separations between the two layers is shown in the bottom left panel of Fig. 3, while the  $y$ -component is given in the bottom right panel of Fig. 3. The  $x$ -component of the magnetization does not completely go to zero even at a distance of 200 nm of the center of the Co domain wall. The  $y$ -profile shows that the magnetization above the center of the Co DW is not perfectly perpendicular to the easy axis but makes an angle  $i$ . This tilt angle  $i$  depends on the separation between the two layers, and a value of  $81^\circ$  with respect to the easy axis is found for a separation of 2.6 nm (the experimental thickness of the alumina layer in our sample). Our simulations show that the induced magnetization angle in the FeNi layer also has an influence on the shape of the Co domain wall: it becomes wider when the interlayer distance gets smaller.

The integrated width of the region with tilted FeNi magnetization, taken from the simulated  $m_y$  curve for a separation of 2.6 nm between the layers, is about 150 nm. From the time-resolved magnetic domain image in Fig. 2(g) we can get an experimental estimate of this width. Taking into account the experimental resolution of  $1 \mu\text{m}$  and the experimental contrast, we find a value of about 250 nm, which agrees reasonably well with the simulations. We can not exclude, however, that more complicated structures, like  $360^\circ$  domain walls, are formed above the Co domain walls in the experiment.

In conclusion, the combination of nanosecond time-resolved XMCD-PEEM measurements and micromagnetic simulations has allowed observing and explaining the preferential nucleation of reversed FeNi domains above Co domain walls in FeNi/Al<sub>2</sub>O<sub>3</sub>/Co trilayers. For magnetic field pulses applied along the easy magnetiza-

tion axis, the torque on the FeNi moments is very small when these are aligned anti-parallel to the applied field. Above the Co domain walls, the FeNi moments are tilted away from this axis and the torque acting on them is thus much larger. The Co domain wall stray field acts as a transverse bias field that locally decreases the energy barrier for nucleation, significantly increasing the local switching speed. The same principle has been exploited to obtain ultrafast magnetization switching in small magnetic structures, using an external transverse bias field [8] This finding is of importance also for current-induced domain wall propagation in trilayer systems, a subject that is widely investigated nowadays because of the potential use in MRAM systems. The effect of domain wall stray fields can be controlled by manipulating the width and position of the domain wall. This allows increasing the local speed and reproducibility of magnetic switching.

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\* Present address: Institut für Experimentalphysik, Freie Universität Berlin, Arnimallee 14, D-14195 Berlin, Germany.

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